

# MASTER'S THESIS

## Development of a Deep Silicon Phase Fresnel Lens Using Grayscale Lithography and Deep Reactive Ion Etching

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DEVELOPMENT OF A DEEP SILICON PHASE FRESNEL LENS  
USING GRAY-SCALE LITHOGRAPHY AND DEEP REACTIVE ION ETCHING

By

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## ABSTRACT

Title of Thesis: DEVELOPMENT OF A DEEP SILICON PHASE FRESNEL  
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REACTIVE ION ETCHING

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A phase Fresnel lens (PFL) could achieve higher sensitivity and angular resolution in astronomical observations than the current generation of gamma and hard x-ray instruments. For ground tests of a PFL system, silicon lenses must be fabricated on the micro-scale with controlled profiles to enable high lens efficiency. Thus, two MEMS-based technologies, gray-scale lithography and deep reactive ion etching (DRIE), are extended to create multiple controlled step heights in silicon on the necessary scale.

A Gaussian approximation is introduced as a method of predicting a photoresist gray level height given the amount of transmitted light through a gray-scale optical mask. Etch selectivity during DRIE is then accurately controlled by introducing an oxygen-only step to a standard Bosch cycle to produce the desired scaling factor between the photoresist and silicon profiles. Finally, a profile evaluation method is developed to calculate the expected efficiency of measured silicon profiles. Calculated efficiencies above 87% have been achieved.

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## TABLE OF CONTENTS

<b>List of Tables</b> .....	iv
<b>List of Figures</b> .....	v
<b>1 Introduction</b>	
1.1 Astronomical Imaging Systems.....	1
1.2 Fresnel Lenses.....	2
1.2.1 Fresnel Lens Derivatives.....	3
1.2.2 The proposed Fresnel Lens-based Telescope.....	4
1.3 Test Lens Considerations.....	6
1.3.1 Material Selection.....	6
1.3.2 Dimensions.....	8
1.4 Micro-Electro-Mechanical Systems (MEMS) Fabrication.....	9
1.4.1 Planar Fabrication.....	10
1.4.1.1 Photolithography.....	10
1.4.1.2 Surface Micromachining.....	13
1.4.1.3 Bulk Micromachining.....	14
1.4.1.4 Planar micro-scale Fresnel Lenses.....	17
1.4.2 3-Dimensional MEMS Fabrication.....	19
<b>2 Gray-scale Technology</b>	
2.1 Introduction.....	22
2.2 Gray-scale Lithography.....	23
2.2.1 Gray-scale Mask Design.....	24
2.2.2 Lithography Processing.....	26
2.2.2.1 Photoresist Selection.....	27
2.2.2.2 Exposure.....	28
2.2.2.3 Development.....	29
2.2.3 Calibration Mask.....	31
2.2.4 Standardized Lithography Process.....	33
2.3 Dry-anisotropic Etching.....	34
2.3.1 Deep Reactive Ion Etching.....	34
2.3.2 Gray-scale Pattern Transfer.....	35
2.3.3 Selectivity Control Experiments.....	37
2.4 Summary.....	41

<b>3</b>	<b>Optical Mask Design</b>	
3.1	Introduction.....	42
3.2	PFL Equations.....	42
3.3	PFL Mask Design considerations.....	45
3.3.1	Pitch Selection.....	46
3.3.2	Pixel Constraints.....	48
3.3.3	Choice of Gray Levels.....	50
3.3.4	Multiple Phase Depths.....	52
3.4	Gaussian Approximation Method.....	53
3.4.1	Experiment.....	54
3.4.2	Integration into C Program Design.....	57
3.5	PFL Device Design Parameters.....	59
3.6	Summary.....	60
<b>4</b>	<b>PFL Fabrication and Evaluation</b>	
4.1	Introduction.....	62
4.2	Lithography Results.....	63
4.2.1	Photoresist PFL Structures.....	63
4.2.2	PFL Metrology (Photoresist).....	68
4.2.3	Gaussian Approximation Confirmation.....	71
4.3	Dry Etching Results.....	72
4.3.1	General DRIE Results.....	73
4.3.2	DRIE with Oxygen-only Step.....	74
4.3.3	Aspect Ratio Dependent Etching.....	80
4.4	Profile Evaluation.....	84
4.4.1	Method.....	85
4.4.2	Profile Measurements.....	88
4.4.3	Lens Design Comparison.....	93
4.5	Summary.....	96
<b>5</b>	<b>Summary and Future Work</b>	
5.1	Summary.....	98
5.2	Future Work.....	101
5.2.1	ARDE Compensation.....	102
5.2.2	Bulk Silicon Removal.....	103
5.3	Conclusion.....	105
	<b>Bibliography.....</b>	<b>106</b>

## LIST OF TABLES

Table 1.1:	Three example PFL telescopes proposed by Skinner.....	5
Table 2.1:	List of gray-scale wedge structures designed on the calibration mask.....	33
Table 2.2:	The standard gray-scale lithography process.....	33
Table 2.3:	DRIE process parameters for Base Etch I.....	37
Table 2.4:	DRIE selectivity characterization results.....	39
Table 3.1:	Mock selection of pixels using a 2.2 pitch and fictional design rules.....	50
Table 3.2:	Gray level height data taken from a structures using a 2.8 $\mu$ m pitch.....	54
Table 3.3:	Properties of the 3 primary PFL's designed for this research.....	60
Table 4.1:	Modified Bosch process using an Oxygen-only step.....	75
Table 4.2:	Modulating the length of the O <sub>2</sub> etch step and the resulting etch selectivity.....	77
Table 4.3:	Theoretical versus calculated lens efficiencies for various lens profiles.....	87
Table 4.4:	Calculated efficiencies of a PFL with each radial scan taken at a different angle from the horizontal.....	89
Table 4.5:	Calculated efficiencies of PFLs at different locations across a 75mm wafer.....	92

## LIST OF FIGURES

Figure 1.1:	(a) Fresnel Zone Plate (FZP). (b) Phase Zone Plate (PZP) or “binary” lens. (c) Phase Fresnel Lens (PFL).....	3
Figure 1.2:	The thickness of different lens materials required to produce a $\pi$ phase shift [16].....	7
Figure 1.3:	The amount of photon absorption through a material with a phase shift of $\pi$ [16].....	7
Figure 1.4:	Photoresist patterns resulting from contact lithography using positive or negative resist with the same optical mask.....	12
Figure 1.5:	A projection lithography system with reduction optics (usually 5X or 10X).....	12
Figure 1.6:	Surface micromachining to release a beam. (a) Sacrificial layer is deposited and patterned. (b) Structural layer is deposited and patterned. (c) Sacrificial layer is selectively etched away leaving a free-standing beam.....	13
Figure 1.7:	Illustration of wet isotropic etching with a small amount of mask undercut.....	14
Figure 1.8:	Illustration of wet anisotropic etching, where a 54.7% angle is revealed between the (111) and the (100) planes in crystalline silicon.....	15
Figure 1.9:	Plasma etching mechanisms: (a) Physical sputtering. (b) Chemical etching. (c) Ion-assisted etching.....	16
Figure 2.1:	Photoresist gray levels after development, on a silicon substrate.....	22
Figure 2.2:	Optical mask design scheme using sub-resolution pixels and a constant sub-resolution pitch. By calculating the area of each pixel and the area of the pitch, the percent transmission through the mask is estimated.....	25
Figure 2.3:	A photoresist gray-scale wedge structure after (a) over-development, (b) under-development, and (c) appropriate development.....	30

Figure 2.4:	SEM images showing two gray levels produced using two different pitches on the gray-scale mask; (a) shows a pitch slightly above resolution and (b) shows a smoother photoresist surface using a pitch right at the resolution limit.....	31
Figure 2.5:	The steps of deep reactive ion etching (DRIE). (a) Silicon wafer is patterned with a masking material. (b) Etching step of the cycle etches the silicon similar to RIE. (c) Conformal passivation layer is deposited over entire wafer. (d) Etch step is repeated. Ion bombardment removes the passivation layer from horizontal surfaces, while sidewalls remain protected. (e) Passivation step is repeated to cover the newly exposed sidewall. (f) Etch step repeats. (g) Vertical trench achieved with slight sidewall scalloping.....	35
Figure 2.6:	(a) Initial sloped photoresist structure on silicon. (b) Sloped pattern begins to transfer into the silicon with a certain selectivity. (c) Final structure in silicon retains lateral dimensions while vertical dimensions are amplified by the etch selectivity.....	36
Figure 2.7:	SEM showing the structures used to measure the inside sidewall angle and outside sidewall angle.....	38
Figure 3.1:	Example PFL profile with a focal length of 118m for 8.4keV photons.....	43
Figure 3.2:	The $4\pi$ phase depth shifts every other grating vertically. This effectively doubles the width of each grating.....	44
Figure 3.3:	An example pixel layout near the boundary of one ridge.....	46
Figure 3.4:	Rectangular pixilation schematic used to define mask design constraints.....	49
Figure 3.5:	Gaussian curve fit through data points relating normalized photoresist heights to the percent transmission through the mask that caused them.....	55
Figure 4.1:	The first demonstration of a photoresist PFL.....	64
Figure 4.2:	PFL ridge exhibiting a digital nature to the curved gray levels. Location on overall PFL is shown in Fig. 4.1 as a red box.....	65
Figure 4.3:	Relative intensity profiles made with $\sim 5$ pixels for (a) horizontal ridge geometries and (b) curved ridge geometries.....	67

Figure 4.4:	SEM image of small curved ridges in photoresist resulting from the non-uniform intensity transmission exhibited in Fig. 4.3(b).....	68
Figure 4.5:	Example 3-D photoresist measurement taken with a contact profilometer.....	69
Figure 4.6:	Locations measured to investigate the accuracy of the Gaussian approximation method.....	71
Figure 4.7:	PFL photoresist profiles fabricated using the Gaussian approximation method closely mimic the desired profile.....	72
Figure 4.8:	Initial pattern transfer of a 90° PFL wedge with a high selectivity.....	74
Figure 4.9:	The steps of a modified Bosch process. (a) Initial photoresist structure on silicon. (b) High selectivity recipe etches silicon while photoresist is virtually untouched. (c) Oxygen-only plasma step removes a layer of photoresist but does not etch the silicon. (d) Conformal passivation layer is deposited over the entire wafer, and the process is repeated.....	76
Figure 4.10:	Fully etched silicon PFL fabricated using an etch selectivity of 14:1 achieved through the addition of the Oxygen-only step to the standard Bosch process.....	78
Figure 4.11:	Optical profilometer measurement of center ridge of a silicon PFL. Height data was taken around the circumference of the top gray level, marked in red.....	78
Figure 4.12:	Close up picture of the PFL shown in Fig. 4.11.....	79
Figure 4.13:	Narrow ridges on test structures did not etch as deeply as the open area and the larger ridges.....	81
Figure 4.14:	Optical profilometer measurement of thin ridges compared to the ideal case.....	82
Figure 4.15:	Etch depths measured for trenches with varying widths. Smaller features etch slower, resulting in a reduced etch depth .....	83
Figure 4.16:	Schematic used as the basis for developing a method to calculate the expected efficiency of a measured silicon profile.....	85
Figure 4.17:	Phasor addition. Normalized resultant magnitude determines efficiency.....	87

Figure 4.18:	Optical profilometer measurement of a silicon PFL. 2-D scans of arbitrary length and orientation may be output into a text file.....	89
Figure 4.19:	Silicon PFL etch with a higher etch selectivity than desired. Calculated efficiency of this lens was 19.3%.....	90
Figure 4.20:	Silicon PFL etch with a selectivity slightly lower than desired, entire photoresist pattern is transferred too early. Calculated efficiency for this lens was 68.7%.....	91
Figure 4.21:	Silicon PFL etched with the appropriate etch selectivity compared to the ideal case. Calculated efficiency of this lens was 87.9%.....	91
Figure 4.22:	Optical profilometer measurement of a silicon lens 'A' design versus the ideal case. Calculated efficiency of this lens was 66.5%....	94
Figure 4.23:	Optical profilometer measurement of a silicon lens 'B' design versus the ideal case. Calculated efficiency of this lens was 81.7%.....	95
Figure 5.1:	Possible method of reducing aspect ratio dependent effects. (a) Thin ridges in an ideal photoresist pattern are not etched as deeply into the silicon as the larger ridges are etched. (b) An offset in the photoresist gives the thinner ridges a head start and all ridges etch to the appropriate depth.....	102
Figure 5.2:	Bulk silicon removal process. (a) Begin with Silicon-on-Insulator (SOI) wafer with device layer slightly thicker than desired lens thickness. (b) Fabricate PFL using gray-scale technology as described earlier, and grow a small thermal oxide layer. (c) Pattern oxide on backside of wafer and use TMAH wet etch to remove silicon bulk beneath the PFL. Buried oxide layer serves as an etch stop.....	104

# 1. Introduction

## 1.1. Astronomical imaging systems

High energy gamma rays can probe some of the most energetic phenomena in nature. Neutron stars, black holes, nuclei of active galaxies, and supernovae are only some of the many astrophysical sources that emit gamma-rays ( $>100\text{keV}$ ) on a consistent basis. Since Earth's atmosphere shields its surface from most gamma-rays, observations of these complex gamma ray sources must be made by space-bound telescope systems.

Given sufficient angular resolution, a space-bound gamma-ray telescope will have the ability to resolve the fine structure of astronomical objects and to separate multiple objects from each other. However, the sensitivity and angular resolution of current X-ray and gamma ray telescopes have suffered from the difficulty in constructing concentrating optics due to the inherent nature of this high-energy radiation. The poor performance can be largely attributed to high detector background noise due to particle interactions, induced radioactivity and diffuse sky emission. Also, to this point, gamma-ray telescope systems have lacked a method of concentrating the flux from a large collection area onto a small detector.

Current systems working with lower energy X-rays, such as Chandra [1], use grazing incidence optics to concentrate flux by 7 orders of magnitude but only offer an angular resolution of 0.5 arcseconds (far above the diffraction-limit), and are still limited to photon energies below  $\sim 10\text{keV}$ . The energy limits of grazing incidence optics are being pushed to 100 keV [2-4], but small grazing angles limit the effective areas feasible

and polishing tolerances inhibit extremely good angular resolution. Other X-ray missions under study [5] could considerably improve sensitivity and spectral resolution, however they will not emphasize improvements in angular resolution, which will remain  $>10^3$  times worse than the diffraction-limit, inhibiting their use for measurements requiring high-angular resolution. As the name implies, the Micro-Arcsecond X-ray Imaging Mission [6] (MAXIM) aims to improve angular resolution to the micro-arcsecond range through an interferometric approach. Unfortunately, to achieve a large entrance aperture, MAXIM's primary mirror segments must be placed on 32 separate spacecraft while the secondary mirror segments and detector array will be located on two more, making spacecraft control a difficult problem.

As the energies being considered increase, the capabilities of present instruments only get worse. The best angular resolution in the 100-1000 keV range offered by current imaging instruments (Integral [7], angular resolution of 12 arcminutes) only leads to a resolving power of approximately half the angular size of a full moon (31 arcminutes). Even the next generation high energy gamma ray observatory, the Gamma-ray Large Area Space Telescope [8] (GLAST), merely expects to achieve an angular resolution of  $<12,600$  arcseconds at 100 MeV and  $<540$  arcseconds at  $>10$ GeV. However, if diffraction-limited gamma ray optics could be constructed, even of modest diameter (1m), the achievable angular resolution would be sub-microarcsecond.

## **1.2. Fresnel lenses**

Recently, G. Skinner proposed a Fresnel lens-based system for astronomical observations at hard X-ray and gamma-ray energies [9,10]. This system would have the

highest diffraction-limited angular resolution of any wavelength band, resulting in a greater than  $10^8$  improvement over current gamma-ray imaging systems. The sensitivity of the proposed system would also be tremendous compared to typical background-limited gamma-ray instruments, resulting in a  $10^3$  improvement. (Improvements based upon comparison of a 5m Fresnel lens-based system to that of INTEGRAL [7].) The following sections will provide the necessary background on Fresnel lenses and their derivatives, as well as discuss the specifics of the Fresnel lens-based telescopes proposed by *G. Skinner*.

### 1.2.1. Fresnel lens derivatives

Invented by Soret in 1875, a Fresnel Zone Plate (FZP) (Fig. 1.1a) is a small grating that will diffract incident radiation towards its focus. As the off-axis (radial) distance is increased, higher deflection angles are required and thus the width of each zone of the diffraction grating becomes smaller. Yet an FZP is limited to an efficiency of 10.1% because the alternating opaque zones block half of the incident radiation and not all energy is transferred to the primary focus, some unfocused energy passes to higher

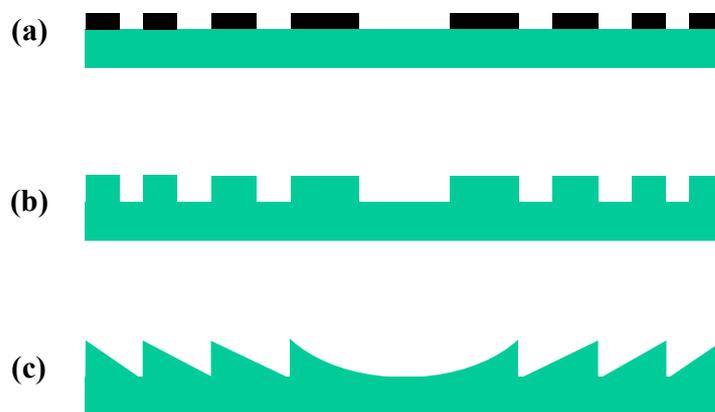


Fig. 1.1: (a) Fresnel Zone Plate (FZP). (b) Phase Zone Plate (PZP) or "binary" lens. (c) Phase Fresnel Lens (PFL).

order foci. The Phase Zone Plate (PZP), or “binary” lens (Fig. 1.1b), is an improvement on the FZP design since a phase shift of  $\pi$  is selectively introduced by replacing the opaque zones with regions of extra thickness of the lens material, raising the theoretical efficiency limit to 40.4%. Ideally, the induced phase change within each zone should be a continuous function of radius, which leads to the cuneiform, or Phase Fresnel Lens (PFL) (Fig. 1.1c), which will be composed of many sloped ridges of different widths. Thus, in a PFL, the diffraction grating is blazed using refraction to concentrate all incident power into the primary focus, resulting in a theoretical efficiency limit of 100%.

As intuition dictates, if a step approximation to the ideal case is necessary, a larger number of steps in the approximation will result in a lens with higher efficiency. Studies have been done regarding this improvement and a relation derived to predict the efficiency ( $\eta$ ) of a multi-level diffractive lens, given the number (N) of steps used [12-14]:

$$\eta_{Lens} = \left[ \frac{\sin\left(\frac{\pi}{N}\right)}{\left(\frac{\pi}{N}\right)} \right]^2. \quad (1.1)$$

Therefore, using (1.1), a binary lens (2 steps) has a maximum theoretical efficiency of 40.5%, in excellent agreement to with the 40.4% stated previously. If the number of steps is merely increased to 8, the maximum theoretical efficiency reaches 95.0%.

### **1.2.2. The proposed Fresnel Lens-based telescope**

As mentioned earlier, a PFL-based telescope could provide many orders of magnitude improvement in both angular resolution and sensitivity. This diffractive technique can be applied to a wide range of energies ( $\sim 10\text{keV}$  to  $10\text{MeV}$ ), and *Skinner*

	(a)	(b)	(c)
Energy (keV)	200	500	847
Max Thickness ( $\mu\text{m}$ )	450	1200	1900
Min period, $p_{min}$ ( $\mu\text{m}$ )	2500	1000	590
Focal length (km)	$10^6$	$10^6$	$10^6$
Theoretical diffraction limited angular resolution (arc sec)	$0.3 \times 10^{-6}$	$0.12 \times 10^{-6}$	$0.07 \times 10^{-6}$

*Table 1.1: Three example PFL telescopes proposed by Skinner.*

provides three example PFL designs for three different gamma ray energies. Aluminum was selected as his example material because it is an established technology and low cost material with expected transmission losses of less than 2%. The properties of the three example lenses are shown in Table 1.1.

Unfortunately, the focal length of each proposed PFL system will be quite large, on the order of  $10^6$  km, requiring that the lens and detector be located on 2 separate spacecraft and aligned appropriately. The long focal length would appear to make pointing the system difficult, but fortunately a detailed mission study [15] indicated that given current propulsion technology, a large focal length should not be prohibitive.

The second main drawback of the proposed PFL system is that it will suffer greatly from chromatic aberrations, meaning that they will perform well for only a small range of energies. This drawback comes from the fact that each section of the lens is specifically designed to focus a certain wavelength radiation to a common focal point through diffraction and refraction. As the incident radiation changes in wavelength, a given thickness of lens material will no longer provide the appropriate phase change and grating sizes will no longer be optimized, which together quickly degrade performance. More elaborate lens designs may help alleviate this problem [10], but the properties of a single lens must be investigated first.

As stated earlier, the proposed PFL system should provide a  $10^8$  increase in angular resolution and a  $10^3$  increase in sensitivity over current instruments. With this increased angular resolution, the proposed PFL system will be ideally suited to observing highly compact, and high surface brightness regions which cannot be sufficiently imaged with current technology.

### **1.3. Test lens considerations**

To first demonstrate the superior imaging properties of a PFL, a number of scaled down lenses must be developed for ground testing at lower energies, such as X-rays. NASA has proposed to carry out such testing at the Marshall Space Flight Center X-ray Calibration Facility (XRFC) in Huntsville, Alabama. The X-ray source at XRFC is located in a tunnel approximately 0.5km from the detector, with an access point to the center section of the tunnel located 400m from the source (to ensure a relatively collimated beam) and 118m from the detector. It is at this access point where a test lens must be mounted. Assuming a tungsten target as the X-ray source, the strongest emission lines of interest will be the L<sub>alpha</sub> line at 8.4keV and the K<sub>alpha</sub> line at 59.3keV. The sections that follow will discuss the selection of the test lens material and the approximate dimensions of a lens given the ground-based testing setup described above.

#### **1.3.1. Material Selection**

The PFL principle is heavily dependent upon the refractive index of the lens material and will dictate the appropriate thickness at each point on the lens. *G. Skinner* was able to show that the thickness ( $t_\pi$ ) to produce a phase shift of  $\pi$  is approximately

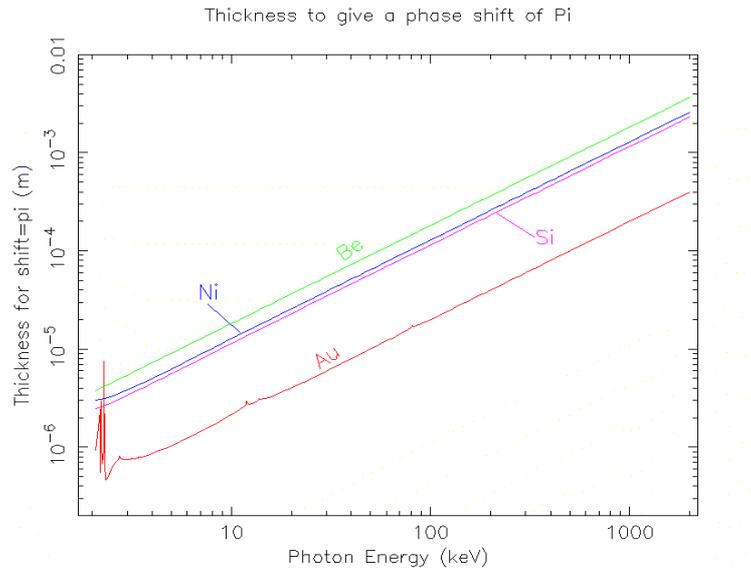


Fig. 1.2: The thickness of different lens materials required to produce a  $\pi$  phase shift [16].

proportional to the photon energy. Since a PFL will have a mean thickness of  $t_\pi$ , the absorption of a  $t_\pi$  layer must be considered as well. Fig. 1.2 and 1.3 show the  $t_\pi$  thickness and amount of absorption in a  $t_\pi$  layer, for 4 materials: beryllium (Be), nickel (Ni), gold (Au), and silicon (Si) [16].

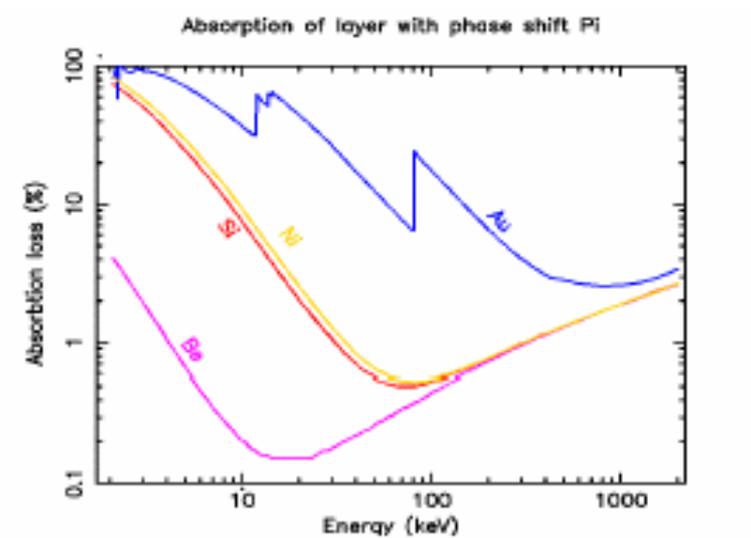


Fig. 1.3: The amount of photon absorption through a material with a phase shift of  $\pi$  [16].

Assuming the tungsten lines of interest at XRFC, the lens material must exhibit good absorption properties at energies between 5keV (soft X-rays) and 100keV (soft gamma-rays). The thickness required for a  $t_\pi$  phase shift must also be in a reasonable range for fabrication. Au is not an ideal lens material for lower energy x-ray tests because of its high absorption. Beryllium (Be) has very low absorption but is not commonly machined. The  $t_\pi$  thickness for Ni will fall in the 10's of micrometers range, but would require multiple (thick) deposition and patterning steps to achieve a high efficiency profile. Fortunately, silicon is cheap, has relatively low absorption properties, and the  $t_\pi$  thickness (10's of micrometers) is within standard fabrication limits. Therefore, silicon is an excellent choice for the lens material.

### **1.3.2. Lens dimensions**

By choosing silicon as the lens material, the photon attenuation length is 80 microns for 8.4keV photons and 1200 microns for 59.3keV photons. Using an arbitrary threshold of ~50%, the lens substrate thickness should be kept smaller than 50 microns, yielding 54% transmission for 8.4keV photons, and 99.6% transmission for 59.3keV photons. For a silicon lens, the thickness required to produce a  $2\pi$  phase shift ( $t_{2\pi}$ ) is quite linear over the range of interest and can be approximated by [16]:

$$t_{2\pi} = 2.55 \left( \frac{\mu m}{keV} \right) * E(keV) \text{ for } (5keV \leq E \leq 60keV). \quad (2)$$

Therefore,  $t_{2\pi} = 21.4\mu m$  for an 8.4keV lens and  $t_{2\pi} = 151.2\mu m$  for a 59.3keV lens.

Due to the inherently long focal length of PFL's at X-ray energies, the outer gratings of a test PFL must be very small to achieve a focal length of 118m required by XRFC. As will be shown later, a ground test lens of appreciable diameter (>1mm) must

then have outer ridge widths on the order of 10's of micrometers, and it should be reiterated that each of these ridges must have a unique sloped or stepped profile in order to achieve a highly efficient PFL.

By choosing silicon as the material for the PFLs described above, we may take advantage of the plethora of technologies available and under development in the area of micro-fabrication, and specifically some fabrication technologies currently being developed in the area of Micro-Electro-Mechanical Systems (MEMS).

#### **1.4. Micro-Electro-Mechanical Systems (MEMS) Fabrication**

Originating from the established fabrication methods of the integrated circuit (IC) industry, the field of MEMS has grown tremendously over the past decade, as planar processes, originally used to make transistors and diodes, were extended to make small mechanical structures and “Microsystems.” Complicated layer structures and established fabrication methods enabled many ‘macro-scale’ components to be mimicked or improved on the ‘micro-scale.’ The possibility emerged for high performance, low power, and low cost systems using similar batch fabrication techniques to those used in the IC industry to fabricate many devices/systems simultaneously.

Many MEMS components use simple physical principles, such as electrostatic actuation or capacitance, to create movement or sensing on a micrometer scale. Some of the most common MEMS applications have come in accelerometers in cars. A significant deceleration (possibly signifying a crash) can be sensed through a change in capacitance from a small proof mass moving on a tiny silicon chip, deploying an airbag if necessary. The field of MEMS has also grown to include areas of research with

applications in biotechnology (a ‘lab-on-a-chip’), optoelectronics (waveguides and switches), power generation (micro-engines), and many more.

Research advances in MEMS fabrication can be applied to an ever increasing range of devices and concepts that wish to take advantage of the batch fabrication, potentially low cost, and high performance of devices made in the micrometer to millimeter scale. The limits are constantly being pushed in both academia and industry as the potential of MEMS is explored even further. The next sections will describe some of the planar fabrication techniques used in MEMS, compare some Fresnel lenses fabricated with such planar techniques, and discuss some of the 3D fabrication techniques currently available and under development in MEMS.

#### **1.4.1. Planar MEMS Fabrication**

Since MEMS fabrication originated from planar IC processes, the basic planar fabrication methods described below are quite well established and as such will not be covered in great detail, nor should this discussion be considered a complete review of all planar techniques used in IC fabrication. Similar fabrication principles can be applied to other material substrates, such as III-V semiconductors, but only silicon is discussed here.

##### **1.4.1.1. Photolithography**

The general method used to define features on a wafer is termed “photolithography.” The enabling material of photolithography, or *lithography*, is a polymeric optically-sensitive material called “photoresist.” Thin photoresist films, usually 1-5  $\mu\text{m}$  thick, are deposited onto a silicon wafer through a spin-coating process.

The photoresist is then baked in a convection oven or on a hot plate at low temperature (110°C) to remove solvent. An optical photomask with clear and opaque patterns is then used to selectively transmit incident UV light, exposing only those areas of photoresist unprotected by an opaque section of the mask. Exposure to UV light will change the chemistry of the photoresist, making it either more or less soluble in a developer solution. If the photoresist is *negative*, areas exposed to UV light become cross-linked, and insoluble in the developer solution, while the unexposed regions remain soluble. Conversely, if the photoresist is *positive*, regions exposed to UV light have their bonds broken and become more soluble than unexposed regions. Upon immersion in a developing solution, soluble regions will be washed away and a spatially selective pattern is formed in the photoresist, exposing certain regions of the wafer to further processing.

Figures 1.4 illustrates the photolithography process for negative and positive photoresists in ‘contact’ lithography, where the optical mask is brought in contact or very close proximity to the photoresist surface. Fig. 1.5 illustrates the principle of ‘projection’ lithography in which the optical mask is located above the photoresist surface and a series of optics are used to transfer the pattern to the wafer in a step-and-repeat method. If reduction optics are used, projection lithography systems may use optical masks that are fabricated typically 4X, 5X, or 10X the dimensions desired on the wafer, making optical masks inexpensive and small linewidths easier to achieve. Projection lithography systems are primarily used in the IC industry, while contact lithography is often used in MEMS fabrication because it can produce most required dimensions and contact systems are less costly and easier to maintain.

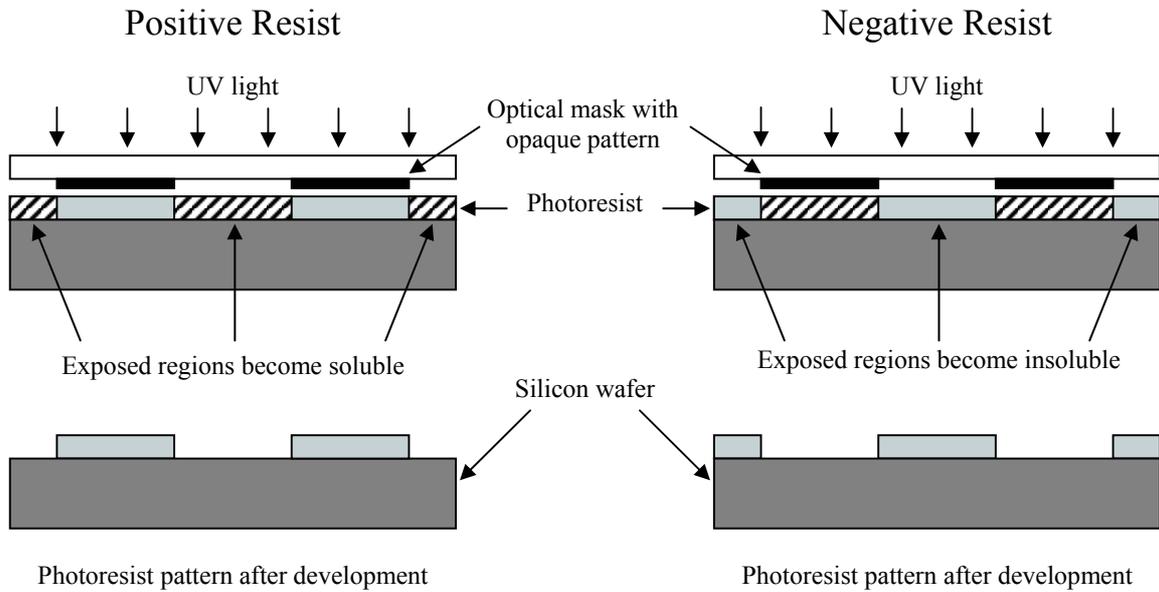


Fig. 1.4: Photoresist patterns resulting from contact lithography using positive or negative resist with the same optical mask.

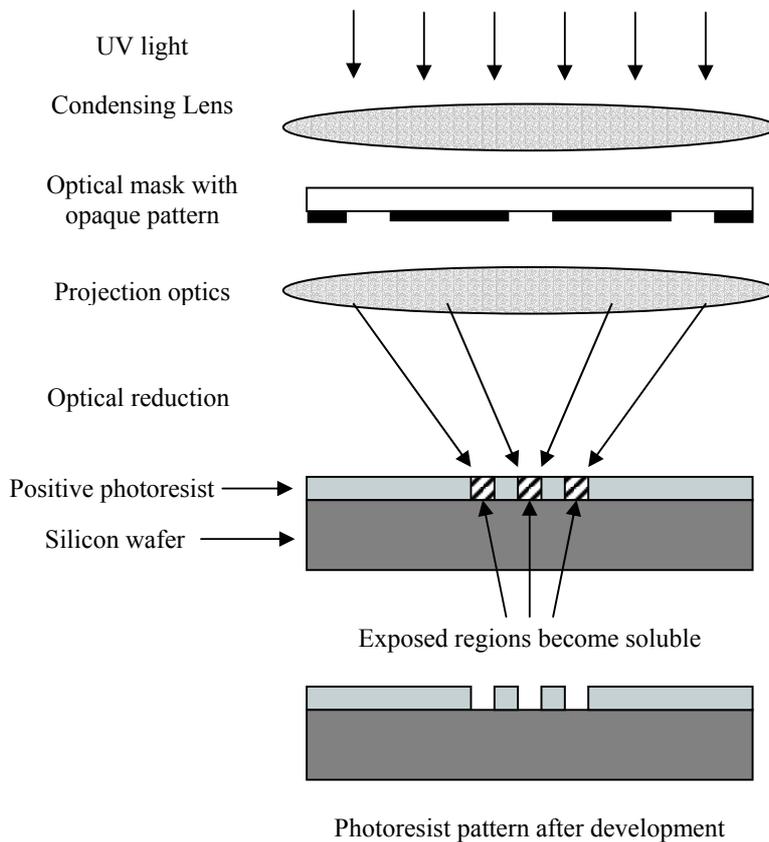
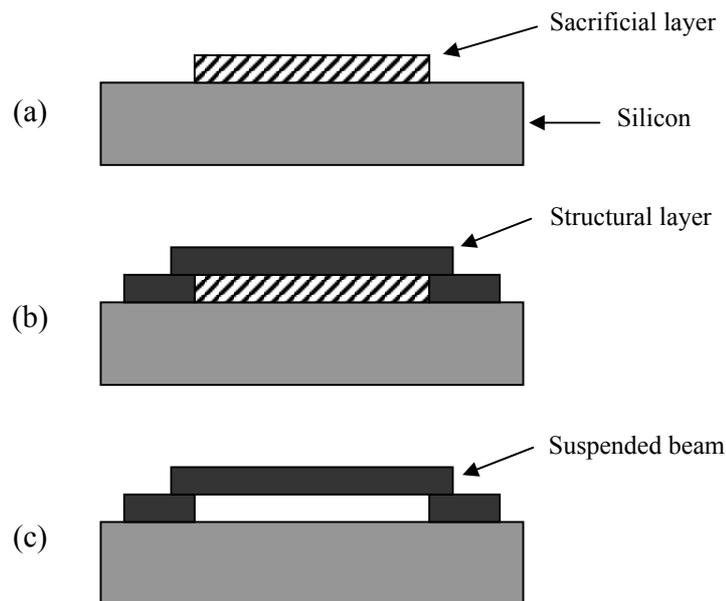


Fig. 1.5: A projection lithography system with reduction optics (usually 5X or 10X).

Photolithography can also be accomplished through the use of an electron-beam, or *e-beam*. By using an electron-beam photoresist material, usually polymethylmethacrylate (PMMA), areas exposed to the *e-beam* depolymerize and become locally soluble, analogous to conventional lithography. However, e-beam lithography has the advantage of being able to create fine resist patterns (on the order of 20nm) since the electron beam/gun can be controlled with extreme accuracy. Unfortunately the process is often slow as each individual pattern on a substrate must be written in a serial manner rather than creating many patterns at once as is done in conventional lithography.

#### 1.4.1.2. Surface Micro-machining

Another key processing technology used in MEMS is surface micro-machining (see Fig. 1.6) [17], where selective etching of a sacrificial layer can produce suspended



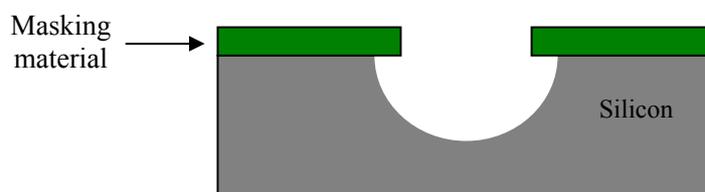
*Fig. 1.6: Surface micromachining to release a beam. (a) Sacrificial layer is deposited and patterned. (b) Structural layer is deposited and patterned. (c) Sacrificial layer is selectively etched away leaving a free-standing beam.*

structures. By first patterning a sacrificial layer, a structural layer may be deposited and patterned on top. After selective removal of the sacrificial layer, a suspended structure remains. With multiple deposition and etching steps, vertical structures may be extruded from the silicon surface, creating stepped profiles with multiple steps. Figure 1.6 shows the steps involved in releasing a suspended beam. However, surface micromachining is a technique used primarily on deposited layers, while our test PFL structures should be made out of a crystalline silicon substrate rather than deposited amorphous layers.

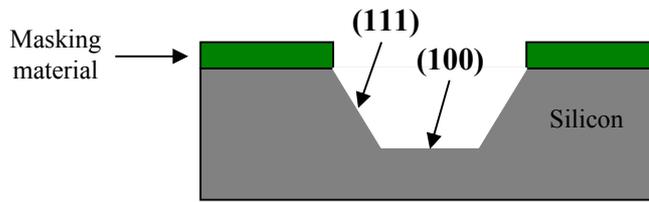
#### 1.4.1.3. Bulk Micro-machining

Bulk micro-machining is essentially the removal of material from the bulk silicon substrate. Such processes can be used in MEMS to define or release structures from the silicon. If silicon could be removed from the substrate appropriately, a crystalline silicon PFL would remain. Bulk micro-machining can be accomplished through a number of different techniques, the most common of which include wet/dry isotropic etching, wet anisotropic etching, and dry plasma etching.

Wet/dry isotropic etching techniques rely on spontaneous chemical reactions to remove the bulk material, where the etch rate is independent of the orientation of the substrate, ideally leaving the masking material untouched. As the etching proceeds, the mask will become “undercut” as some silicon is removed horizontally, beneath the



*Fig. 1.7: Illustration of wet isotropic etching with a small amount of mask undercut.*



*Fig. 1.8: Illustration of wet anisotropic etching, where a 54.7° angle is revealed between the (111) and the (100) planes in crystalline silicon.*

masking layer. See Fig 1.7. For silicon, a common chemistry includes vapor-phase xenon-difluoride ( $\text{XeF}_2$ ), which can dissociate into many forms ( $\text{XeF}^+$ ,  $\text{F}^+$ , etc) and the  $\text{F}^+$  atoms spontaneously react with the silicon to form a volatile ( $\text{SiF}_4$ ). This volatile then desorbs from the silicon surface along with any residual xenon. Unfortunately, isotropic etching is usually used for shallow applications ( $<5\mu\text{m}$ ) and gives little control over the etch profile, making it a poor fabrication technique to adopt for a test PFL in silicon.

When etching single crystal silicon, wet anisotropic etching takes advantage of the orientation-dependent etching characteristics of different chemicals, such as potassium hydroxide (KOH) and tetramethyl ammonium hydroxide (TMAH) [18]. In standard (100) silicon, etching with one of the above solutions will reveal the (111) plane, shown in Fig. 1.8, exhibiting a  $54.7^\circ$  angle with the substrate. For silicon, the etch rate in the (111) plane is typically 100 times smaller than the etch rate in the (100) plane, resulting in very small amount of mask undercut. While this technique can be used to etch deeply into a substrate, only specific angles may be achieved corresponding to the crystal planes in silicon, and is therefore not suitable for a PFL.

Reactive species in glow-discharge plasma can be highly effective etchants, and dry plasma-assisted etching is a technique often used to etch vertically in a silicon substrate, independent of the orientation of that substrate. In reactive ion etching (RIE),

the most common dry etching technique for silicon (Si), a plasma is used to dissociate and ionize a gas, such as carbon-tetrafluoride ( $\text{CF}_4$ ) or sulfur-hexafluoride ( $\text{SF}_6$ ). A wafer is then placed on an anode which is driven with a negative DC bias that accelerates the ions towards the wafer surface.

Once at the wafer surface, there are three main etching mechanisms that take over, shown in Figure 1.9. First (a), physical sputtering occurs when material is removed by the direct bombardment of ions colliding with the surface. This mechanism is very directional for shallow depths ( $<20\mu\text{m}$ ), but comprises only a small part of the overall etch rate. Second (b), ions chemically react with the wafer surface to produce volatile species which are desorbed from the surface. Chemical etching of silicon is responsible

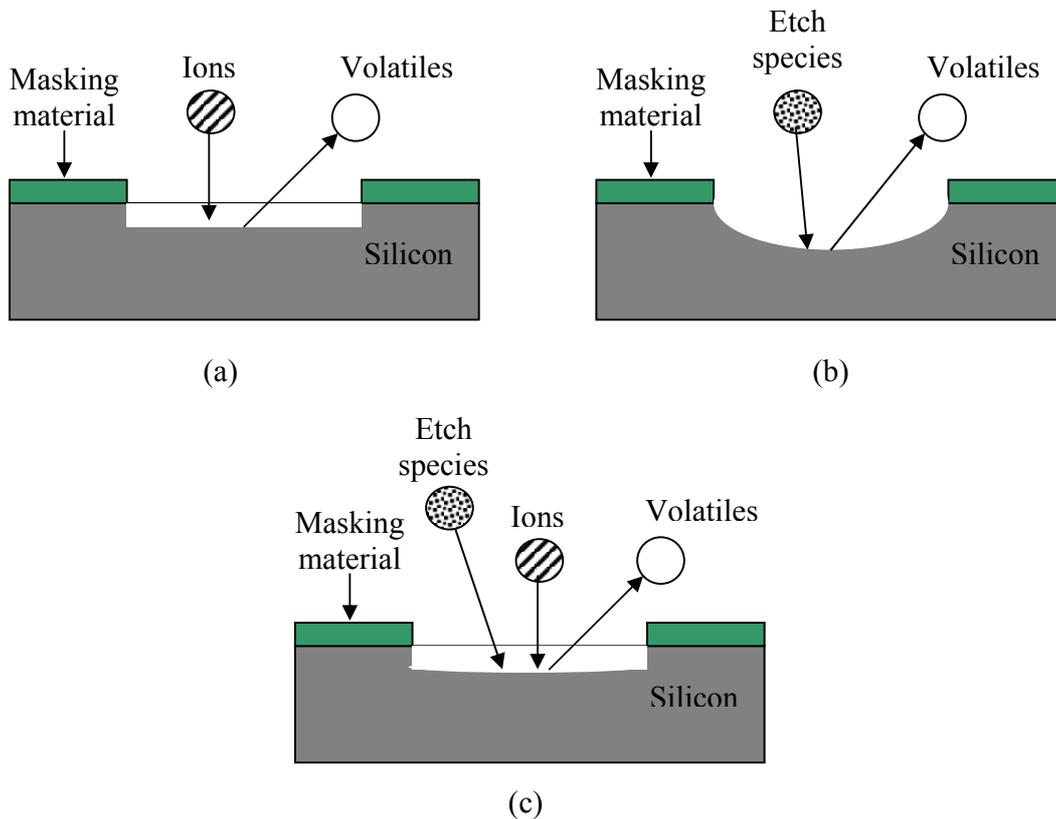


Fig. 1.9: Plasma etching mechanisms: (a) Physical sputtering. (b) Chemical etching. (c) Ion-assisted etching.

for the majority of the etch rate, and is usually achieved by ionizing a gas containing F, such as  $\text{CF}_4$  or  $\text{SF}_6$ . The third etch mechanism in RIE (c), often called ion-assisted etching, a combination of the first two mechanisms whereby ion-bombardment damages the silicon surface in order to make it more chemically reactive. This mechanism can assist in creating vertical etch profiles as the incident ions damage the horizontal surface of the substrate more than the sidewalls, increasing the etch rate vertically but not horizontally.

RIE is widely used to remove silicon vertically from the substrate, and is a potential candidate for removing silicon selectively from a bulk material for a PFL. However, the PFL target depths being considered here (20 to  $150\mu\text{m}$ ) are beyond the standard range of RIE ( $<20\mu\text{m}$ ). A technique called deep reactive ion etching (DRIE) has been developed [19] for applications, such as ours, requiring deeper etches. DRIE uses a very similar process to that of RIE, however etching and passivation steps are cycled in a high density plasma to provide a temporary protection of the silicon sidewall during etching, enabling vertical sidewalls and deep etches (beyond  $400\mu\text{m}$ ). DRIE will be discussed in more detail in Chapter 2 of this thesis.

#### *1.4.1.4. Planar micro-scale Fresnel Lenses*

To this point, X-ray microscopy has been the most common motivation for fabricating Fresnel lenses, yet the dimensions of an astronomical test lens in silicon are quite different than those designed for most X-ray microscopy applications.

*Spector et al.* [20] have fabricated FZP's in both germanium (Ge) and nickel (Ni) for photons in the 500eV range. Using e-beam lithography and electroplating, lenses

with diameters  $<200\mu\text{m}$  were fabricated with outer zone widths as small as 20nm. Since the authors motivation was x-ray microscopy, each FZP emphasized *spatial* resolution rather than *angular* resolution. Not only were these lenses for lower target energies than those being considered in this thesis, but the lens diameters would be too small to collect an appreciable number of photons from a distant x-ray source, such as the one at XRCF.

*Chen et al.* [21] were successful in fabricating a 1.8mm diameter gold (Au) FZP for 8keV photons, much closer to the target energy and diameter of a test PFL in silicon. The Au layer was reported to be  $1.6\mu\text{m}$ , relatively thick for an Au layer in MEMS. Being chiefly concerned with good *spatial* resolution, and short focal length ( $\sim 3\text{m}$ ), outer zone widths were designed to be sub-micrometer. Fabrication was again accomplished using *e-beam* lithography and electroplating. While the authors did achieve a large diameter x-ray lens with sub-micron zone widths, the lens was still *binary* and made of Au, limiting its maximum efficiency and having higher absorption than silicon. Also, to achieve increased angular resolution, the dimensions of an FZP would change dramatically, rendering most of the customized e-beam work done by the authors irrelevant.

*DiFabrizio and Gentili* [22] used a similar e-beam lithography and electroplating approach as the authors mentioned above, however they chose to work towards higher efficiency performances: namely developing a stepped PFL. Lenses of  $150\mu\text{m}$  in diameter were fabricated in both nickel (Ni) and gold (Au) with material layers  $<5\mu\text{m}$  thick. Lenses were designed with 4 levels per zone for energies between 5 and 8keV, again for x-ray microscopy applications concerned with *spatial* resolution. These lenses were able to demonstrate efficiencies greater than the *binary* limit of 40%, however a test

PFL in silicon needs to be much larger in diameter and depth, while exhibiting low absorption, high efficiency, and high *angular* resolution.

#### **1.4.2. 3Dimensional MEMS Fabrication**

It has been established thus far that the proposed test PFL in silicon must have zone widths up to 100 times larger than those created previously and must also be much thicker ( $>20\mu\text{m}$ ) to achieve the appropriate phase shift over a range of energies. Conventional planar MEMS fabrication methods merely extrude shapes from the substrate with deposited layers, or remove material vertically or on crystalline planes. However, the fabrication method used to create the PFL proposed for this research must be somewhat unconventional to achieve this unique combination of size and performance. Therefore, 3Dimensional MEMS fabrication methods must be explored.

Microstereolithography [23] is a method of creating 3D SU-8 (a photosensitive polymer) molds by consecutively depositing and patterning many thin SU-8 layers one at a time. Using many patterning steps, 3D structures can be made, but the process becomes quite time consuming. Also, the approach is best suited to polymers and is not appropriate for silicon applications such as our PFL.

3D metal structures have been achieved with angled sidewalls by using a rotating light source in a customized lithography setup and then electroplating [24]. A rotating light source exposes the desired film at an angle (through two separate masks) and upon development and electroplating, an angled structure may be created. However, this technique requires many customized instruments and again is geared towards fabrication of 3D structures in materials other than silicon.

Other groups have directed ions in DRIE with buried dielectric layers [25] to achieve angled etch sidewalls in silicon. This process requires the bonding of multiple wafers with a charged dielectric between them. Upon etching through the dielectric, charged ions are directed away from the dielectric layer creating angled etching characteristics. However, the authors state that the etch results are not always repeatable, and the bonding of multiple wafers is often avoided due to processing complications.

Ideally, a test PFL in silicon should be fabricated using a method that creates the entire 3D structure at one time, eliminating all alignment steps. This fabrication method must also achieve a wide range of vertical and horizontal dimensions, allowing each zone of a PFL to be individually contoured to maximize lens efficiency.

Recently, an alternative method of fabricating 3D structures, called “Gray-scale technology,” has been investigated as a MEMS fabrication method [26]. 3D structures with stepped profiles and horizontal dimensions ranging from 5 $\mu$ m to 20mm are created in a single lithography step, yielding a controllable profile in photoresist. This photoresist pattern may then be used as a nested mask in a dry-anisotropic etching process, such as RIE or DRIE, to selectively transfer the 3D photoresist profile into a 3D silicon structure. Since our target depths are beyond the normal limits of standard RIE, DRIE can be used as the transfer technique, where etch depths greater than >20 $\mu$ m can be easily achieved.

Gray-scale technology has thus been chosen as the fabrication technique to be used in this research to develop a high efficiency, high angular resolution PFL in silicon. Gray-scale technology will be presented in detail in Chapter 2 of this thesis, while Chapter 3 will discuss design considerations for gray-scale optical masks and introduce a

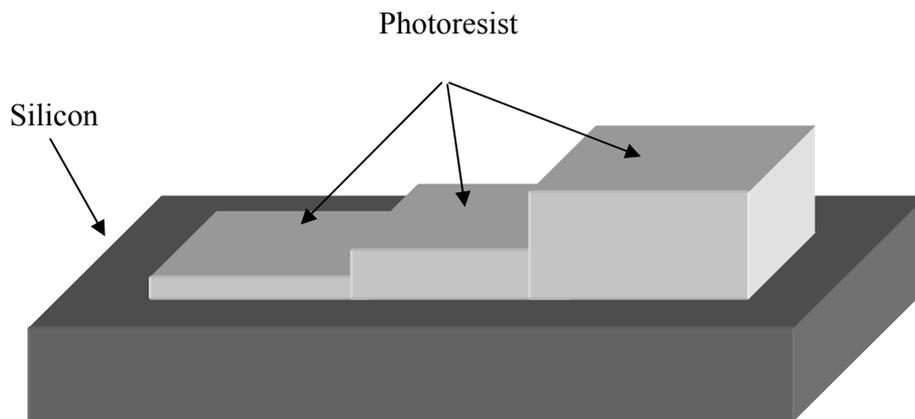
novel method for predicting the 3D photoresist profiles of various PFLs. Chapter 4 will summarize the lithography and etching results achieved while fabricating test PFLs, and evaluate the resulting silicon profiles. Concluding remarks about the use of gray-scale technology for fabricating test PFLs in silicon will be provided in Chapter 5, and some future work on silicon PFLs and gray-scale technology will be discussed.

## 2. Gray-scale Technology

### 2.1. Introduction

Previously, gray-scale lithography has been widely utilized in the area of diffractive optics [27-29]. However, gray-scale technology using dry-anisotropic etching has recently emerged as an enabling fabrication method for creating 3D structures for MEMS applications [30,31]. For clarity, the term ‘gray-scale *lithography*’ will refer to a photolithography process using a ‘gray-scale optical mask,’ while the term ‘gray-scale *technology*’ will refer to the combination of ‘gray-scale *lithography*’ and a dry-anisotropic etching step to transfer the photoresist pattern into the silicon.

Gray-scale mask design and lithography can be approached in a number of ways, however all methods rely on the same general principles [26,32-35]. A gray-scale optical mask is used to transmit only a portion of the incident intensity of light, partially exposing sections of a positive photoresist to a certain depth. This exposure renders the top portion of the photoresist layer more soluble in a developer solution, while the bottom portion of the photoresist layer remains unchanged. Therefore, after a standard development step, some thickness of photoresist, called a ‘*gray level*,’ will remain behind in areas that received a partial exposure. By locally modulating this intensity pattern with a specially designed gray-scale optical mask, many gray levels may be created at once, forming a 3D structure in the photoresist (see Fig. 2.1). By patterning the photoresist on a silicon wafer, a dry-anisotropic etching technique, such as reactive ion etching (RIE) or deep reactive ion etching (DRIE), may be used to subsequently transfer this pattern into the silicon.



*Fig. 2.1: Photoresist gray levels after development, on a silicon substrate.*

This chapter will provide a basic understanding of the two main steps involved in creating a 3D silicon profile using gray scale technology: (1) gray-scale lithography, and (2) photoresist pattern transfer using DRIE. The effects of critical design and fabrication parameters regarding the fabrication of a PFL in silicon will be discussed, and experiments will be presented that investigate the effects of DRIE process parameters on etch selectivity.

## **2.2. Gray-scale Lithography**

The first step in creating a PFL in silicon is the formation of a 3D photoresist nested mask using gray-scale lithography. Gray-scale lithography is accomplished through the use of specialized optical mask designs and tight process controls, thus realizing an entire 3D photoresist structure in only one lithography step. The following sections will describe gray-scale optical mask design and the lithography processing involved in creating 3D photoresist structures composed of *gray levels*. The design of a standard calibration mask will be outlined and a standard lithography process defined as the lithography process used throughout the silicon PFL development.

### 2.2.1. Gray-scale mask design

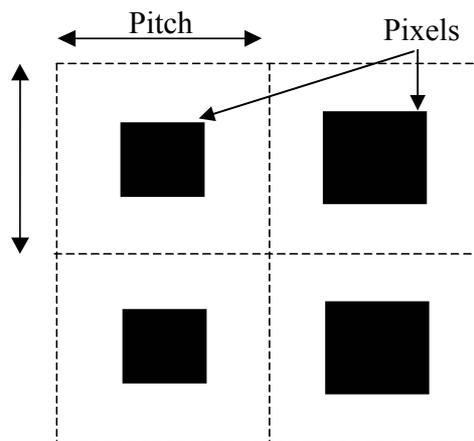
The chosen method of gray-scale patterning, as detailed by *Waits et al.* [26,32] uses various sized opaque features, called ‘*pixels*’ on a transparent background. These pixels block a fraction of the incident light and cause diffraction as the light passes between them. The light transmitted between pixels can be modeled as the transmission of a square wave function. By expanding this square wave into a Fourier series [36], it can be seen that the 0<sup>th</sup> diffraction order acts as a ‘DC’ component to the transmitted light, while all higher diffraction orders are spatial frequency terms that depend on the spatial dimensions of the features on the mask. By recalling the elements of a projection lithography system (Fig. 1.6), the objective lens serves to collect the diffracted light from the optical mask and transmit it through the system to the wafer. As the spatial dimensions of opaque features on the optical mask are decreased, the higher diffraction orders become spread out in the plane of the objective lens. Once the pixels and their spacing are below the resolution of the chosen projection lithography system, only the 0<sup>th</sup> diffraction order is collected by the objective lens, effectively truncating the Fourier series leaving only the ‘DC’ component to the transmitted light [26]. The resolution of the stepper system may be approximated as [37]:

$$Resolution = \frac{\lambda}{(1 + \sigma) NA} \quad (2.1)$$

where  $\sigma$  is the partial coherence of the light source, NA is the numerical aperture, and  $\lambda$  is the wavelength of light being used. By varying the size of the sub-resolution pixels, this ‘DC’ component of the transmission can be locally modulated, resulting in the exposure of a photoresist film to various depths.

Using a projection lithography system with 5X reduction and an estimated resolution of 0.5-0.8 $\mu\text{m}$  on the wafer, the resolution limit on our gray-scale optical mask should be on the order of 2.5-4.0 $\mu\text{m}$ , meaning each pixel must be at least that small. Chrome on quartz optical mask vendors may easily be found to produce masks with minimum feature sizes of 0.5 $\mu\text{m}$  on the optical mask and dimensional accuracy in 0.1 $\mu\text{m}$  steps above that minimum size [38], which should be sufficient to create our PFL structures. Therefore, gray levels may be designed using pixels of 0.5 – 4.0 $\mu\text{m}$ .

Multiple pixel layout schemes may be adapted for use in gray-scale lithography. Some methods use a constant pixel size while changing the pixel spacing, and others change both pixel spacing and pixel size. However the programming of these two methods becomes increasingly complicated. Therefore, the method used by *Waits et al.*, and also for this research, employs a constant sub-resolution pixel spacing, or pitch, and varies only the size of the opaque pixels. See Fig 2.2. This method allows the center of each pixel to be easily located by X-Y coordinates, essentially defining the transmission



*Fig. 2.2: Optical mask design scheme using sub-resolution pixels and a constant sub-resolution pitch. By calculating the area of each pixel and the area of the pitch, the percent transmission through the mask is estimated.*

at that point on the mask. Since each pixel is sub-resolution, the actual shape of the pixel should not be reconstructed and therefore only the total area of the pixel will be important. For this research, rectangular pixels were used to make mask design and fabrication as simple as possible.

It is then useful to calculate the relative amount of light blocked by the opaque pixel, or equivalently the percentage of light transmitted through the optical mask. By calculating the area of each pixel ( $A_{pixel}$ ) and the area of the square pitch ( $A_{pitch}$ ), we can estimate the percent transmission through the optical mask ( $Tr$ ) as:

$$Tr = 1 - \frac{A_{pixel}}{A_{pitch}} . \quad (2.2)$$

Mask fabrication provides some constraints on pixel dimensions and their possible increments, so the choice of a mask pitch immediately dictates the upper and lower bounds of  $Tr$ . The number of different size rectangles that may be created between these limits will be finite. The result is a discrete  $Tr$  set that depends on the selected pitch and the mask vendor limitations, with each  $Tr$  creating a distinct gray level in photoresist.

### **2.2.2. Lithography Processing**

Once mask design and fabrication is complete, a gray-scale lithography process must be created. Traditionally, exposed photoresist is either totally cleared or entirely left behind, depending on its polarity (positive or negative). The goal of gray-scale lithography is to only clear away a fraction of the photoresist thickness, leaving behind a series of *gray levels* that together form a 3D photoresist structure. The partial light transmission through the specially designed gray-scale optical mask coupled with time of exposure, time of development, and photoresist contrast, will determine these final

intermediate *gray level* heights in photoresist. The lithography processing for this research was performed by a collaborator on this research, C. Michael Waits, on a 5X projection lithography system (CGA-Ultratech) in the clean-room facilities at the Laboratory for Physical Sciences (LPS), in College Park, MD. Further details on the gray-scale lithography process may be found in [26].

#### 2.2.2.1. *Photoresist Selection*

Gray-scale lithography processing begins with the spinning of a thin photoresist film onto a silicon wafer. In the case of this research, 75mm <100> test grade silicon wafers were used for compatibility with the projection lithography system at LPS. However, photoresists used in gray-scale lithography may have some properties that differ from high-resolution photoresists used in other applications. Typically, gray-scale lithography uses photoresists that exhibit low ‘*contrast*,’ where contrast is essentially a measure of how sensitive the photoresist is to changes in exposure dose.

Ideally, as the exposure dose is increased on a photoresist with the highest possible contrast, there will be one point at which the photoresist changes from unexposed to totally exposed, usually referred to as the ‘clearing dose.’ However, no photoresist exhibits this ideal clearing dose behavior, every photoresist has a small exposure dose ‘range’ in which the photoresist will be partially removed during development. The *contrast* of a photoresist is defined as the slope of the development rate vs. log(exposure dose) curve, and essentially measures the size of this ‘range.’ Photoresists with low contrast should realize more *gray levels* because a large portion of the available *Tr* values from the mask design will fit inside the operating range, enabling

a larger number of achievable *gray levels*. Photoresists with high contrast behave closer to the ideal case, making the operating range for gray-scale lithography quite small.

Additionally, thicker photoresists are better suited to gray-scale lithography because the *gray levels* may be more pronounced and a larger margin of error afforded to the process. Clariant's AZ9245 was chosen as the photoresist for this research because it has relatively low contrast and can be spun to a nominal thickness of  $>6\mu\text{m}$  with ease. The spin coating procedure is followed by a 90 sec soft bake step at  $110^{\circ}\text{C}$  to remove solvent. No hard bake step is required according to the manufacturer.

#### 2.2.2.2. *Exposure*

To find the optimum exposure dose for gray-scale lithography, a *focus-exposure* matrix must be performed, as is done in conventional lithography. In a focus exposure matrix, the exposure dose is varied along the rows and the focus is varied along the columns of the dies on a wafer, giving many combinations of focus and exposure. By inspection, the die with the best results will then determine the optimum settings for both the focus and exposure. However, the evaluation of the exposure results after a fixed development step is different if your purposes are gray-scale in nature. In gray-scale lithography, it is desirable to find the exposure dose where the fully exposed photoresist is barely removed; ensuring the largest possible range for *gray levels*. To evaluate this optimum exposure point, visual inspection determines when the photoresist is fully cleared away in open areas. The number of *gray levels* remaining on a calibration structure containing a wide range of *gray levels* can then be counted. (The design and properties of such a calibration optical mask will be discussed in section 2.2.3.) The die

with the largest number of gray levels remaining after development is interpreted to have the exposure settings closest to ideal for gray-scale lithography processing.

### 2.2.2.3. *Development*

The one element that has been taken as a given until this point has been the development step. During photoresist development, even areas of unexposed photoresist will be removed, albeit extremely slowly. Depending on the time of development, areas receiving any exposure dose will eventually develop fully. By inspection an appropriate development range can be established, but in this range there are multiple development recipes that yield many *gray levels*. Depending on previous processing steps, different development recipes may appear ideal, but for this research, the development step is held constant, and gray-scale structures are optimized primarily through optical mask design.

During development, the concentration of developer solution and the time necessary for development are inversely proportional. The developer solution used for this research was Clariant's AZ400K, mixed in a concentration of 3:1, DI water to developer solution. This yielded development times in the 3-5 minute range, much longer than conventional development times of 1-2 minutes. Using a stronger concentration of developer solution may decrease the development time, but the faster development rate will make it more difficult to stop the development at the appropriate time. If over-development occurs, there will be a loss of lower gray levels, and all remaining *gray levels* will be lower than expected. The converse is also true, where under-development will cause gray levels to be higher than expected and open areas may not be fully developed. Fig. 2.3 shows three gray-scale stepped wedge structures. Fig.

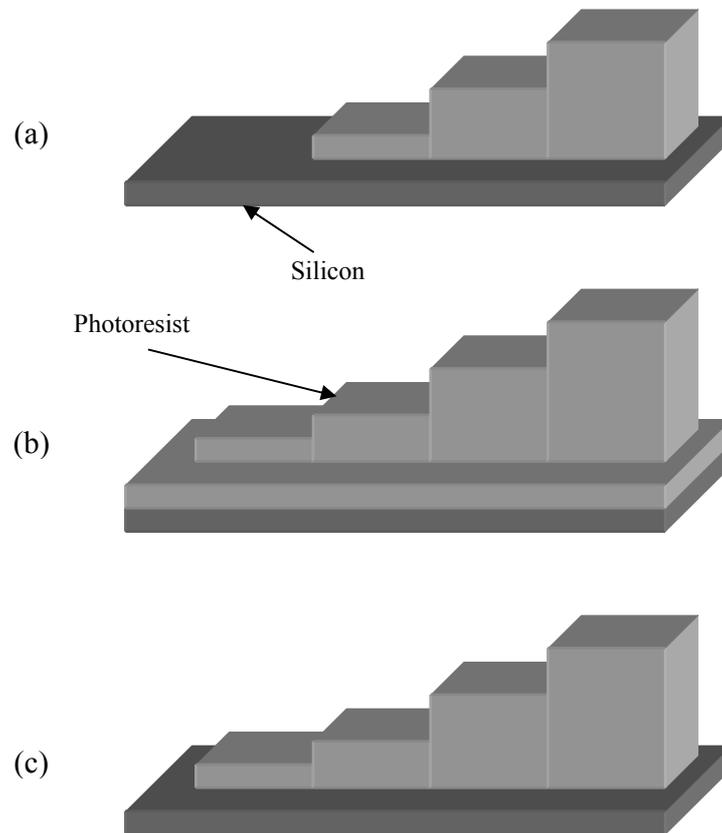


Fig. 2.3: A photoresist gray-scale wedge structure after (a) over-development, (b) under-development, and (c) appropriate development.

2.3(a) shows a structure which has been over-developed and has lost the lowest of 4 *gray levels*, while the remaining 3 *gray levels* are lower than expected. Fig. 2.3 (b) shows a structure which has been under-developed where all *gray levels* are higher than expected and the open areas were not fully cleared away. Fig. 2.3 (c) shows a structure developed for the appropriate amount of time. When developing a precise silicon PFL, all *gray levels* must be consistently achieved at the same height to ensure profile accuracy, enabling high lens performance. Therefore, a longer development time must be tolerated for high quality gray-scale structures like the PFL being developed in this research.

### 2.2.3. Calibration mask

A generic calibration mask was created to experimentally investigate the resolution of the projection lithography system being used. Gray-scale wedge structures (like those shown in Fig. 2.2) were designed with various pitches ranging from  $1.5\mu\text{m}$  to  $4.0\mu\text{m}$  in  $0.1\mu\text{m}$  increments to cover the estimated resolution range of  $2.5\text{-}4.0\mu\text{m}$  pitch on the optical mask. Using only square pixels, *gray levels* of fixed length ( $50\text{-}100\mu\text{m}$ ) were designed adjacent to each other. By using a combination of pitches and pixels, an achievable range of pixels can be established for a wide range of pitches simultaneously. An opaque area on the mask was included adjacent to the highest *gray level* as a reference point, so the heights of *gray levels* after fabrication can be measured and correlated to the pixel size that created them. Using scanning electron micrographs (SEM) and optical microscopes, *gray levels* in photoresist were inspected to obtain the resolution of the projection lithography system. See Fig. 2.4. *Gray levels* created by over-resolution pitches appear rough as the tiny pixels are partially resolved in the photoresist, see Fig. 2.4(a). *Gray levels* created by using sub-resolution pitches will

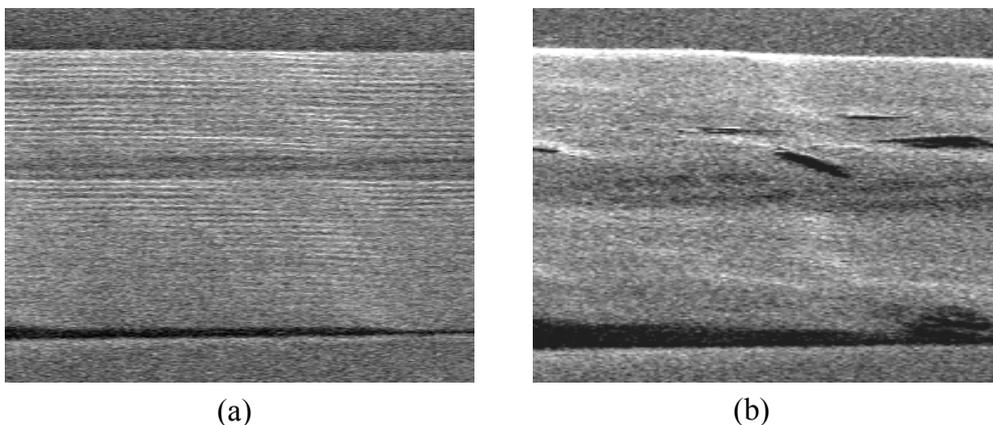


Fig. 2.4: SEM images showing two gray levels produced using two different pitches on the gray-scale mask; (a) shows a pitch slightly above resolution and (b) shows a smoother photoresist surface using a pitch right at the resolution limit.

appear smooth, see Fig. 2.4(b). The resolution of the system is defined here to be the largest mask pitch that still creates a smooth *gray level*. The pitch determined as the resolution limit for our 5X reduction projection lithography system was 2.8 $\mu\text{m}$ , meaning a horizontal resolution of  $\sim 0.56\mu\text{m}$  could be expected on the wafer for a photoresist PFL.

One major constraint that arises from the use of a 5X reduction system pertains to the exposure area. When using a traditional 100mm optical mask plate, after 5X reduction the exposed area will be only 20mm by 20mm, limiting the area that can be patterned at one time. While this exposure area should be sufficient for most ground test lenses, this limitation may prohibit this technology from being extended to space-bound applications with much larger single lens dimensions.

The advantage of using a calibration mask to experimentally obtain a lithography system's gray-scale properties is that the many factors that determine the success of gray-scale lithography can be simultaneously investigated by looking only at the final gray-scale photoresist structures. Table 2.1 describes the wedge structures included on the calibration mask designed for this research and described above.

Structure	Pitch	Largest Pixel ( $\mu\text{m}$ )	# of levels	Level Length ( $\mu\text{m}$ )
D1	1.2	0.7	4	100
D2	1.3	0.8	5	50
D3	1.4	0.9	6	50
D4	1.5	1.0	7	50
D5	1.6	1.1	8	100
D6	1.7	1.2	9	50
D7	1.8	1.3	10	50
D8	1.9	1.4	11	50
D9	2.0	1.5	12	100
D10	2.1	1.6	13	50
D11	2.2	1.7	14	50
D12	2.3	1.8	15	50
D13	2.4	1.9	16	100
D14	2.4	1.9	16	100
D15	2.5	2.0	17	50

D16	2.6	2.1	18	50
D17	2.7	2.2	19	50
<b>D18</b>	<b>2.8</b>	<b>2.3</b>	<b>20</b>	<b>100</b>
D19	2.9	2.4	21	50
D20	3.0	2.5	22	50
D21	3.1	2.6	23	50
D22	3.2	2.7	24	100
D23	3.3	2.8	25	50
D24	3.4	2.9	26	50
D25	3.5	3.0	27	50
D26	3.6	3.1	28	100
D27	3.7	3.2	29	50
D28	3.8	3.3	30	50
D29	3.9	3.4	31	50
D30	4.0	3.5	32	100

Table 2.1: List of gray-scale wedge structures designed on the calibration mask. All structures are 100 $\mu\text{m}$  wide and have a smallest pixel size of 0.5 $\mu\text{m}$  on the optical mask. The determined resolution of the lithography system is shown in bold.

#### 2.2.4. Standardized lithography process

Creating a standard lithography process enables the lithography processing to be considered a constant, allowing photoresist structures to be designed using primarily pixel selection. The standard lithography process used for this research was developed using the calibration mask described in section 2.2.3, and the 5X projection lithography system at LPS with an observed resolution of 0.56 $\mu\text{m}$ . The standard process is detailed below in Table 2.2. A hard bake step is omitted from this process as suggested by the photoresist manufacturer, and to avoid any potential photoresist re-flow during hard bake.

Photoresist	Spin Coating (rpm : sec)	Soft Bake ( $^{\circ}\text{C}$ : sec)	Focus ( $\mu\text{m}$ )	Exposure (mJ)	Developer : Time (min)
AZ9245	2500 : 60 (6.0 $\mu\text{m}$ thick)	110 : 120	-1.0	250	AZ 400K 1:3 DI water : 5

Table 2.2: The standard gray-scale lithography process.

## 2.3. Dry-anisotropic Etching

Once a photoresist gray-scale structure has been created, the pattern must be transferred into the silicon via a selective etching process. As mentioned in section 1.4.1.3, deep reactive ion etching (DRIE) will be the preferred method of pattern transfer for creating a silicon PFL to achieve the desired dimension range of 20 to 150  $\mu\text{m}$ . The DRIE system used in this research was a Unaxis, PlasmaTherm 770 Inductively Coupled Plasma (ICP) etching system located in the clean-room at the Army Research Laboratory (ARL) in Adelphi, MD.

### 2.3.1. Deep Reactive Ion Etching (DRIE)

Robert Bosch GmbH established the basic DRIE process in 1996 [19], while *Whitley et al.* [39] briefly demonstrated and received a patent on the transfer of gray-scale structures into silicon using DRIE in 2002. A basic DRIE process is shown in Fig. 2.5. First, a masking material (often photoresist) is patterned; exposing only selected areas for etching, see Fig. 2.5(a). The etching step is executed first using an inductively coupled plasma (ICP) containing  $\text{SF}_6$  and Ar gases, which enables relatively vertical sidewalls over small depths (usually  $\sim 1\mu\text{m}$ ), however there will be a limited amount of isotropic lateral etching of the silicon, see Fig. 2.5(b).  $\text{C}_4\text{F}_8$  gas is then cycled into the chamber as part of a passivation step, creating a conformal teflon-like film over the entire surface, see Fig. 2.5(c). When the etching step is repeated, the passivation layer is preferentially removed from horizontal surfaces by the  $\text{Ar}^+$  ions in the plasma allowing the vertical etching to continue, while the previously etched silicon sidewall is temporarily protected from continued etching by F ions, see Fig. 2.5(d). Passivation, Fig. 2.5(e) and etching,

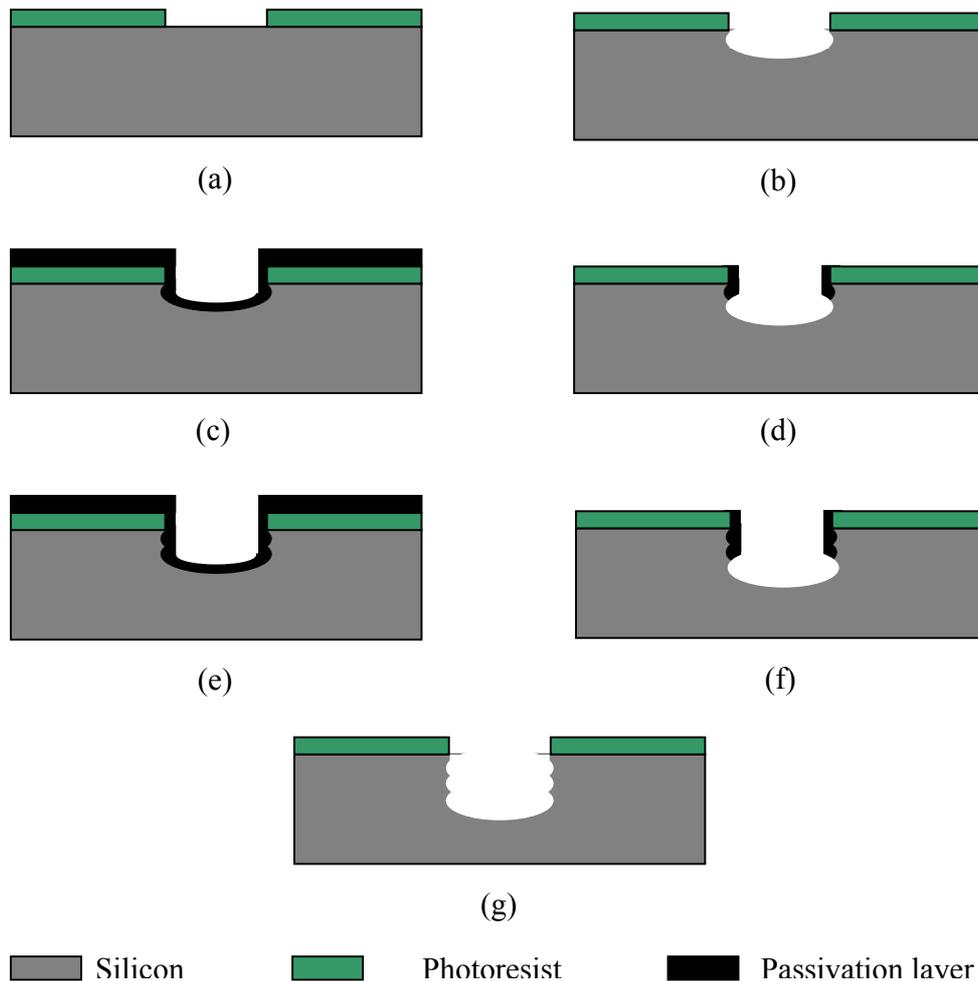
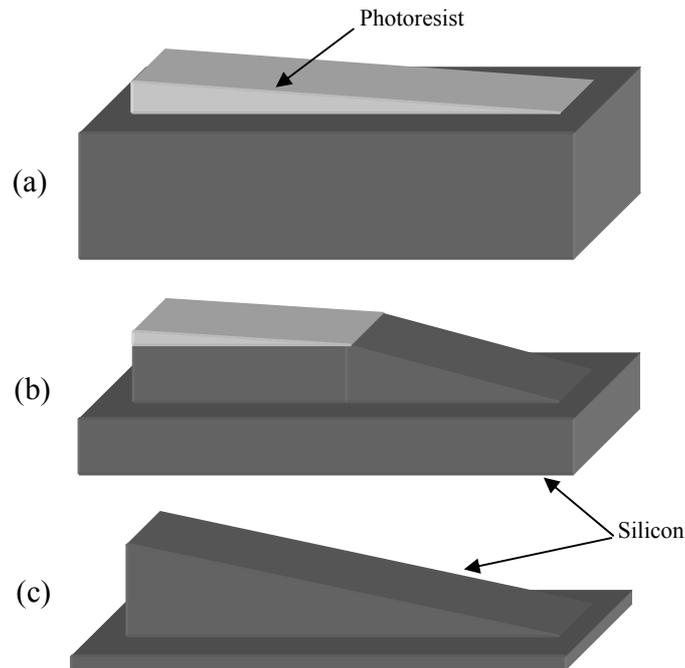


Fig. 2.5: The steps of deep reactive ion etching (DRIE). (a) Silicon wafer is patterned with a masking material. (b) Etching step of the cycle etches the silicon similar to RIE. (c) Conformal passivation layer is deposited over entire wafer. (d) Etch step is repeated. Ion bombardment removes the passivation layer from horizontal surfaces, while sidewalls remain protected. (e) Passivation step is repeated to cover the newly exposed sidewall. (f) Etch step repeats. (g) Vertical trench achieved with slight sidewall scalloping.

Fig. 2.5(f), steps are cycled until a desired etch depth is achieved in the silicon, resulting in a deep vertical etch with slight scalloping on the sidewalls, Fig 2.5(g). Typical recipes etch 0.5-1.0 $\mu\text{m}$  per cycle.

### 2.3.2. Gray-scale Pattern Transfer

During the DRIE process, the masking material will be simultaneously etched along with the substrate. However, the etch rate of the masking material, in our case



*Fig. 2.6: (a) Initial sloped photoresist structure on silicon. (b) Sloped pattern begins to transfer into the silicon with a certain selectivity. (c) Final structure in silicon retains lateral dimensions while vertical dimensions are amplified by the etch selectivity.*

photoresist, is many times lower than the etch rate of silicon. This ratio of the silicon to photoresist etch rates is referred to as the ‘etch selectivity.’ Etch selectivity for a photoresist mask is typically around 80 to 1, usually written as 80:1 or just 80.

In gray-scale technology, the etch selectivity becomes an important parameter to control because the difference in the etch rates of the two materials causes an amplification of all vertical dimensions. Fig. 2.6 shows a small photoresist wedge on a silicon substrate. As this wedge is etched in a DRIE process, any exposed silicon will etch quickly, while the photoresist nested mask etches more slowly (the photoresist is primarily etched by ion bombardment). The transferred gray-scale structure retains its horizontal dimensions, however the vertical dimensions are amplified by the etch selectivity. Therefore, selectivity control is absolutely necessary for the fabrication of precise 3D structures like a PFL in silicon.

### 2.3.3. Selectivity control experiments

To investigate the etch properties of the DRIE system being utilized at ARL, a series of experiments were used to gauge the effects of individual parameter changes in the different DRIE process steps [31]. Samples were prepared with opaque and gray-scale photoresist test structures on 75mm <100> test-grade silicon wafers. Each sample wafer was then etched for 150 cycles, while individual etch parameters were varied during each run. The etch recipe used as the starting point in this characterization, ‘Base Etch I,’ is shown in Table 2.3.

Step	Ar (sccm)	C <sub>4</sub> F <sub>8</sub> (sccm)	SF <sub>6</sub> (sccm)	Pressure (mTorr)	Water Temp(°C)	Electrode Power (W)	Time (sec)
Passivation	40	70	0	19	20	1	5
Etch	40	0	100	19	20	10	8

Table 2.3: DRIE process parameters for Base Etch I.

Three parameters were initially chosen to investigate the characteristics of our DRIE tool: global silicon loading, electrode power, and chamber temperature (as determined by the temperature of the water flowing through the chamber walls). Each successive etch differed from Base Etch I in only one respect so that the effects of each parameter could be isolated. The silicon loading of Base Etch I was 41%, while Base Etch II had a silicon loading of 27% and Base Etch III had a silicon loading of 52%. The high electrode power etch was done with a power of 12W during the etch step, while the low electrode power etch had a power of 8W. The increased chamber temperature etch was done with the water temperature set to 50°C instead of 20°C.

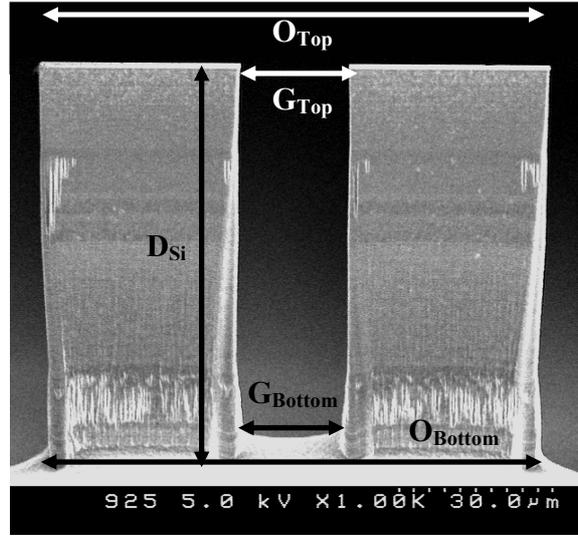


Fig. 2.7: SEM showing the structures used to measure the inside sidewall angle and outside sidewall angle.

Height measurements of gray-scale structures at 5 points on each wafer were taken in photoresist (before etching) and in silicon (after etching) to obtain the selectivity resulting from the particular etch recipes:

$$Selectivity = \frac{H_{Silicon}}{H_{Photoresist}}. \quad (2.3)$$

Sidewall angles were calculated using two opaque structures on the optical mask separated by  $30\mu\text{m}$ , shown in Fig. 2.7. Measurements were taken of the gap between the two structures at the top and the bottom of the etch, and also of the outer dimensions of the two structures at the top and bottom of the etch. The sidewall angles were then calculated according to:

$$\text{Gap Angle} = \tan^{-1} \left[ \left( \frac{G_{Top} - G_{Bottom}}{2} \right) / D_{Si} \right] \quad (2.4)$$

and

$$\text{Outside Angle} = \tan^{-1} \left[ \left( \frac{O_{Bottom} - O_{Top}}{2} \right) / D_{Si} \right]. \quad (2.5)$$

By calculating the angle in this way, re-entrant profiles will result in negative sidewall angles and positive profiles will result in positive sidewall angles.

The etch results from each of the 6 etches completed for this preliminary investigation are outlined in Table 2.4. The selectivity achieved for Base Etch I (75) is considered in the normal range of operation for a DRIE system.

Parameter	Silicon Loading (%)	Selectivity	Si Etch Rate ( $\mu\text{m}/\text{min}$ )	PR Etch Rate ( $\mu\text{m}/\text{min}$ )	Sidewall Angle (Inside/Outside)
<b>Base Etch I</b>	<b>41</b>	<b>75</b>	<b>0.71</b>	<b>0.010</b>	<b>-0.1°/-2.0°</b>
Base Etch II	57	42	0.73	0.016	-0.1°/-2.0°
Base Etch III	27	92	0.90	0.010	-0.3°/-2.1°
Higher Electrode Power	41	55	0.71	0.013	-0.5°/-2.2°
Lower Electrode Power	41	103	0.54	0.005	2.3°/0.4°
Increased Temperature	41	48	0.75	0.015	-1.3°/-2.9°

*Table 2.4: DRIE selectivity characterization results.*

When the silicon loading was decreased to 27%, allowing the same amount of F ions attacked a smaller amount of exposed silicon. This increased the silicon etch rate while the photoresist etch rate remained relatively constant, resulting in an increase in selectivity to 92. Conversely, when the silicon loading was increased to 57%, the etch selectivity was significantly reduced to 42. With an increase in electrode power, the photoresist etching rate increased because the energetic ions sputtered off the photoresist at a faster rate than in Base Etch I, resulting in a selectivity of 55. When the electrode power was decreased, the silicon and photoresist etch rates decreased because each incident ion was less energetic. However the photoresist etch rate (primarily due to ion bombardment) was decreased proportionately more, resulting in a selectivity of 103 for the low electrode power case. By increasing the water temperature in the chamber, the surface of the wafer was effectively heated up. Both the photoresist and silicon etch rates

increased, however the photoresist etch rate grew proportionately more resulting in a net lowering of the selectivity to 48.

The sidewall profiles for the final 3 etches were considerably different from the three Base etches, which were all fairly vertical. It should be noted that the outside angle is consistently more re-entrant than the inside angle because the outside walls of the structures being considered are subject to greater bombardment from ions with high incident angles, whereas the inside walls are protected from highly angular incident ions by the structure itself. In the high electrode power case, the increased energy of each incident ion caused faster removal of the sidewall passivation layer, allowing slight horizontal etching during each cycle, which in turn results in a more re-entrant profile when compared to Base Etch I. Conversely, when the electrode power was lowered, the sidewall passivation is not completely removed during each cycle causing a gradual build-up of the passivation layer, resulting in a slightly positive profile (again being more positive in the gap due to the decreased angular ion bombardment). When the chamber temperature was increased, the rate of reactions involving the sidewall passivation layer increased. This increased temperature caused the sidewall passivation layer to be removed prematurely, resulting in a highly re-entrant sidewall profile.

This DRIE characterization enables recipes to be developed for tuning etch selectivity to the desired values for etching a PFL in silicon. The effects of changing individual parameters may be anticipated as they pertain to both etch selectivity and sidewall profile. By combining multiple parameter changes and adjusting the passivation layer thickness (length of passivation step), highly vertical etches with precise etch selectivity are possible over a wide range of etch selectivities.

## **2.4. Summary**

The goal of this chapter was to introduce in detail gray-scale lithography and 3D photoresist pattern transfer using DRIE. The terms and concepts presented herein will be used freely throughout chapters 3, 4, and 5 of this thesis as the specific design and fabrication steps taken to achieve a PFL in silicon are described. Limitations imposed by the chosen method of gray-scale patterning, such as horizontal resolution and maximum exposure area, have been discussed as they pertain to the creation of a PFL in silicon, and a preliminary DRIE characterization completed towards the creation of vertical etch profiles with precisely controlled etch selectivity.

## **3. Optical Mask Design**

### **3.1. Introduction**

The proper design of a gray-scale optical mask is the single most important step when creating a 3D structure. No matter how well the lithography and etching steps are controlled, if the initial intensity pattern is incorrect, all subsequent processing is pointless. It has been established that the chosen pixilation scheme will use a set of sub-resolution pixels spaced on a sub-resolution pitch (section 2.2.1), however the selection of exactly which pixels and which pitch to use depends on many factors when developing a PFL in silicon. There will be some freedom afforded to the designer at certain stages, however tradeoffs will become apparent as some selections will necessitate others.

This chapter will outline the PFL profile equations to precisely define the profiles of interest and comment on the specific challenges posed by such a profile. A number of conditions will be derived regarding the precise design of pixilated gray-scale optical masks, and their impact on a potential PFL profile. An experimental approximation method, tied closely to the calibration mask used in section 2.2.3, will also be introduced as a means of predicting and defining an arbitrary profile in a photoresist film. Finally, the optical mask designs for multiple lens iterations will be discussed.

### **3.2. PFL equations**

To create a PFL in silicon, first we must define and assess the exact profile that we wish to emulate. As mentioned in section 1.2.1, an ideal PFL will not only diffract the incident photons to a common focal point, but also counteract the phase shift caused

by the path length difference from each radial point on the lens to a common focal point.

Therefore, the thickness ( $t$ ) of a PFL, as a function of radius ( $r$ ), can be defined as [40]:

$$t(r) = t_0 + t_{2\pi} * MOD \left[ \left( \frac{r}{A} \right)^2, 1 \right] \quad (3.1)$$

where  $t_0$  is the substrate thickness,  $t_{2\pi}$  is the thickness of material required to produce a  $2\pi$  phase shift, and  $A$  is a function of focal length ( $f$ ) and target photon energy ( $E$ ):

$$A = 49.8 * \sqrt{\frac{f(m)}{E(keV)}} \mu m. \quad (3.2)$$

Thus, using (3.1) and (3.2), the radial profile of a PFL (ignoring  $t_0$ ) will look like Fig. 3.1.

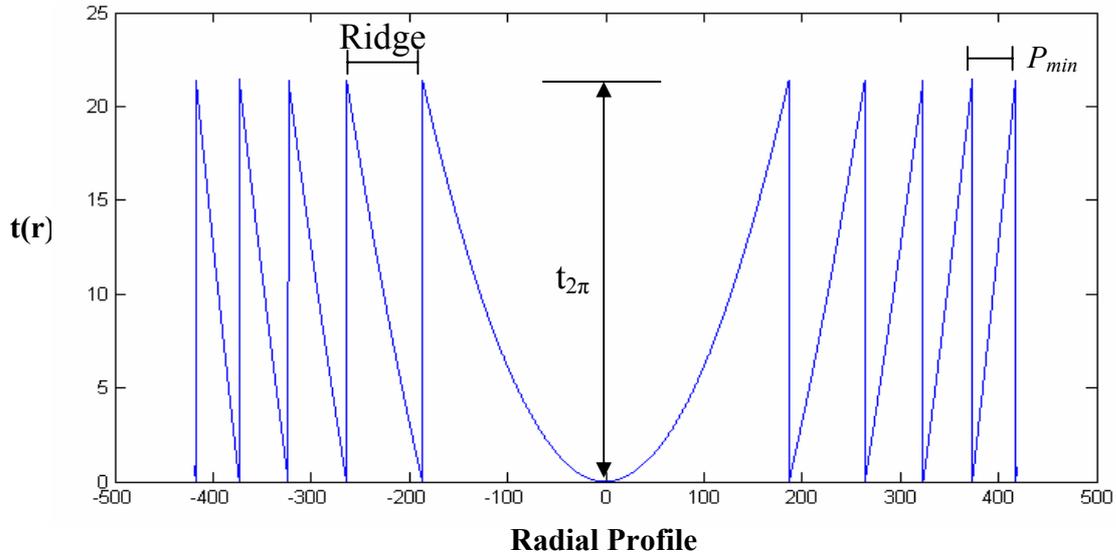


Fig. 3.1: Example PFL profile with a focal length of 118m for 8.4keV photons.

The focal length of a PFL is then a function of the zone (or ridge) widths, characterized by the minimum zone (ridge) width ( $p_{min}$ ) at the outer edge [11]:

$$f = \frac{d * p_{min}}{2\lambda} \quad (3.3)$$

where  $d$  is the diameter of the lens and  $\lambda$  is the wavelength of incident photons. Equation (3.3) can be modified for easier calculations, where  $E$  is the photon energy in keV:

$$f = 400 \left( \frac{p_{\min}}{1\text{mm}} \right) \left( \frac{d}{1\text{m}} \right) \left( \frac{E}{1\text{keV}} \right) \text{km}. \quad (3.4)$$

Equation (3.4) is useful when designing lenses because usually the energy ( $E$ ) and focal length ( $f$ ) are determined by the testing conditions and it is necessary to investigate the trade-offs between lens diameter ( $d$ ) and minimum ridge ( $p_{\min}$ ) width.

Furthermore, PFL's are not required to produce only a  $2\pi$  phase shift through the silicon. Equivalent lenses may be fabricated by increasing the phase depth in increments of  $2\pi$  [41]. If  $m$  is the number of  $2\pi$  phase depths, (3.1) becomes:

$$t(r) = t_0 + m * t_{2\pi} * \text{MOD} \left[ \left( \frac{r}{\sqrt{m} * A} \right)^2, 1 \right]. \quad (3.5)$$

The main difference seen with a deeper phase depth is that every other grating is shifted vertically by a  $2\pi$  phase depth (see Fig. 3.2). This will also cause an increase in average lens thickness, and therefore slightly increase the photon absorption in the lens.

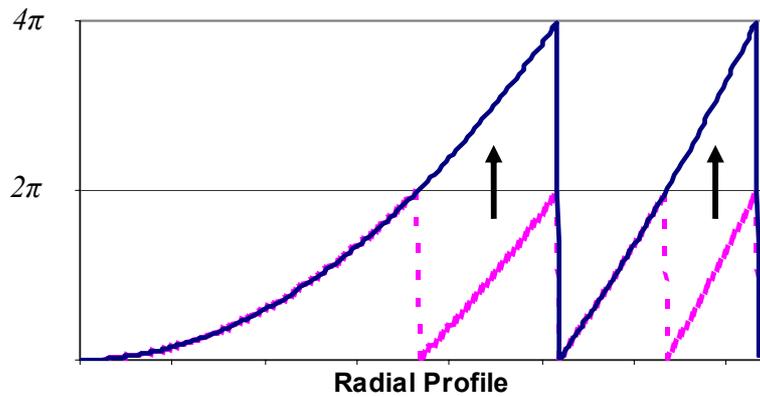


Fig. 3.2: The  $4\pi$  phase depth shifts every other grating vertically. This effectively doubles the width of each grating.

Once such a diffractive lens has been fabricated, the diffraction limited spatial resolution, or image spot size ( $s_d$ ), may be calculated using the Rayleigh criterion of a 60% power diameter [42]:

$$s_d = \frac{1.22\lambda f}{d}. \quad (3.6)$$

Although the spot size is of great importance when using Fresnel lenses in microscopy, angular resolution ( $\theta_d$ ) is much more important in astronomy and will determine the success of any tested PFL's. The angular resolution can be calculated by:

$$\theta_d = \frac{1.22\lambda}{d} = \frac{s_d}{f}. \quad (3.7)$$

### **3.3. PFL Mask Design Considerations**

The success of gray-scale technology is dependent not only on the design of the optical mask, but also on the quality of the pattern actually generated by the mask vendor. As such, the cost of an optical mask will largely be a function of the feature dimensions required. Since gray-scale technology relies on sub-resolution pixels in a projection lithography system, the dimensions of the individual pixels on a mask will be quite small ( $<3.0\mu\text{m}$ ). As such, a large number of pixels are required to make a gray-scale structure, which tends to increase the write time required by the mask fabrication tool, and that cost is often relayed to the customer. Therefore, gray-scale optical masks will likely be more expensive than other MEMS optical masks because of the high resolution and the long write times required for the intricate patterns, a price you must pay to create a 3D photoresist structure in one lithography step.

The following sections will describe the important factors to consider when defining the pitch on the optical mask, as well as the constraints imposed on pixel design by the mask vendor. These 2 factors together will limit the number of possible pixels. A procedure is also outlined for selecting the appropriate pixels to be used in the design of a

final device, and an argument provided to warrant the focus of our PFL designs to lenses with a  $4\pi$  phase depth rather than a  $2\pi$  phase depth.

### 3.3.1. Pitch selection

The choice of a ‘rectangular’ format (i.e. pixels on an X-Y grid) for designing a PFL in silicon, which is inherently circular, has some apparent drawbacks. To create a curved, sloped surface over a large area, an origin must be established and the radial distance to the center of each pixel determined. Using this radius, the desired pixel can be calculated and placed appropriately. Shown in Fig. 3.3 is an example pixel layout near the boundary of one ridge with radius  $R$ . If the center of a pixel lies just inside  $R$ , a large pixel will be chosen to create the peak of the ridge. However, an adjacent pixel that lies just outside  $R$  (labeled  $x_i$  and  $y_i$  in the figure) will be much smaller in order to start the

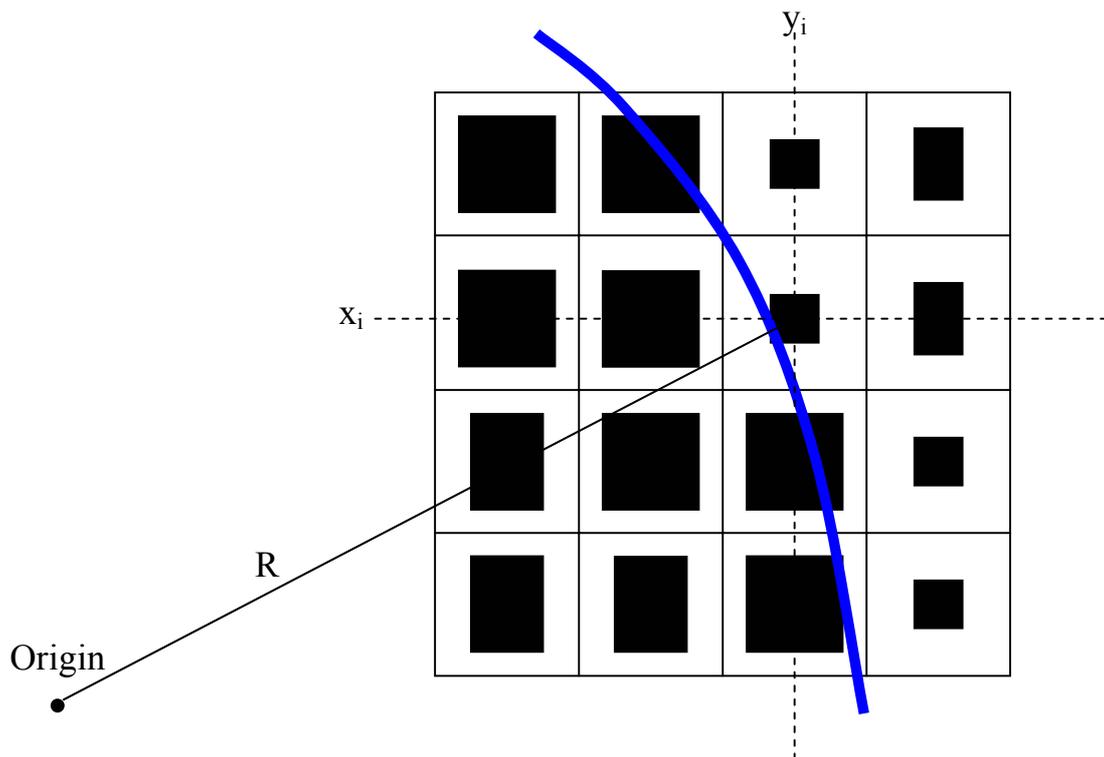


Fig. 3.3: An example pixel layout near the boundary of one ridge, shown in blue.

lower portion of the next ridge. This will cause some digital nature to our circular PFL's as the radius sweeps out a circle. The problem becomes analogous to making a circle on a piece of graph paper by filling in boxes – the edges are never quite smooth, and the accuracy of the circle depends on how large each box is (the pitch).

Thus, by selecting a pitch, the horizontal resolution of our pixel placement is defined. A pitch of  $2.8\mu\text{m}$  on the mask (the resolution limit determined in section 2.2.3), means that after 5X reduction, the resolution of our PFL will be  $\sim 0.56\mu\text{m}$ . This resolution is sufficient when making large features ( $>20\mu\text{m}$  wide), however as each ridge is reduced in width, the number of pixels that can fit in that ridge decreases. Therefore, control of the profile is reduced as fewer pixels are asked to do the work of many. The obvious method of alleviating this problem is to reduce the chosen pitch to allow a larger number of pixels to fit horizontally in the smaller ridges. However, as the pitch is decreased, the number of pixels available within the constraints of the mask fabrication will also decrease, possibly lowering PFL efficiency. Pixel constraints will be discussed in detail in the following section.

A balance must be found between the horizontal resolution of gray-scale structures in our PFL and the number of pixels available to produce the most accurate 3D slope possible. While it should be possible to increase the number of available pixels by using more accurate mask fabrication technologies, the cost of each optical mask would increase and the gain in resolution would likely be small for the PFL's being considered here. Therefore, this research will use only the critical pitch ( $P_C = 2.8\mu\text{m}$  on the optical mask), resulting in an approximate resolution of  $\sim 0.56\mu\text{m}$  on the silicon PFL.

### 3.3.2. Pixel constraints

When comparing mask vendors, there are 2 main parameters of interest in gray-scale lithography that will determine your pixel constraints: (1) minimum feature size ( $F_m$ ) and (2) electron beam spot size ( $S_0$ ). The minimum feature size is the smallest feature dimension, opaque or clear, expected to be resolved on the mask after fabrication (i.e. opaque boxes must have all dimensions larger than  $F_m$  and their edges cannot be spaced closer than  $F_m$ ). The spot size of the electron beam used to fabricate a mask is important because it defines the smallest possible increment between subsequent pixel sizes. Smaller spot sizes are advantageous because they can lead to larger pixel sets as the permutations of pixel sizes can be increased while remaining within the  $F_m$  constraints.

As described in Chapter 2 of this thesis, the method of gray-scale patterning chosen to fabricate a PFL in silicon utilizes sub-resolution pixels located in an X-Y format. Since optical masks are fabricated on systems that also use an X-Y format, it is best to use rectangular pixels conforming to such a format to avoid pixilated approximations of our pixels. Therefore, the dimensions of each rectangular pixel, height(x) vs. length(y), must satisfy the following 3 identities derived from Fig. 3.4:

$$X, Y \geq F_m \quad (3.8)$$

$$X, Y \leq (P_C - F_m) \quad (3.9)$$

$$\left. \begin{array}{l} X = F_m + n * S_0 \\ Y = F_m + m * S_0 \end{array} \right\} \text{ where } n, m = 0, 1, \dots, \frac{P_C - 2 * F_m}{S_0} \quad (3.10)$$

where  $P_C$  is the critical pitch, or resolution limit, discussed in chapter 2 of this thesis. The limiting case of (3.8) will result in the largest transmission through the mask, while (3.9)

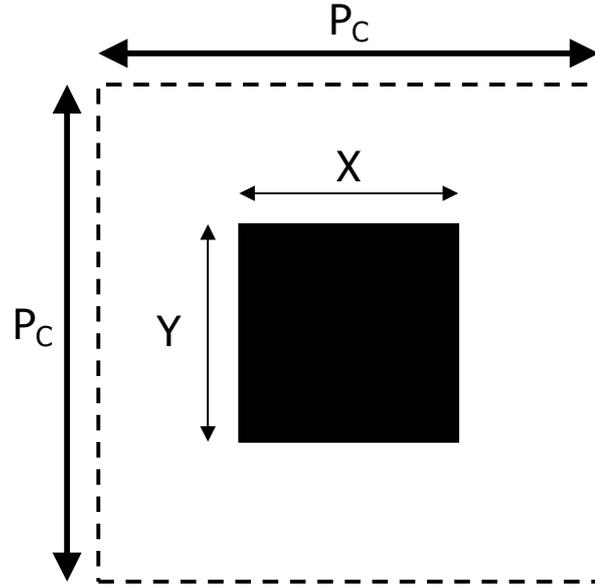


Fig. 3.4: Rectangular pixilation schematic used to define mask design constraints.

will define the smallest transmission through the mask. Equation (3.10) will determine the final number of transmissions available.

Mask fabrication for this research was done at the Northrop Grumman Corporation where  $F_m = 0.5\mu\text{m}$  and  $S_\theta = 0.1\mu\text{m}$ . Recalling the value for  $P_C$  obtained in section 2.2.3 ( $P_C = 2.8\mu\text{m}$  on the optical mask), a set of rectangular pixels can quickly be created within the constraints of  $S_\theta$  and  $F_m$ . It is then useful to modify equation (2.2), and to define the percent transmission ( $Tr$ ) to fit the rectangular case shown in Fig. 3.4:

$$Tr = 1 - \frac{X * Y}{(P_C)^2}. \quad (3.11)$$

Pixels with identical  $Tr$  values may be eliminated from any pixel set since they will be redundant. The reader must also keep in mind that every pixel will not necessarily result in a repeatable gray level in photoresist. The final pixel set is determined through the use of a calibration mask to experimentally establish the useable range of gray levels (pixels), as outlined in section 2.2.3. The selection of a large pitch

(at or near  $P_C$ ) allows more gray levels, however it means that the shape of the pixels may be non-trivial as small ‘spikes’ in the photoresist have been observed between gray levels with highly elongated pixels. Consequently, an extra design condition was imposed to restrict the pixel height (x) and length (y) to have a ratio close to 1 (square), namely:

$$\frac{2}{3} < \frac{X}{Y} < \frac{3}{2} \quad (3.12)$$

### 3.3.3. Choice of gray levels

Once a set of pixels has been established for the chosen pitch, it must be decided which pixels will be used in a design. Under certain circumstances it may not be realistic to use every possible pixel, and in some cases small tradeoffs must be considered. As an example, Table 3.1 Shows a list of 21 pixels possible for a pitch of 2.2  $\mu\text{m}$  assuming  $F_m = 0.6 \mu\text{m}$  and  $S_0 = 0.2 \mu\text{m}$  (2.8  $\mu\text{m}$  pitch is not used as an example because of the large

Pixel #	X ( $\mu\text{m}$ )	Y ( $\mu\text{m}$ )	Tr	Yes/No
1	0.6	0.6	0.074	No
2	0.6	0.8	0.099	No
3	0.6	1.0	0.124	No
4	0.8	0.8	0.132	No
5	0.6	1.2	0.149	No
6	0.8	1.0	0.165	Yes
7	0.6	1.4	0.174	No
8	0.8	1.2	0.198	Yes
9	0.6	1.6	0.198	No
10	1.0	1.0	0.207	No
11	0.8	1.4	0.231	Yes
12	1.0	1.2	0.248	No
13	0.8	1.6	0.264	Yes
14	1.0	1.4	0.289	Yes
15	1.2	1.2	0.298	No
16	1.0	1.6	0.331	Yes
17	1.2	1.4	0.347	No
18	1.2	1.6	0.397	Yes
19	1.4	1.4	0.405	Yes
20	1.4	1.6	0.463	Yes
21	1.6	1.6	0.529	Yes

Table 3.1: Mock selection of pixels using a 2.2 pitch and fictional design rules.

increase in rectangular permutations). A mock pixel selection procedure would go as follows: It is important to eliminate pixels that will not be created consistently because any loss of bottom gray levels will result in flat areas at the bottom of each ridge, deforming the PFL profile and decreasing the efficiency. Supposing from the calibration mask that  $Tr < 0.15$  does not result in a useable gray level, pixels 1-5 are eliminated. Pixels 8 and 9 result in the same  $Tr$  value and are therefore redundant, so pixel 9 is eliminated. One can then compare the  $Tr$  values for the remaining pixels. If 2 successive pixels are sufficiently close, it is unlikely that they will be consistently resolved in the photoresist and one can usually be eliminated (the uncertainty during the rest of the gray-scale process will be more of a determining factor). Using an arbitrary threshold of 0.02, pixels 7, 10, 12, 15, and 17 are eliminated. The remaining 10 pixels can then be used to produce any final structures.

Depending on the PFL dimensions, it may be beneficial to design for a smaller number of well-defined levels rather than a larger number of less-uniform levels. In this way, the levels could be more consistently defined and achieved. Additionally, the pattern transfer may benefit from having a more step-like profile because of cycling the passivation and etch steps during DRIE. If the profile becomes truly sloped, then the passivation will no longer be on a 'vertical' sidewall where it is protected, and may be prematurely removed, resulting in a loss of profile control.

The selection of a pixel set will obviously have an impact on the PFL design. If only 10 pixels are possible, and therefore 10 gray levels, then the max efficiency using equation (1.1) is 96.7%, which may or may not be sufficient depending on the other design parameters of the lens. However, the number of gray levels available becomes

increasingly important if a deeper phase depth is used, since twice as many gray levels will be necessary to maintain a similar efficiency. As a minimum, 8 or more levels per phase depth should be available in our PFL designs in order to target efficiencies at or above 90%.

#### **3.3.4. Multiple Phase depths**

As stated earlier in section 3.2, equivalent PFL's may be constructed by increasing the phase depth from  $2\pi$  to  $4\pi$  or more, with the penalty being an increase in absorption. Since silicon PFL's are being considered here, the increase in absorption for a  $4\pi$  lens, especially as the energy is increased, is tolerable. Instead of an average thickness of  $t_{2\pi}/2$ , there will be an average thickness of  $t_{2\pi}$  over the entire lens. For 8.4keV photons with a photon attenuation length of  $80\mu\text{m}$  in silicon, the intrinsic absorption in the lens increases from 13% for a  $2\pi$  phase depth to 23% for a  $4\pi$  phase depth. The intrinsic absorption will be smaller for higher energy lenses.

Designs using multiple phase depths for this PFL are specifically investigated because of the nature of gray-scale technology and the peculiar dimensions required. By increasing the phase depth from  $2\pi$  to  $4\pi$ , the width of each ridge in photoresist is effectively doubled. Since gray-scale technology can easily deliver 30+ gray levels using  $P_C$ , more than 16 gray levels per phase depth are still possible. One advantage of doubling the width of each ridge in photoresist is that the lithography processing is more readily achieved because the aspect ratio of the photoresist structures is reduced.

A second important aspect of increasing the phase depth is that the etch depth required in DRIE is doubled, which will assist in the pattern transfer. Consider an

8.4keV PFL. The  $t_{2\pi}$  phase depth using (1.2) is  $21.4\mu\text{m}$  in silicon, which can be easily achieved in silicon using DRIE. Since the lithography processing is more consistent with thicker photoresist and thicker gray levels, an entire gray-scale pattern is commonly  $\sim 3\mu\text{m}$  thick. Such a pattern would then require a selectivity of 7:1 to transfer it entirely, which is extremely low (see section 2.3.2). By increasing the target phase depth to  $4\pi$  and doubling the number of gray levels, the target depth is now  $42.8\mu\text{m}$  and the selectivity can be increased to 14:1. This etch selectivity is still very low for DRIE, but is much more reasonable. Specific steps taken to address this issue will be discussed in Chapter 4 of this thesis. Increasing the phase depth beyond  $4\pi$  for an 8.4keV lens will further increase the absorption and require a larger number of gray levels. Therefore, 8.4keV PFL's with  $4\pi$  phase depths were chosen as the focus of this research.

### **3.4. Gaussian Approximation Method**

To create silicon PFL's with high efficiency, every pixel in the optical mask design must be individually placed. Previously, rectangular gray levels were created in this research by placing a single pixel and repeating it in an array. For small geometries with few gray levels, a structure design could be done entirely by hand with only a small knowledge of previous experiments. However, for millimeter sized PFL's, each gray level will be curved and as such each pixel must be placed individually. With  $>10^6$  pixels required to make each test lens, an automated system is necessary to create the appropriate pixel at every point. The Gaussian approximation method presented in the next sections will outline the method by which an optimum pixel may be selected for each pixel location to accurately re-create any desired profile, including but not limited to, the PFL profile defined in equation (3.5). Alternate gray-scale photoresist

linearization schemes have been investigated [43]. Linear gratings with 8 levels were fabricated, however the authors did not demonstrate arbitrary profiles, nor did the model facilitate integration into complicated structure designs. The authors also used High-Energy-Beam-Sensitive (HEBS) glass in contact lithography rather than the pixilated approach with projection lithography that was selected for this research.

### 3.4.1. Experiment

The Gaussian approximation relies on the calibration mask of section 2.2.3 and the standard lithography process defined in section 2.2.4. First, the standard gray-scale lithography process is performed using the calibration mask. Each wedge designed on the calibration mask has an opaque photoresist block at one end as a reference point, with adjacent large square gray levels made with pixels of descending size. The photoresist height of each level is associated with a particular pixel by counting the steps from the opaque square. Thus, for a given photoresist wedge produced by the calibration mask, the  $Tr$  value can be calculated and related to the final photoresist height. Table 3.2 shows some example data taken from structures with a pitch of  $2.8\mu\text{m}$ . The  $Tr$  value, from the

Square Pixel Size ( $\mu\text{m}$ )	$Tr$	Normalized Height in Photoresist, $H_p$ ( $\mu\text{m}$ )			
		Sample 1	Sample 2	Sample 3	Sample 4
2.1	0.438	0.39	0.42	0.42	0.42
2.0	0.490	0.30	0.33	0.32	0.32
1.9	0.540	0.23	0.24	0.25	0.25
1.8	0.587	0.18	0.19	0.19	0.19
1.7	0.631	0.12	0.14	0.14	0.14
1.6	0.673	0.09	0.09	0.10	0.10
1.5	0.713	0.05	0.05	0.06	0.07

Table 3.2: Gray level height data taken from a structures using a  $2.8\mu\text{m}$  pitch.

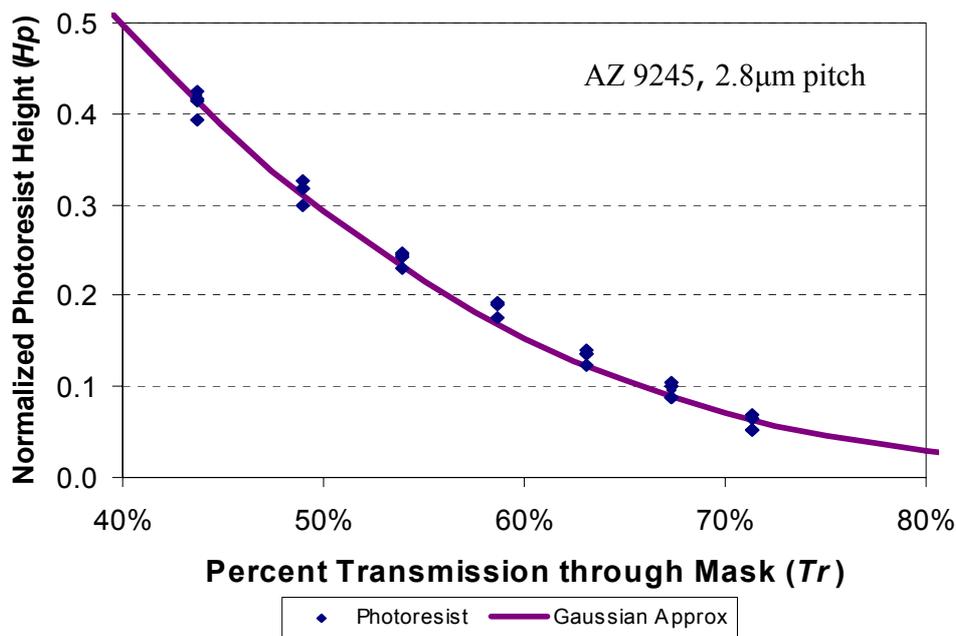


Fig. 3.5: Gaussian curve fit through data points relating normalized photoresist heights to the percent transmission through the mask that caused them.

pixel used, can then be plotted against the normalized height in photoresist ( $H_p$ ) that each pixel produced (measured by contact profilometer), as done in Figure 3.5. A Gaussian curve was then fit between the data points to create a relation between  $Tr$  and  $H_p$ :

$$H_p = A_0 * \exp(-(Tr)^2 * \gamma) \quad (3.13)$$

where  $A_0$  and  $\gamma$  are the fit parameters for the particular photoresist being used.

For the case shown in Figure 3.5,  $A_0 = 1.28$  and  $\gamma = 5.9$  (with  $R^2=0.986$ ). Although it would be possible to fit this data more accurately, perhaps using a high order polynomial, a Gaussian fit was chosen for two specific reasons. First, photoresist absorption curves strongly resemble exponential or one-sided Gaussian functions, so the resulting heights in photoresist were expected to follow this trend as well. Second, a Gaussian is much easier to manipulate than a high order polynomial or other exotic function that might have a better  $R^2$  value. The fit shown in Fig. 3.5 could be further

optimized by either incorporating small offsets in  $x$  or  $y$ , or possibly using a slightly simpler exponential form:

$$H_p = A_0 * \exp(-Tr * \gamma) \quad (3.14)$$

As is evident in Figure 3.4, the Gaussian approximation provides a reasonably good fit for the data in question for initial PFL iterations, and as such will be extended as the method of choice throughout this analysis.

Once suitable fit parameters have been determined, the true power of the Gaussian approximation is realized as the equation is reversed:

$$Tr = \sqrt{\frac{-\ln\left(\frac{H_p}{A_0}\right)}{\gamma}}. \quad (3.15)$$

In equation (3.15), we now have a case where  $Tr$  is only a function of  $H_p$ , the normalized height in photoresist. Therefore, equation (3.15) has the ability to calculate the optimum  $Tr$  value necessary to produce a desired  $H_p$ . Naturally the desired  $H_p$  is simply a normalized version of equation (3.5), meaning that the optimum  $Tr$  value is now simply a function of the radius. By combining equations (3.15) and (3.11), the pixel size can be directly related to the desired  $H_p$  at each radial point:

$$A_{pixel} = X * Y = (P_C)^2 * \left[ 1 - \sqrt{\frac{-\ln\left(\frac{H_p}{A_0}\right)}{\gamma}} \right]. \quad (3.16)$$

By using the Gaussian approximation method described above, much of the modeling and theory behind the gray-scale lithography process may be abbreviated. This approximation is especially helpful since the gray-scale process is so heavily dependent

upon process conditions that are difficult to model from one lab to the next. Additionally, creating a custom Gaussian approximation for a new lab can be carried out simultaneously as a standardized lithography process is developed with the calibration mask, as would be done for traditional lithography anyway. Therefore, with the addition of only a few profilometer measurements and one curve fit, gray-scale masks may be designed to precisely mimic any desired slope.

### **3.4.2. Integration into C Program Design**

As stated earlier, the individual placement of each pixel may not be realistically done by hand. By incorporating equation (3.16) into a program developed using the C++ programming language, all pixels may be automatically determined with ease and read into a CAD layout tool with scripting capabilities (Tanner's L-edit). (Throughout the following analysis, it is assumed that a pixel set has already been established.) Additionally, by symmetry, a circle defined on an X-Y grid can be created by defining only the 1<sup>st</sup> quadrant, and then using the CAD tool to mirror and/or rotate 3 more 90° wedges. Thus only the pixels necessary to make ¼ of a PFL must be individually defined (or 1/8 of a PFL with 7 more identical wedges).

Assuming the desired focal length, photon energy, phase depth, and radius (R) on the wafer have been defined by the user, a C program can simply walk through and address every point on the mask's X-Y grid from the origin (0,0) to the boundary of the lens (5R, 5R) (assuming a 5X reduction by the projection lithography system will result in a PFL of radius R). By incrementing x and y by  $P_C$  on the optical mask, every pixel location in x and y on the interval 0 to 5R will be individually addressed. Since PFL

equation (3.5) is defined by the radial distance from the origin, the program will transform each  $x_A$  and  $y_B$  coordinates into a radial coordinate,  $r_{AB}$ :

$$r_{AB} = \sqrt{\left(\frac{x_A}{5}\right)^2 + \left(\frac{y_B}{5}\right)^2}. \quad (3.17)$$

To find the desired  $Hp$  at  $r_{AB}$ , the part of equation (3.5) that is radially dependent is normalized to the highest expected  $Hp$  ( $Hp_{max}$ ) of the pixel set being used:

$$Hp = t(r_{AB}) = Hp_{max} * MOD\left[\left(\frac{r_{AB}}{\sqrt{m} * A}\right)^2, 1\right]. \quad (3.18)$$

Plugging equation (3.18) into (3.16), the optimum pixel size required to produce such a height in photoresist is obtained. Since the optimum pixel size will probably not exist, the optimum value can be compared to the available pixels and rounded to the closest pixel. It is through this final rounding procedure that the stair-step nature of our PFL comes about, as each point on the curve is rounded to the nearest available step.

As a quick aside, this rounding effect may influence the desired pixel set. As the size of each pixel increases, the difference in area between pixel permutations become farther and farther apart. This gap in area creates large gaps between  $Tr$  values which will cause the profile to look rather poor as the steps towards the top of a device become much larger than the steps at the bottom of a device. Such a non-uniform step approximation to the desired profile, which was highlighted by the Gaussian approximation, will decrease the efficiency of any PFLs created in this manor. Consequently, it may be appropriate to truncate the pixel set before using all possible pixels in order to maintain evenly spaced gray levels.

Once the best pixel has been selected for a particular point, the program outputs a set of commands into a text file specifying the location and size of a box to be created by the CAD layout tool. Once all locations have been addressed, the final text file may be imported into the CAD layout tool using command line scripting. Since the number of pixels is often very large ( $>10^6$ ), the final text file may take as long as 1-2 days to import an entire design into a layout file (Dell PC with 1.8GHz Pentium 4 processor and 512MB RAM), creating files as large as 500MB. Such large files make data storage and manipulation tedious while finalizing the rest of the mask design, and any changes that require re-creating the design will come with an enormous time penalty. This penalty can be alleviated somewhat by making multiple smaller command files to allow for some design inspection, hopefully avoiding minor programming mistakes that could cost days of computation time.

### **3.5. PFL Device Design Parameters**

In addition to various test structures, three primary PFL designs were developed and placed on two successive optical masks. Table 3.3 shows the properties of each of these three lenses. Chronologically, lens A was the first full PFL to be fabricated in silicon. Lenses B and C followed with more accurate Gaussian fit parameters and larger diameters.

The combination of these three lenses will enable the investigation of multiple aspects of PFL design and fabrication using gray-scale technology. To evaluate the profile difference between fewer sharply defined gray levels and a larger number of ‘blended’ gray levels, lenses A and B have been designed with 32 levels, while lens C has

Lens	A	B	C
Mask/structure	DP/F6	4GF/X3	4GF/X6
Energy (keV)	8.4	8.4	8.4
Focal Length (m)	118	118	118
Diameter (mm)	1.6	3.0	4.7
Number of Ridges	10	32	80
Phase Depth	$4\pi$	$4\pi$	$4\pi$
Number of Gray levels	32	32	16
Gray levels / $2\pi$ phase depth	16	16	8
$P_{min}$ ( $\mu\text{m}$ )	43	24	15
Max Aspect Ratio	1.0	1.7	2.7
Pitch ( $\mu\text{m}$ )	2.8	2.8	2.8
Gaussian Fit Parameters ( $A_0, \gamma$ )	(1.65, 6.9)	(1.28, 5.9)	(1.28, 5.9)
Number of Pixels ( $90^\circ$ wedge)	$1.75 \times 10^6$	$5.63 \times 10^6$	$13.8 \times 10^6$

*Table 3.3 Properties of the 3 primary PFL's designed for this research.*

been designed with only 16 levels. By incorporating a gradual increase in diameter from lens A to lens C, any factors that could limit  $P_{min}$  and/or the maximum aspect ratio, can be investigated, which could ultimately limit the diameter of any future PFL devices. Additionally, the improvement seen between successive, and more accurate, Gaussian approximations should become clear when the profiles are evaluated.

### **3.6. Summary**

The main steps taken to develop precisely defined gray-scale optical masks have been outlined in this chapter. The constraints and suggestions presented regarding the design and selection of a pixel set should be considered guidelines to be easily adapted to any gray-scale application. The Gaussian approximation method was introduced as a

method of precisely placing many appropriate pixels to create a desired height profile in photoresist, in our case re-creating the PFL equations given in section 3.2. A method of integrating pixel design limitations and the Gaussian approximation into a custom C program was outlined and used to develop multiple gray-scale optical masks containing three different PFL designs. The success and accuracy of the design methodologies presented herein will be assessed in Chapter 4 of this thesis as the fabrication aspects of developing a PFL in silicon are discussed and their profiles evaluated.

## **4. Lens Fabrication**

### **4.1. Introduction**

As discussed in the previous chapter, a large amount of work goes into the design of each gray-scale structure created. However, a wide understanding of the fabrication challenges involved is necessary when attempting to produce an accurate structure in the silicon, largely because the mask design is dependent on consistent lithography processing and precise control of etching characteristics. The typical process flow for a PFL wafer starts with the lithography, then the photoresist height is measured to determine the appropriate etch selectivity, and the wafer is subsequently etched in DRIE with the best available etch recipe. For the results presented in this chapter, all lithography processing was carried out by C. Mike Waits at the Laboratory for Physical Sciences (LPS) in College Park, MD, while I performed all dry etching at the U.S. Army Research Laboratory (ARL) in Adelphi, MD. Analysis tools used to evaluate and image various results were located at ARL or on the campus of the University of Maryland.

This chapter will address all aspects of this research regarding the fabrication of a PFL in silicon. Lithography results will be discussed in detail for ridge widths ranging from hundreds of micrometers down to a few micrometers. Examples will be given that show the impact of using a pixilated design for fabricating PFL's in photoresist, with comments on how these results will limit future designs. Contact and optical profilometer measurements of photoresist profiles will confirm that the methodologies employed for designing our PFL masks, namely the Gaussian approximation and pixel design methods outlined in Chapter 3, accurately mimic the desired PFL profile. Key

issues regarding the DRIE transfer of a photoresist PFL will be discussed, and a modification to the standard Bosch process will be introduced as a means of lowering etch selectivity to the desired range. Etch results obtained using this low selectivity process will be presented and the method by which metrology is performed on a silicon PFL profile will also be discussed. Two problems will be analyzed with respect to their possible impact on lens performance: aspect ratio dependent etching (ARDE) and surface roughness. Finally, a method will be introduced to calculate the approximate efficiency of a silicon PFL profile without testing the lens in an X-ray beam line. This method will be used to compare the different silicon PFL profiles designed and fabricated.

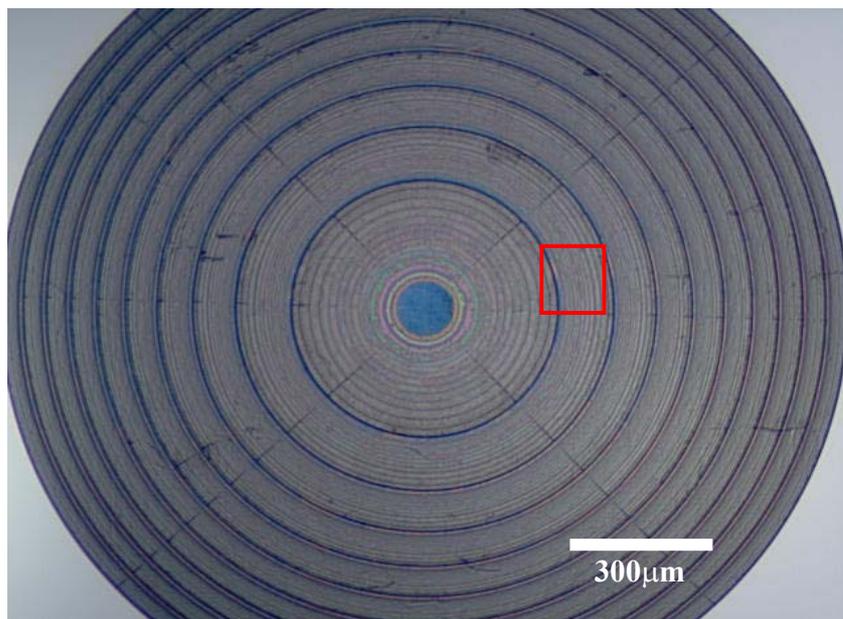
## **4.2. Lithography Results**

All lithography results in this chapter were achieved using the standard lithography process developed in Chapter 2 of this thesis. The following sections will present and comment on multiple photoresist PFL structures that have been fabricated using gray-scale lithography. First, lithography results will confirm the digital behavior of the circular approximations made when designing our PFL structures (discussed in Chapter 3). The limiting cases of this digital behavior will be presented. The metrology method used to investigate the photoresist profiles will be discussed as it is an important aspect of evaluating each photoresist profile. Ultimately, these metrology measurements will be used to confirm the accuracy of the mask design method developed in Chapter 3.

### **4.2.1. Photoresist PFL Structures**

A number of photoresist PFL's and test structures have been fabricated during this research, yet only a small sampling will be presented in this thesis in order to highlight

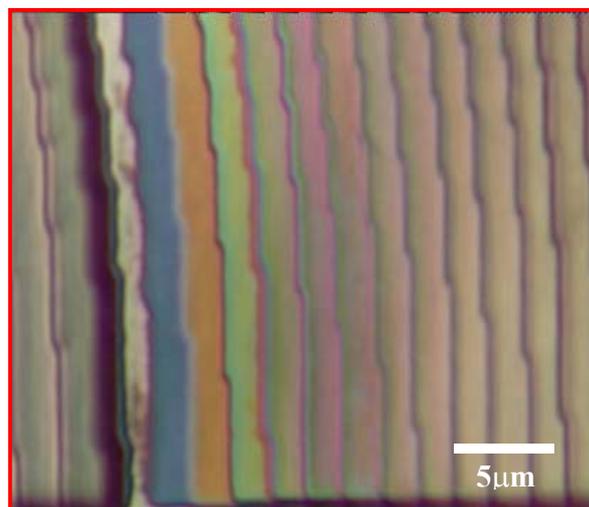
the issues regarding the fabrication of a PFL with this technology. A picture of the first demonstration of a photoresist PFL is shown below in Fig. 4.1. This PFL is described as lens ‘A’ in Table 3.3. Fig. 4.1 is helpful in pointing out an important characteristic of gray-scale photoresist structures, namely that each thickness of photoresist appears as a different color. This characteristic is especially apparent in the lower gray levels where the center of the PFL appears blue in color indicating that it is a very thin gray level, which aids in identifying the appropriate development time for a particular sample. As the development step progresses, the photoresist is slowly thinned across the entire wafer. Since the areas outside of the PFL receive a full exposure dose, they will develop out first, indicating a complete development step. When the open areas have their blue level washed away, the development step should be terminated to avoid over development of the gray-scale structure, effectively washing away any lower gray levels as well. Thus, the color of the photoresist can be used to accurately control the manual development step. It must also be noted that the apparent ‘cross-hair’ lines in Fig. 4.1 are a result of



*Fig. 4.1: The first demonstration of a photoresist PFL.*

creating only a wedge of the PFL in the mask design and rotating multiple wedges to create the entire profile. The lines are a result of the overlap of pixels where the wedges intersect.

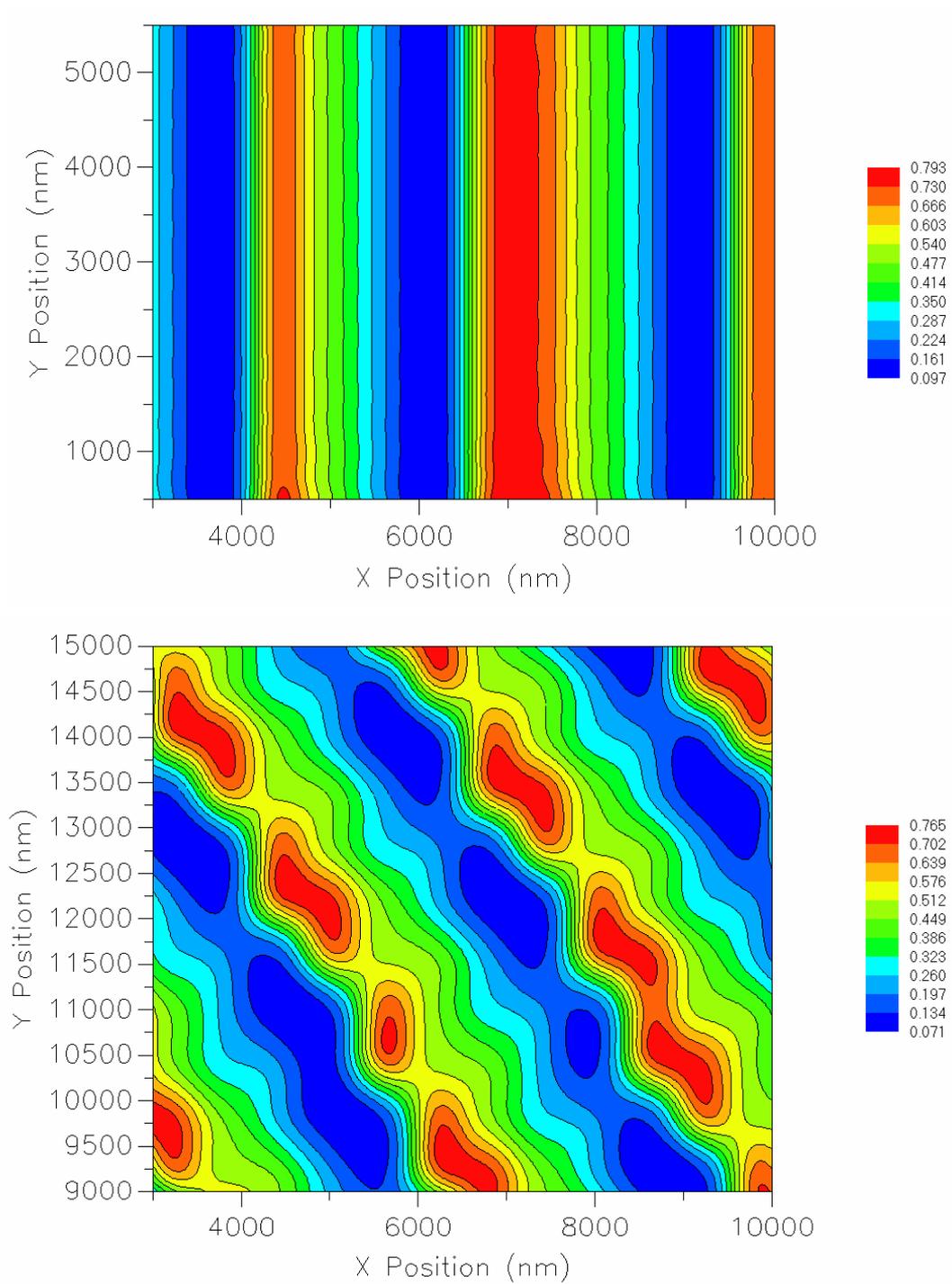
Fig. 4.2 has zoomed in on a single ridge of a photoresist PFL (marked with a red box in Fig. 4.1) so that each gray level can be easily identified. Again, the thinnest remaining gray level appears blue in color. This figure points out two distinct challenges when fabricating a curved PFL. First, should the sample be over-developed, the lowest gray levels will be washed away, causing the profile to have a flat region at the bottom of each ridge. Such a flat region will deform the ideal sawtooth profile and decrease the lens efficiency. Second, there is a small digital nature to each gray level as it curves around a circle. The digital nature arises from the fact that we are creating a circle out of many small pixels placed on a periodic X-Y grid (as described in section 3.31 and Fig. 3.3). The raster effect exhibited in each gray level in Fig. 4.2 is approximately  $0.5\mu\text{m}$ , the horizontal resolution of our pixilated mask design. This shift is an insignificant portion of large ridges ( $>15\mu\text{m}$  wide), so the intensity and profile patterns are not greatly



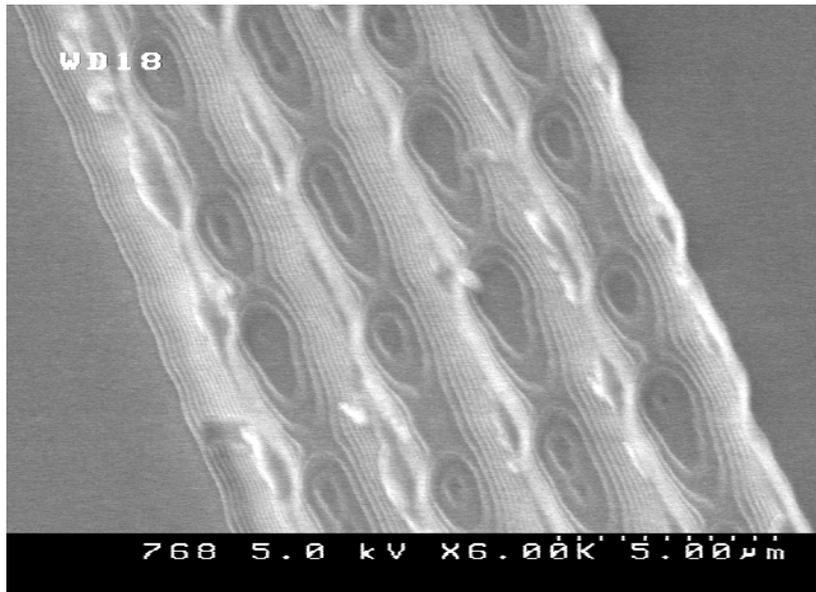
*Fig. 4.2: PFL ridge exhibiting a digital nature to the curved gray levels. Location on overall PFL is shown in Fig. 4.1 as a red box.*

affected, and the PFL is quite symmetric (radial symmetry will be evaluated later in section 4.4.2). However as ridge dimensions decrease and fewer pixels are used to make each gray level, this raster effect becomes a large fraction of the overall ridge width, resulting in a non-uniform intensity transmission through the mask.

The photolithography simulation software PROLITH (from KLA-Tencor), which uses Electro-Magnetic wave propagation, was used to simulate the relative intensity profile transmitted through a gray-scale optical mask for some test PFL structures with small ridges  $\sim 3\mu\text{m}$  wide (made with  $\sim 5$  pixels per ridge). Fig. 4.3 (a) shows the resulting relative aerial intensity for a horizontal ridge geometry, while (b) shows the aerial intensity resulting from a curved ridge geometry. As is evident from Fig. 4.3(a), pixels arranged to create a horizontally graded intensity profile are successful in doing so. However, Fig. 4.3(b) shows aerial intensity resulting from the same pixels arranged to create a graded intensity profile at an arbitrary direction, while the desired intensity profile would look like Fig. 4.3(a) rotated  $45^\circ$  to the left. The effects on the intensity profile are disastrous. Since few pixels are being used to create the profile, the digital nature of the pixilated mask design causes drastic changes in intensity from one pixel location to the next. Fig. 4.4 is an SEM image of a photoresist sample resulting from the non-uniform intensity transmission in Fig. 4.3(b). The photoresist exhibits a non-uniform profile with similar contour lines to the intensity profile in Fig. 4.3(b). Using a smaller pitch during optical mask design could reduce this digital effect, however this will likely require more expensive optical masks. Therefore, the PFL's developed in this thesis do not contain ridges smaller than  $15\mu\text{m}$  wide in order to avoid severely non-uniform intensity transmission through the optical mask.



*Fig. 4.3: Relative intensity profiles made with ~5 pixels for (a) horizontal ridge geometries and (b) curved ridge geometries.*



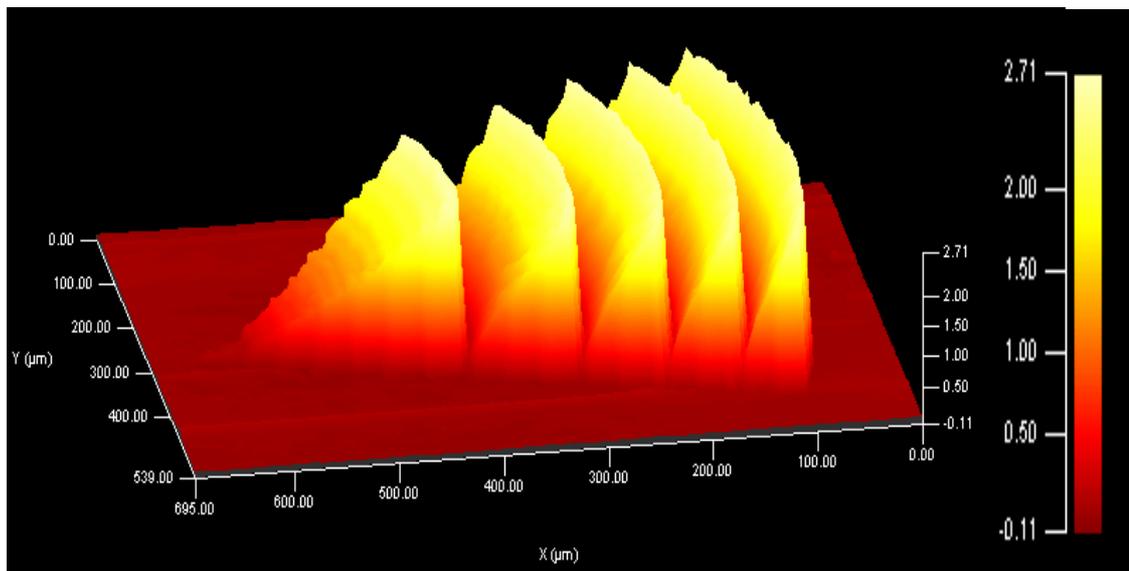
*Fig. 4.4: SEM image of small curved ridges in photoresist resulting from the non-uniform intensity transmission exhibited in Fig. 4.3(b).*

#### **4.2.2. PFL Metrology (Photoresist)**

We have shown that photoresist PFL structures have been fabricated using gray-scale lithography, however the goal of this research is to produce structures with specific profiles, and therefore we must investigate exactly what profile has been fabricated in the photoresist. Metrology measurements of photoresist are often done with a contact profilometer, which physically drags a stylus tip across a photoresist sample in order to measure the profile. Provided the contact force is low, this method generally will not scratch or deform your photoresist. Another option for measuring our photoresist profiles is to use an optical profilometer, which measures light reflected from the surface. Since photoresist is semi-transparent, samples measured by optical profilometry were coated with a 20nm layer of Au. Each measurement method has advantages and will be used for separate purposes described below.

Beginning with contact profilometry, a number of photoresist PFL's and test structures were measured. Shown in Fig. 4.5 is an example contact profilometer measurement taken with a Tencor P-15 Long Scan Profilometer at ARL. The 3D image is constructed by taking multiple line-scans and interpolating the regions between scans to create a 3D plot. Measurements such as this can easily extract the height of the photoresist structure, essential for determining the desired etch selectivity (depending on the structure height in photoresist, etch selectivity may need adjustment). The photoresist sample may be subsequently etched to the desired depth with the appropriate etch recipe.

Unfortunately, the stylus tip on a contact profilometer has a finite radius of curvature (usually over  $1\mu\text{m}$ ), and the tip narrows to that point with an angle near  $60^\circ$ . This may be sufficient for measuring a large step height, but it presents problems for investigating a PFL profile made with gray-scale technology. First, the finite size of the stylus tip may obscure some detail about each gray level present in the photoresist if the levels are narrow and close in height. Second, the approach angle of the stylus tip means



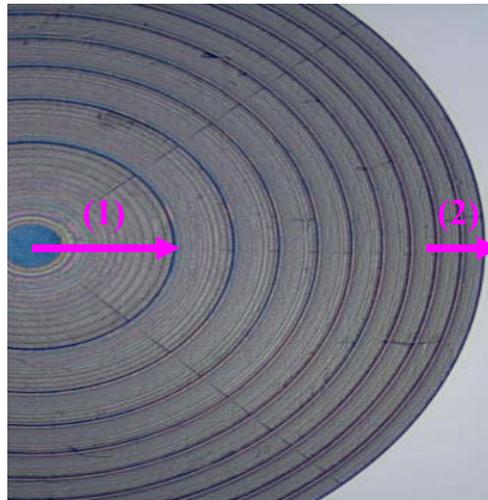
*Fig. 4.5: Example 3D photoresist measurement taken with a contact profilometer.*

that vertical profiles will not be traced precisely. As the dimensions of our ridges are reduced, the stylus tip will no longer fit between the ridges to accurately trace the photoresist profile. Finally, contact profilometers only scan in a straight line and the precise location of the scan on the structure is hard to determine. Thus, if a scan has an unexpected feature in the profile, it can be difficult to differentiate between a true profile characteristic and a fabrication defect scanned by chance. If a 3D measurement is taken, a photoresist defect, such as a streak or bubble, may be identified. Unfortunately, the 3D scan can only extract 2-D data plots for each line scan taken, so the operator cannot choose an arbitrary direction on the structure to avoid the defect. The sample could be rotated and re-scanned, but the time to re-scan a 3D area can be as long as a few hours. Because of these limitations, contact profilometer scans in this research were used primarily to measure the height of PFL photoresist profiles. These measurements were then used to establish the etch selectivity required to produce the desired PFL height in silicon, and not to evaluate the contours of the photoresist PFL.

The primary drawback of using an optical profilometer (in this case a WYKO NT1100 by Veeco) is that the thin layer of deposited Au, used to increase the reflectivity of the sample, prevents the particular *photoresist* sample from further processing to become a silicon PFL. However, optical profilometers are not subject to the same limitations imposed by a stylus tip, and can therefore measure nearly vertical profiles with high vertical and horizontal resolution. Additionally, optical profilometers take an area scan in under 30 seconds (vs. hours using a contact profilometer) and profile measurements may be taken along any line drawn in the scan area. Thus, an optical profilometer is used to measure only those samples that will not be processed further.

### 4.2.3. Gaussian Approximation Confirmation

To confirm the validity of the Gaussian approximation method introduced in chapter 3, an optical profilometer was used to image photoresist samples coated with Au because of its ability to accurately image a large area profile regardless of aspect ratio. Fig. 4.6 shows the 2 primary locations measured on our photoresist PFL's. These locations were chosen because the ideal lens profile at location (1) resembles an exponential function, while at location (2) it should appear linear. If successful, this should demonstrate that the Gaussian approximation method is capable of fabricating arbitrary profiles.



*Fig. 4.6: Locations measured to investigate the accuracy of the Gaussian approximation method.*

Fig. 4.7 shows photoresist measurements taken with an optical profilometer, plotted with the desired radial photoresist profile. The optical mask that produced these structures was designed using the Gaussian approximation method presented in section 3.4. It can easily be seen that the measured photoresist profiles are quite close to the designed case. We attribute some of the scattering of the data to be scattered light

reflecting from the thin Au coating and the photoresist surface roughness. The strong resemblance between the fabricated photoresist PFL profiles and the desired profiles confirms that the Gaussian approximation method introduced in this thesis is a valid and useful tool for precisely designing 3D structures using gray-scale technology.

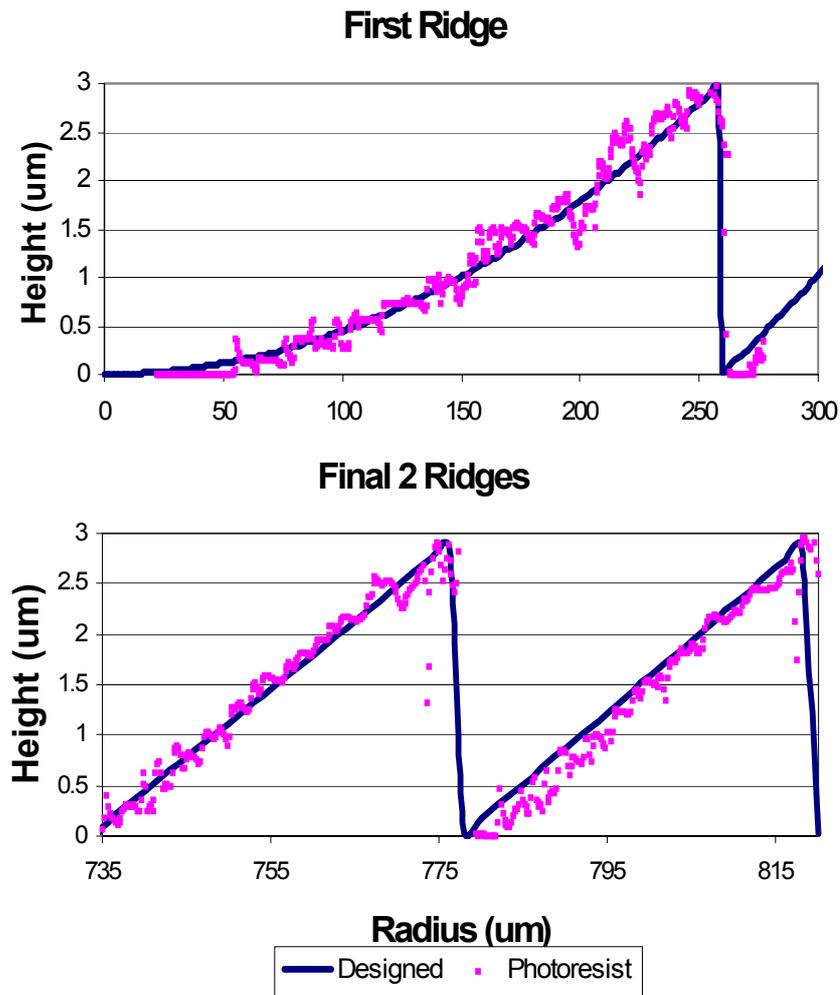


Fig. 4.7: PFL photoresist profiles fabricated using the Gaussian approximation method closely mimic the desired profile.

### 4.3. Dry Etching Results

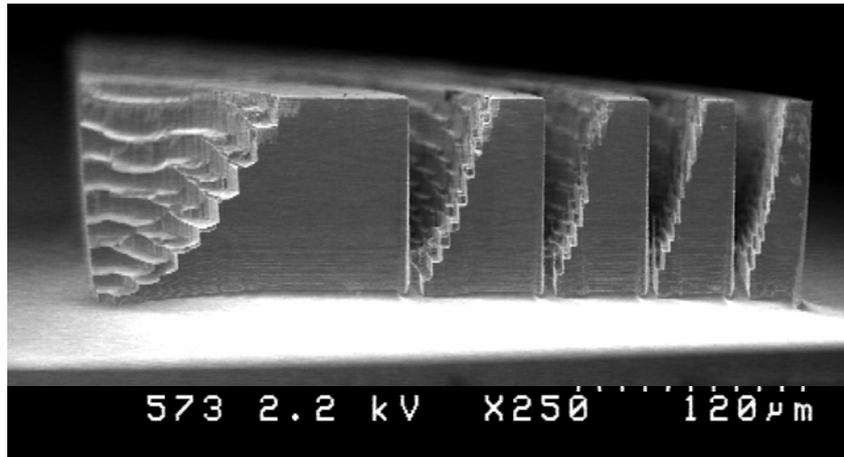
The importance of accurately transferring a 3D photoresist gray-scale structure was underscored in section 2.3.2 as the vertical dimensions of a silicon gray-scale

structure will be dependent on the etch selectivity during DRIE. An initial investigation regarding the control of etch selectivity during DRIE was presented in section 2.3.3. Since the PFL photoresist patterns described in the preceding sections are routinely  $\sim 3\mu\text{m}$  thick, and given the target  $4\pi$  phase depth of  $43\mu\text{m}$ , a controlled DRIE etch selectivity of  $\sim 14:1$  will be necessary to achieve the appropriate vertical dimensions in silicon (each wafer will be measured individually before etching to allow any necessary final tuning of etch selectivity).

The following sections will discuss the results achieved using traditional etch recipes like those presented in section 2.3.3, and also results achieved using a novel modified Bosch process with an Oxygen-only step. Analysis will then be provided regarding the aspect ratio dependent etching (ARDE) characteristics of the successful etch recipes, and the expected impact of surface roughness on PFL performance.

#### **4.3.1. General DRIE Results**

During the etch selectivity characterization of section 2.3.3, a number of test PFL structures were successfully transferred into the silicon. However, each etch selectivity in that characterization was much higher than desired, producing adverse results. An example of a transferred structure is shown below in Fig. 4.8. This  $90^\circ$  PFL wedge was etched using a recipe with an etch selectivity of 70:1. After etching to a depth of  $100\mu\text{m}$  in the silicon, the entire gray-scale photoresist pattern still has not fully transferred, when ideally the entire pattern needs to be transferred in less than half this depth. It is also easy to see in Fig. 4.8 that some gray levels appear uneven, and this non-uniformity is attributed to the photoresist structures and not the etch process. Fig. 4.8 shows the first successful transfer of a curved PFL test structure into the silicon.



*Fig. 4.8: Initial pattern transfer of a 90° PFL wedge with a high selectivity.*

#### **4.3.2. DRIE with Oxygen-only step**

For the successful fabrication of the silicon PFL's being considered in this research, a recipe with extremely low etch selectivity is imperative. By referring back to section 2.3.1 where DRIE is described, recall that etch selectivity is simply a ratio of etch rates, namely the silicon etch rate divided by the photoresist etch rate. Therefore, to achieve low etch selectivity, the photoresist etch rate must be increased without a large effect on the silicon etch rate. In order to accomplish such a goal, I will introduce the concept of a modified Bosch process with an Oxygen-only step.

One process that is widely used in MEMS and the IC industry, is an Oxygen ( $O_2$ ) plasma to remove residual photoresist films after lithography. Such a process is advantageous in MEMS because the  $O_2$  plasma removes most organic materials quickly, like photoresist, without affecting underlying materials such as silicon or silicon-dioxide. This fact works to my advantage for two reasons: First, most dry etch tools have  $O_2$  plumbing already available because it is commonly used for surface or chamber cleaning. So no additional modifications to the DRIE system are required. Second, the  $O_2$  plasma

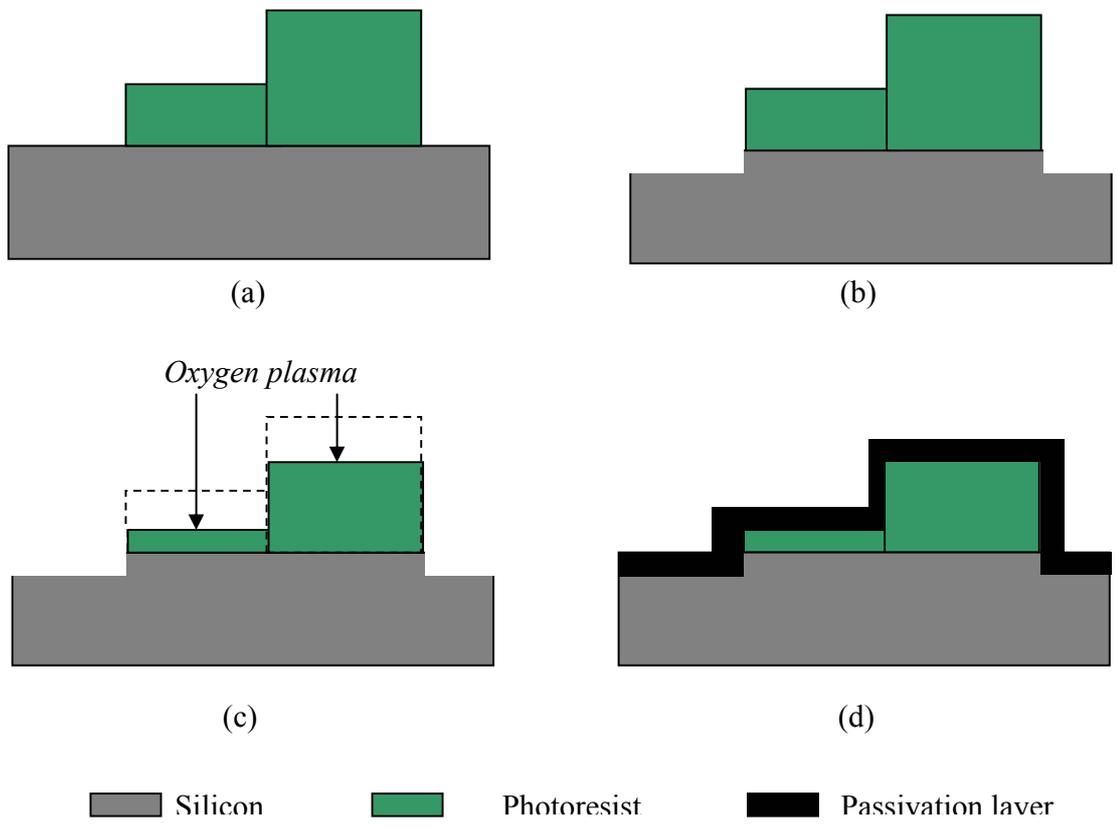
Step	Ar (sccm)	C <sub>4</sub> F <sub>8</sub> (sccm)	SF <sub>6</sub> (sccm)	O <sub>2</sub> (sccm)	Pressure (mTorr)	Chamber Temp(°C)	Electrode Power (W)	Time (sec)
Passivation	40	70	0	0	19	20	1	5
Etch	40	0	100	0	19	20	10	8
Oxygen- step	40	0	0	70	19	20	8	0-6

*Table 4.1: Modified Bosch process using an Oxygen-only step.*

should only attack the photoresist layer, and not the underlying silicon, increasing only the photoresist etch rate.

I have chosen to integrate an Oxygen-only etch step into the standard Bosch process to increase the photoresist etch rate without increasing the silicon etch rate, thus creating a novel modified Bosch process with low selectivity. Table 4.1 shows an etch recipe with an original selectivity of 70:1 that now incorporates an Oxygen-only step. The additional Oxygen-only step also contains a small flow of Ar to help maintain a steady and uniform plasma. The other gasses, C<sub>4</sub>F<sub>8</sub> and SF<sub>6</sub>, were not included in the new step to minimize their chemical reactions with the O<sub>2</sub>, thus keeping the steps as separated as possible.

The goal behind modifying the Bosch process with a short Oxygen-only etch step is to remove a thin layer of photoresist from the top of the 3D photoresist pattern after each passivation and etch step, to increase the overall photoresist etch rate. An exaggeration of this process is shown in Fig. 4.9. Fig. 4.9(a) shows an initial photoresist 3D structure, while Fig. 4.9(b) shows this structure after the first etch step has been completed with a high etch selectivity, leaving the photoresist virtually unchanged. The subsequent Oxygen-only step, shown in Fig. 4.9(c), removes a thin layer of photoresist before moving on to the passivation step of the following cycle, shown in Fig. 4.9(d). By



*Fig. 4.9: The steps of a modified Bosch process. (a) Initial photoresist structure on silicon. (b) High selectivity recipe etches silicon while photoresist is virtually untouched. (c) Oxygen-only plasma step removes a layer of photoresist but does not etch the silicon. (d) Conformal passivation layer is deposited over the entire wafer, and the process is repeated*

modulating the length of the Oxygen-only plasma step, the amount of photoresist removed during each cycle may be changed dramatically, therefore reducing the etch selectivity.

The primary advantage of changing the etch selectivity in this manner is that it can be tuned over a wide range (with minimal effect on sidewall profile since the main etch steps that determine sidewall profile remain unchanged). Table 4.2 shows data taken from 4 samples etched with various lengths of an Oxygen-only step. The etch selectivity was successfully modulated from 70:1 down to 14:1 by increasing the length of the

Oxygen-only step from 0-6 seconds, thus taking a larger slice off the top of the 3D photoresist structures. All etches remained extremely vertical ( $>88^\circ$ ).

Etch	Length of O <sub>2</sub> Step (sec)	Etch Selectivity
1	0	70.4
2	2	37.2
3	4	25.6
4	6	13.7

*Table 4.2: Modulating the length of the O<sub>2</sub> etch step and the resulting etch selectivity.*

Using etch recipes that incorporate the novel Oxygen-only step, I was able to successfully transfer entire photoresist PFL structures into the silicon with depths below 50 $\mu\text{m}$ . Using the appropriate Oxygen-only step length, an etch selectivity of 14:1 was used to transfer the PFL profiles into the silicon with the appropriate vertical amplification of the photoresist profile. An SEM image of a PFL, described as lens ‘A’ in Table 3.3 is shown in Fig. 4.10. An optical profilometer measurement taken of this sample is shown in Fig. 4.11. As mentioned previously, the optical profilometer system being used in this research has the ability to measure the height of a profile around the circumference of a circle defined by the user. The height of the top gray level of this particular PFL sample was measured to be 42.5 $\mu\text{m}$  (stand. dev. = 0.7 $\mu\text{m}$ ) taken around the circle marked in red on Fig. 4.11. This height is remarkably close to the desired height of 42.8 $\mu\text{m}$ , meaning that the etch selectivity achieved with the Oxygen-only step was quite accurate. The circumference measurement has also confirmed that the curved pixilation technique developed in Chapter 3 of this thesis produced a radially-symmetric gray level (which will be confirmed again during section 4.4.2 using the profile evaluation method discussed therein).

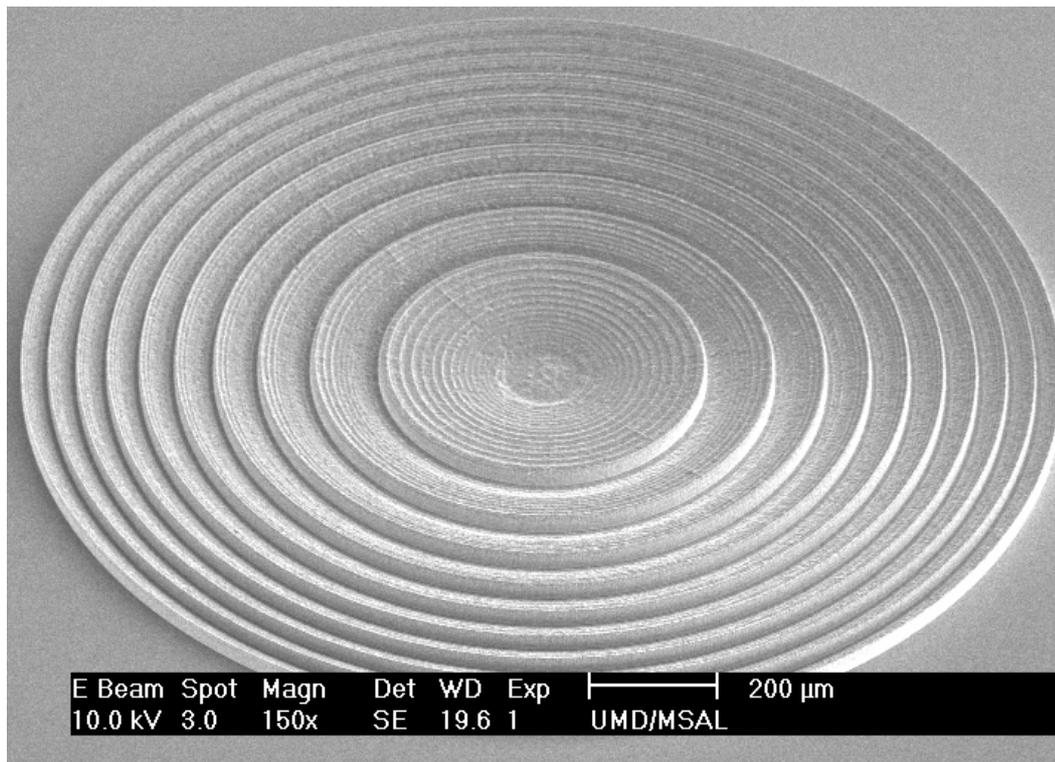


Fig. 4.10: Fully etched silicon PFL fabricated using an etch selectivity of 14:1 achieved through the addition of the Oxygen-only step to the standard Bosch process.

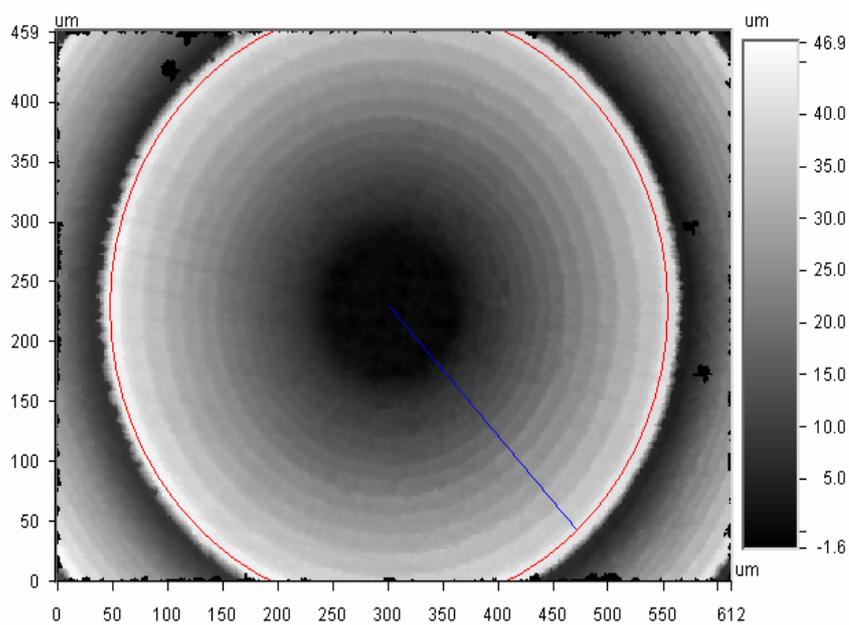
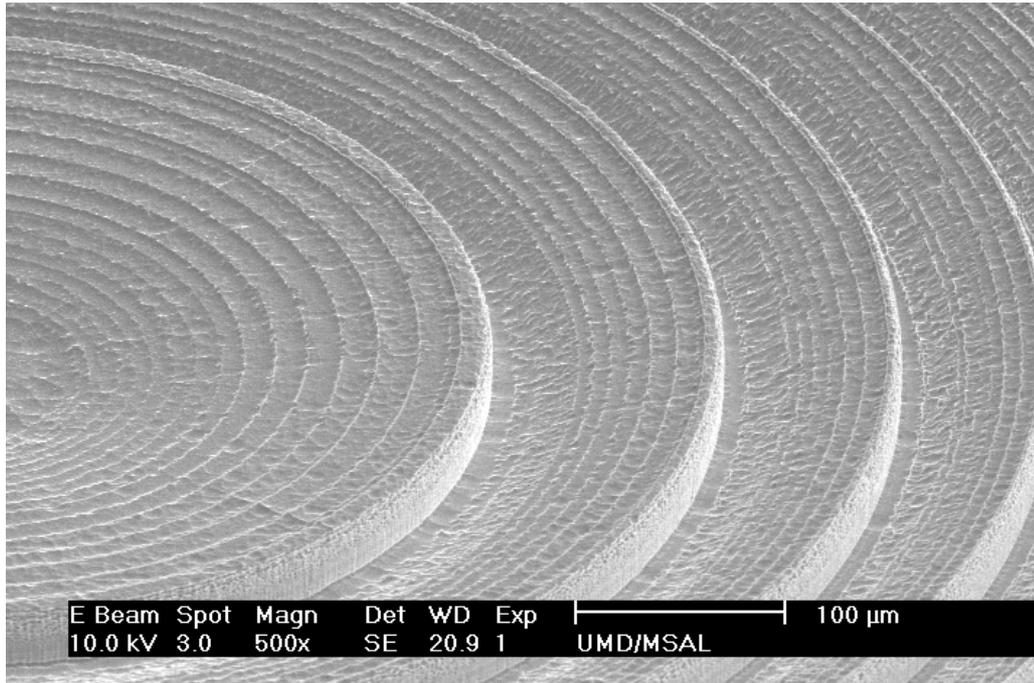


Fig. 4.11: Optical profilometer measurement of center ridge of a silicon PFL. Height data was taken around the circumference of the top gray level, marked in red.



*Fig. 4.12: Close up picture of the PFL shown in Fig. 4.11.*

Fig. 4.12 shows a close-up picture of same PFL shown in Fig. 4.10. This picture assists in pointing out a few major features of these lenses. First, each large gray level is clearly defined in the silicon. Due to the large number of available pixels with small pixel areas, the lower gray levels tend to be close together in the photoresist, blending the lower levels together (actually helping approximate a smooth profile). Additionally, as the silicon is etched deeper, the small amount of isotropic etching present in the DRIE process will slowly smooth the silicon surface, and as a result this effect will be more apparent in the lower gray levels.

The second point that should be taken from Fig. 4.12, is that the surface of each gray level is not smooth (RMS roughness  $\sim 0.2\mu\text{m}$ ), however rough levels should have a minimal impact on lens performance because the index of refraction of silicon at photon energies above 8keV is very close to 1 ( $n \approx 0.999$ ). Similarly, one could argue that the extra phase shift will be negligible through a quick investigation of the phase shift caused

in silicon at X-ray energies. If 8.4keV photons are considered, the thickness of silicon required to produce a  $2\pi$  phase shift, using equation (1.2), is 21.4 $\mu\text{m}$ . If a local section of the lens has an extra 0.2 $\mu\text{m}$  of silicon (due to roughness), the extra phase shift introduced is slightly less than 1%. Since a step approximation to the ideal profile is already being made, an extra 1% phase shift may actually help performance if the step is inducing a non-ideal phase shift at that point anyway. Although the presence of surface roughness on gray levels will be a concern as gray-scale technology is developed for other applications, the surface roughness currently on our PFL structures is acceptable. Should the surface roughness become unacceptable or cause unforeseen effects, it could possibly be reduced using an isotropic etch, or a thermal oxidation and strip, which will be discussed further in Chapter 5.

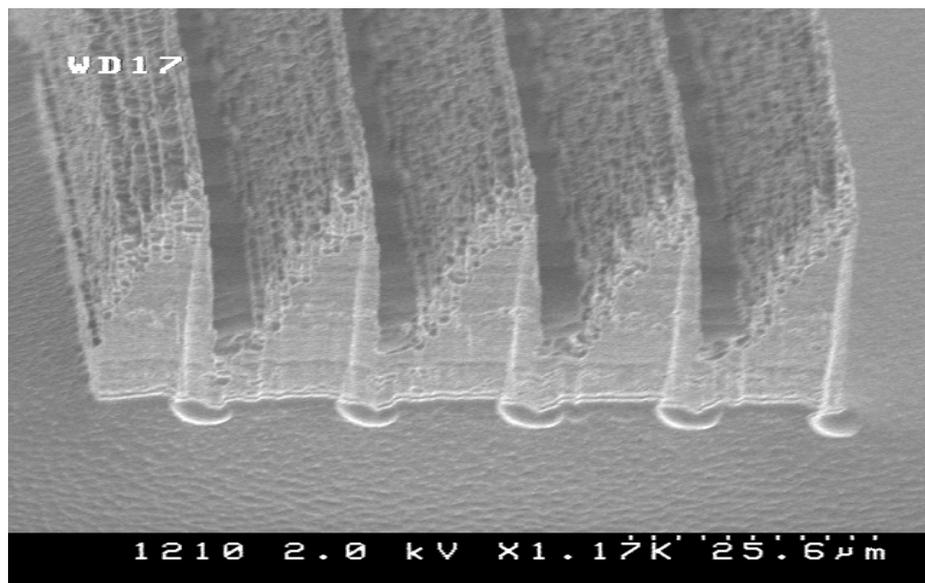
A final feature to notice in Fig. 4.12 is that the backside of each ridge is sharply defined. As discussed earlier, optical profilometry is the only method available to measure the profile in these deeply etched silicon areas with vertical sidewalls. These measurements taken with an optical profilometer have helped quantify another characteristic of our etch recipes, namely that as the ridge width is decreased, the final etch depth in silicon decreases as well. This phenomenon is known as Aspect Ratio Dependent Etching (ARDE) and is discussed in more detail in the following section.

### **4.3.3. *Aspect Ratio Dependent Etching***

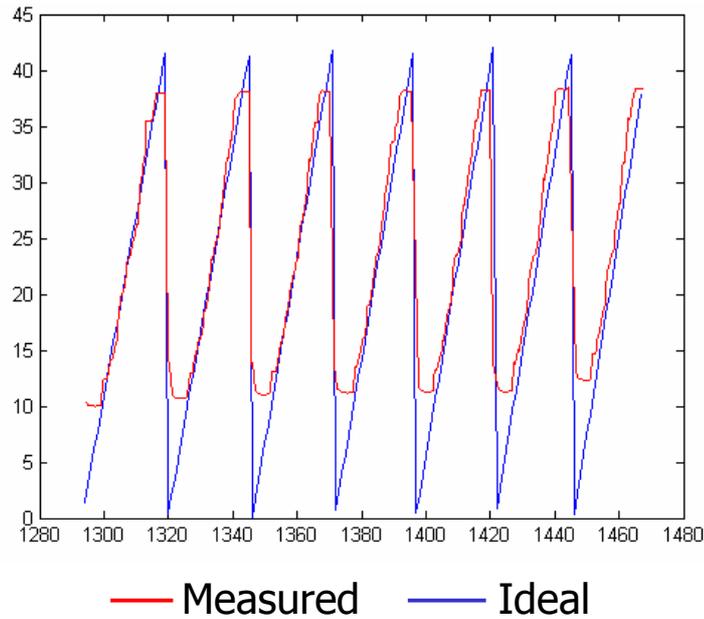
Aspect Ratio Dependent Etching (ARDE) can be observed in virtually any RIE or DRIE application. As discussed in Chapter 2, the majority of the etching in a dry plasma process is driven by chemical reactions. Therefore, the reactive species and ions present

in the plasma must be transported to the surface in order to react with the silicon. For features with the same dimensions, such as trenches, the transport of the reactive species into the trench, as well as the removal of the etch volatile from the trench, will be relatively constant. However, as the feature dimensions change across a single wafer, the reactive species and etch volatiles have a harder time getting in and out of the smaller features. Thus, there are fewer chemical reactions in the thinner trenches, and the etch volatiles are not easily removed after each reaction. The end result is a reduced etch rate as the aspect ratio increases.

For the silicon PFL's being developed in this research, ARDE causes the etch rate of the thin ridges to be smaller than that of the wide ridges. Since there is a gradual change in ridge size over the radius of the PFL, an etch rate gradient has effectively been created, with higher etch rates in the center of the lens and lower etch rates at large radii. Fig. 4.13 shows a cross section of a test structure with ridges  $<20\mu\text{m}$  wide. The bottom of each ridge is obviously not etched as deeply as the surrounding open areas (effectively



*Fig. 4.13: Narrow ridges on test structures did not etch as deeply as the open area and the larger ridges.*



*Fig. 4.14: Optical profilometer measurement of thin ridges compared to the ideal case.*

an infinitely large ridge). By using the optical profilometer, the profile of seven ridges at a radius of 1.3mm were measured and compared to the ideal case, see Fig. 4.14. These measurements show the lower gray-levels to be lagging significantly behind the rest of the etched areas, resulting in heights up to  $10\mu\text{m}$  above the ideal profile. Such a profile mismatch will cause a loss in efficiency for any tested lenses.

To better understand the ARDE effects being observed in the PFL structures, a mask was designed containing a number of rectangular trenches of different widths, between  $5\text{-}150\mu\text{m}$ . Three samples were then etched to various depths near  $50\mu\text{m}$  using 3 different etch selectivities to investigate the relative etch depth as the trench width was changed. Fig. 4.15 is a graph of the etch depth versus trench width for the 3 samples. It is easy to see that the thinnest trenches ( $5\mu\text{m}$  wide) consistently etch  $5\text{-}10\mu\text{m}$  less than the widest trenches ( $150\mu\text{m}$ ), depending on the total etch depth and etch recipe used. This result was expected given the reasoning described above.

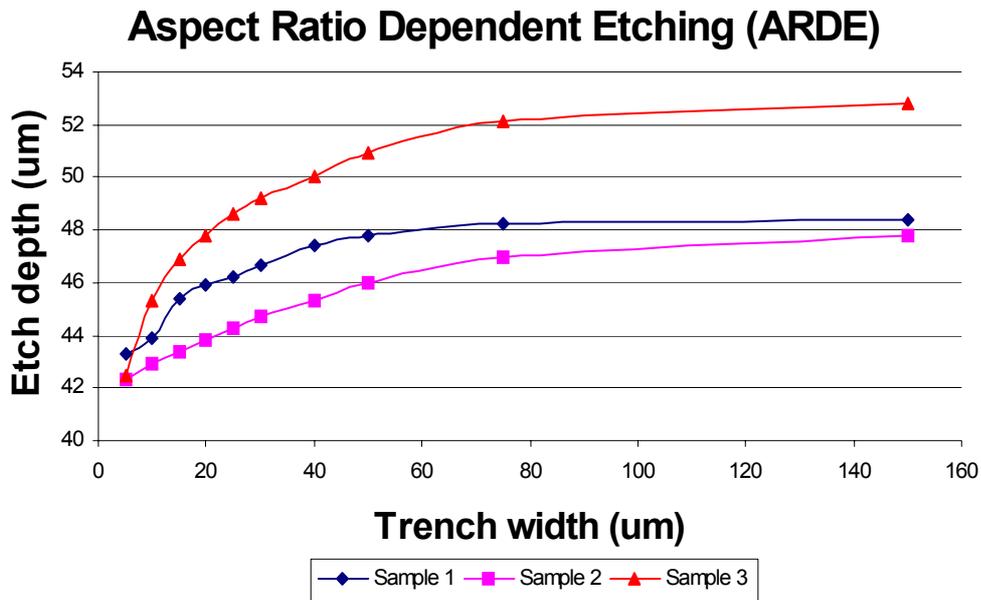


Fig. 4.15: Etch depths measured for trenches with varying widths on 3 samples. Thinner features etch slower, resulting in a reduced etch depth.

The ARDE effect in the PFL test structure shown in Fig. 4.14 was observed to be  $\sim 10\mu\text{m}$ , however the structure measured did not have horizontal dimensions smaller than  $15\mu\text{m}$  wide. When referring to Fig. 4.15, the ARDE effect between large ridges and  $15\mu\text{m}$  wide ridges should have been closer to  $4\text{--}6\mu\text{m}$  depending on the etch recipe and depth. This fact revealed that other aspect ratio dependent effects must be occurring.

The first possibility is that the etch mechanisms of a trench and a sloped ridge are not analogous, namely that the lag in a sloped ridge  $15\mu\text{m}$  wide may be inherently larger than the lag in a rectangular trench of the same width. The transport of etch species in sloped geometries could be causing the etch rate to change locally because the trench is narrower at the bottom of a sloped ridge. Another effect that had not been considered to this point, and could be causing some aspect ratio dependence, is aspect ratio dependent development. Similar to the reasoning of reactive species transport in the dry etching of silicon, as the aspect ratio of the photoresist features is increased, fresh developer

solution will have a harder time getting in and out of the smaller features to remove the exposed photoresist. Therefore, photoresist gray levels located in thin trenches may not be developed as efficiently, resulting in slightly higher photoresist heights. The investigation of these two effects will become increasingly important as the diameter of the PFL is increased, however such investigations fall beyond the scope of this thesis. A solution that has prospects for diminishing, and possibly eliminating, the aspect ratio effects seen in final silicon PFLs will be discussed in Chapter 5 as part of the future work.

#### **4.4. Profile Evaluation (Silicon)**

Now that 3D silicon PFL profiles have been successfully designed and fabricated, a quantitative method must be developed by which the final silicon profile may be evaluated. A new X-ray beam line testing facility is currently under construction at NASA-Goddard Space Flight Center (near the University of Maryland campus), but until the facility is operational, a simple method with the ability to calculate an approximate theoretical PFL efficiency will greatly assist in evaluating each design and fabrication iteration. The following sections will describe the method developed in this research to calculate an expected lens efficiency using simple physical principles. Efficiencies calculated from perfect stepped lens profiles (binary, 4-level, etc) will be shown to agree well with the theoretical efficiencies calculated using equation (1.1). Measurements of silicon profiles will then be used to calculate the expected efficiency of a variety of lenses, including those lenses fabricated using etch recipes with the incorrect etch selectivity. Finally, comments will be made regarding the differences in calculated efficiency for the 3 primary lenses described in Table 3.3.

#### 4.4.1. Method

The method of calculating the expected efficiency of a particular lens is based on ray tracing in Fig. 4.16. The major assumptions in this analysis are as follows: First, I assume an ideal photon source sufficiently distant to approximate that all arriving photons will be perpendicular to the lens surface and have a consistent wavelength, phase, and amplitude. Second, I assume that all energy is collected at the designed focal point, a reasonable assumption since the focal length of a PFL is primarily dependent on the horizontal dimensions of each lens, and not the vertical dimensions being tuned in gray-scale technology. Third, the lens is assumed to have perfect alignment and zero tilt. Finally, photon absorption is neglected because it will be relatively independent of the lens profile. Ideally, these assumptions yield a calculated lens efficiency that is entirely dependent on the profile produced using gray-scale technology.

The first step in estimating the lens efficiency is to measure the silicon profile with an optical profilometer. The data is then output into a text file containing a table of

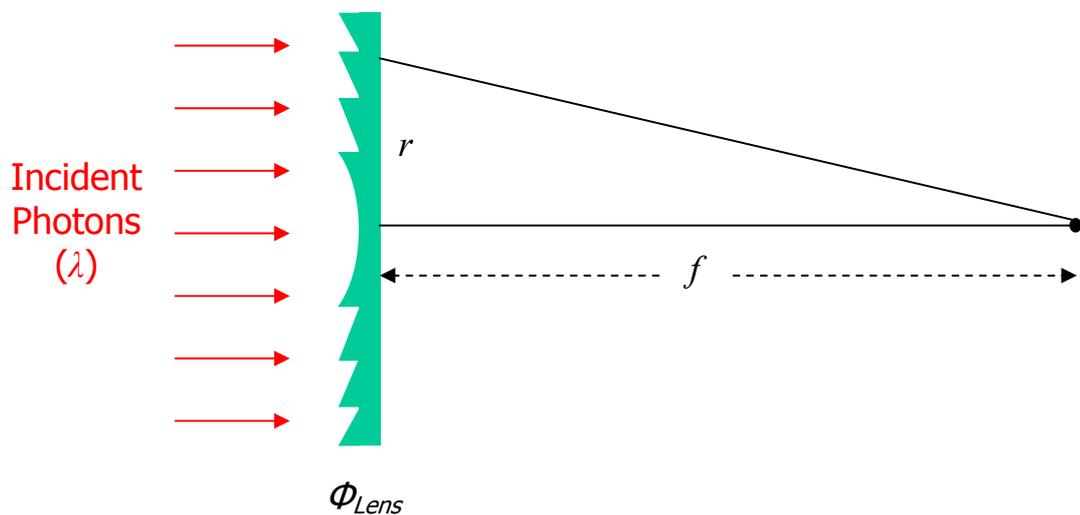


Fig. 4.16: Schematic used as the basis for developing a method to calculate the expected efficiency of a measured silicon profile.

locations and heights. This data is imported into MATLAB, and a virtual photon ( $\vec{a}$ ) is sent “through” each point and propagated to the lens focus. The sum of the virtual photons with different phases will determine the lens efficiency.

The phase shift caused by the local thickness of the silicon lens ( $\Phi_{Lens}$ ) is calculated by:

$$\Phi_{Lens} = 2\pi * \left( \frac{t_M}{t_{2\pi}} \right) \quad (4.1)$$

where  $t_M$  is the measured lens thickness and  $t_{2\pi}$  is the thickness of silicon required to produce a phase shift of  $2\pi$ . The phase shift caused by the change in path length ( $\Phi_{Geom}$ ) from the radial point ( $r$ ) to the focal point ( $f$ ) is then calculated by:

$$\Phi_{Geom} = 2\pi * \left( \frac{\sqrt{r^2 + f^2}}{\lambda} \right) \quad (4.2)$$

where  $\lambda$  is the photon wavelength in meters. Finally, the phase of the arriving photon at the focal point is the difference of the two phase shifts:

$$\Phi_{Total} = \Phi_{Geom} - \Phi_{Lens} \quad (4.3)$$

Once the phase of each photon arriving at the focus is calculated, they are summed together as unit length phasors with their respective  $\Phi$  values, as shown for 2 phasors in Fig. 4.17. The efficiency ( $\eta_{Lens}$ ) is estimated by normalizing the resultant vector by the number of phasors sent ‘through’ the lens, and squaring that value:

$$\eta_{Lens} = \left( \frac{\left| \sum_{k=1}^n \vec{a}_k \right|}{n} \right)^2 \quad (4.4)$$

Ideally, if each phasor has an identical phase as it arrives at the focus, the phasors will add perfectly (as if simply adding their magnitudes) and the lens efficiency will be 100%.

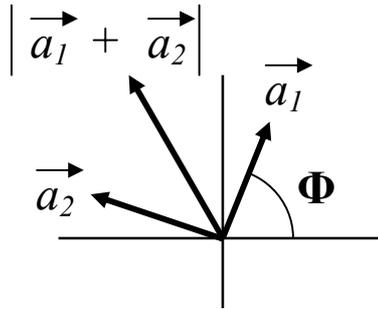


Fig. 4.17: Phasor addition. Normalized resultant magnitude determines efficiency.

To verify that the method described above is valid for calculating expected lens efficiencies for different profiles, the MATLAB program was used to calculate the efficiency of a number of ideal stepped PFL profiles. Binary, 4-level, and 8-level stepped PFL's, as well as an ideal PFL profile, were all defined in MATLAB and the efficiency of each was calculated. Table 4.3 compares the MATLAB calculated efficiencies for each profile to the theoretically predicted efficiencies using equation (1.1). It is evident from the table that the efficiencies calculated in the MATLAB program are quite close to the theoretical values, within tenths of one percent as the number of levels increases. The probable sources of error are (1) accumulated error from the finite precision used in

PFL profile	Lens Efficiency	
	Theory	MATLAB
Binary	40.4%	39.2%
4-Level	81.0%	80.5%
8-Level	95.0%	94.6%
Ideal	100.0%	99.6%

Table 4.3: Theoretical versus calculated lens efficiencies for various lens profiles.

MATLAB (16 decimal places), and (2) the lens profile equations already contains a slight approximation since the constant coefficient in equation (3.2) is only carried to 1 decimal place. For the purposes of this research, Table 4.3 confirms the method of calculating lens efficiency described in this section is sufficient to evaluate each silicon PFL profile.

#### **4.4.2. Profile Measurements**

As mentioned before, an optical profilometer scan of a particular silicon PFL will be taken to obtain a list of radial locations in ‘x’ and heights in ‘y.’ One problem that became apparent as these measurements progressed was the definition of a common origin. The optical profilometer takes the origin of the scan to be the first data point, however the first data point ( $x=0$ ) is almost certainly not actually located at the exact center ( $r=0$ ) of the lens design. Thus the entire measured profile must be shifted until each ‘x’ point has been translated to correspond to the appropriate ‘r’ on the lens design. To accommodate this, the MATLAB program was modified to sweep through a large range of offsets and calculate the lens efficiency with each offset. When the measured profile is shifted the appropriate amount, the lens efficiency should be at a maximum.

An example optical profilometer 3D measurement is shown in Fig. 4.18. Once each measurement is taken, an arbitrary “SCAN” may be defined with any direction and length. The 2-D data is output to a text file. Table 4.4 shows the resulting efficiency for 5 scans taken on a single 3D measurement. Each scan began at the center of the lens and scanned at the specified angles for the first 3 ridges (angles are approximate, and measured from a horizontal line across the width of the page). The calculated efficiencies for these 5 scans are grouped quite closely. Thus, within a small amount of

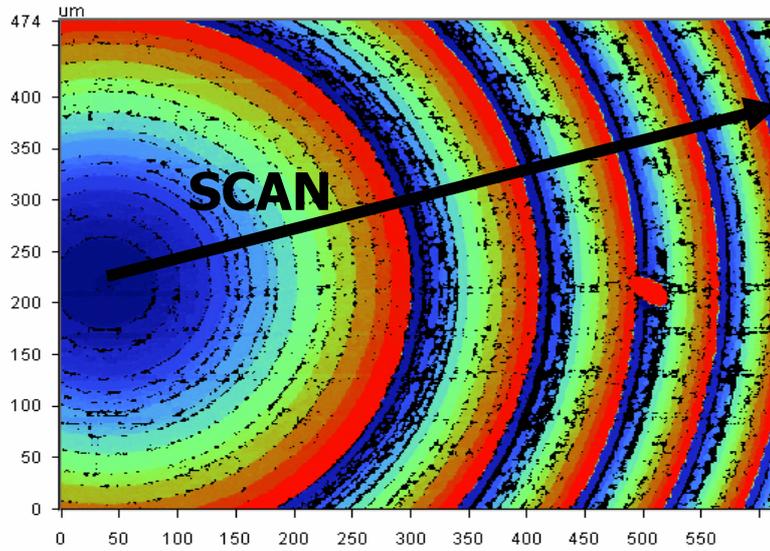


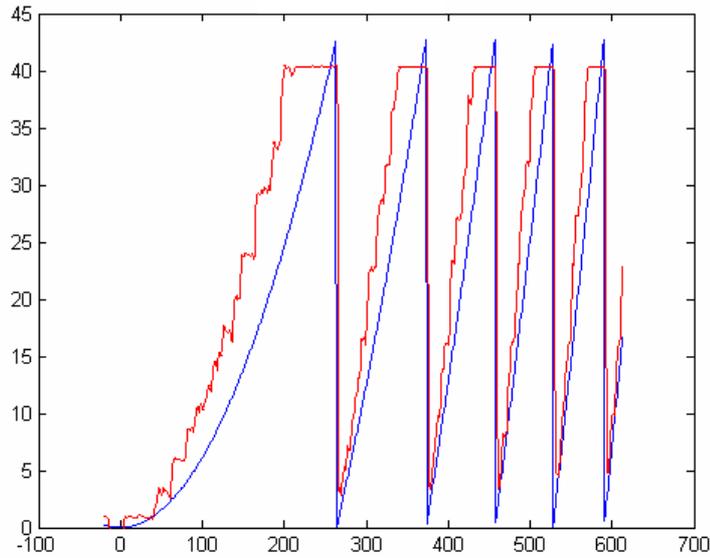
Fig. 4.18: Optical profilometer measurement of a silicon PFL. 2-D scans of arbitrary length and orientation may be output into a text file.

Angle	Calculated Efficiency
0°	84.7%
20°	83.1%
45°	85.0%
70°	84.2%
90°	84.2%

Table 4.4: Calculated efficiencies of a PFL with each radial scan taken at a different angle from the horizontal.

process variation and non-uniformity, we may assume the lens is in fact radially symmetric. Therefore, for an estimate of the efficiency of each lens profile, only a single 2-D data set must be obtained, as it is assumed that similar efficiencies will be calculated using alternate scan directions.

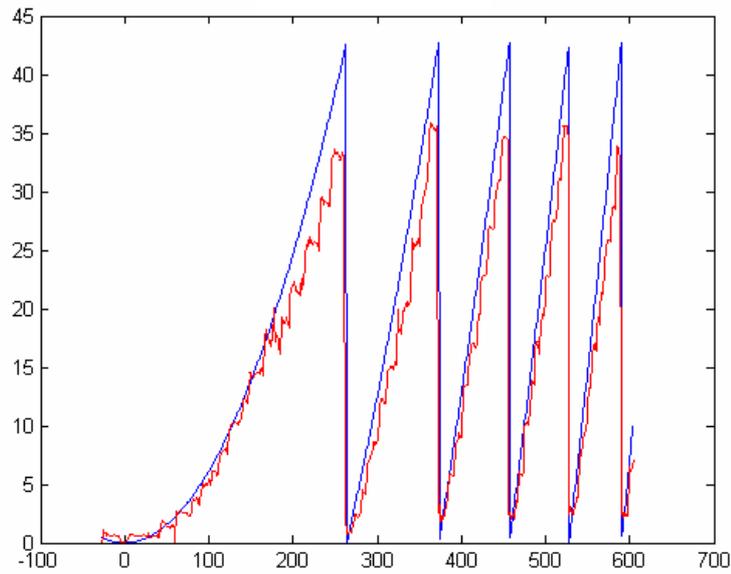
The importance of etch selectivity control is underscored by comparing the calculated efficiency for 3 lenses (described in Table 3.3 as lens ‘C’) that have been processed with different etch selectivities. Their profiles are shown in Fig. 4.19 – 4.21. First, Fig. 4.19 shows a sample that has been etched with a selectivity that was too high.



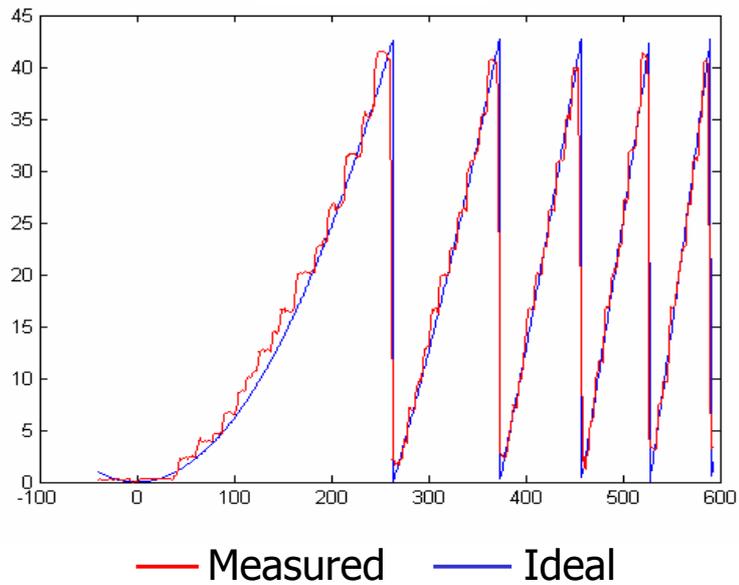
*Fig. 4.19: Silicon PFL etch with a higher etch selectivity than desired. Calculated efficiency of this lens was 19.3%*

The sample was actually taken out of the DRIE system at the appropriate etch depth before the entire pattern was transferred. As shown in Fig. 4.19, the measured profile is similar in shape to the ideal lens, but the vertical dimensions are wrong. The phase shift induced by the extra silicon results in a calculated efficiency of only 19.3%.

Fig. 4.20 shows a sample that has been etched with an etch selectivity slightly lower than desired, transferring the entire photoresist pattern prematurely. The slight non-uniformity of the photoresist and pixelated approximation can be seen in the uneven tops of the ridges (unlike Fig. 4.19). The resulting calculated efficiency for this sample, etched with a lower etch selectivity, was 68.7%. Fig. 4.21 shows a sample etched to the desired depth, with the appropriate etch selectivity. The sample was taken out of the DRIE system just before the top photoresist gray level had transferred, resulting in relatively uniform ridge tops. Due to precise etch selectivity and mask design, the ideal and measured profiles are extremely close, resulting in a calculated efficiency of 87.9% (compared to an ideal calculated efficiency of 94.6% for an 8-level lens in Table 4.3).



*Fig. 4.20: Silicon PFL etch with a selectivity slightly lower than desired, entire photoresist pattern is transferred too early. Calculated efficiency for this lens was 68.7%.*



*Fig. 4.21: Silicon PFL etched with the appropriate etch selectivity compared to the ideal case. The calculated efficiency of this lens was 87.9%.*

These calculated efficiencies are significantly higher than the efficiencies of Fresnel lens derivatives fabricated by other groups (see Chapter 1). Even after a reasonable amount of performance loss introduced by a testing setup, the measurements

presented above should provide a good starting point for estimating the lens performance in an X-ray beam line. It must be noted that the efficiencies quoted above are only accurate for a lens with a radius equal to the scan length. As the lens radius increases, the outer ridges will become progressively less efficient because of the ARDE effect discussed previously, lowering lens performance.

Die	Calculated Efficiency
2	84.8%
4	87.3%
5	86.4%
6	87.9%
8	83.1%

*Table 4.5: Calculated efficiencies of PFLs at different locations across a 75mm wafer.*

Since the demonstration of the imaging properties of a PFL may be accomplished with a small number of high quality PFL's, it is not necessary to produce large quantities of identical lenses. Nonetheless, uniformity is a chief concern for the integration of gray-scale technology into other applications. If the uniformity can be controlled sufficiently, coded PFL arrays may be possible to investigate alternative high energy telescope arrangements. Table 4.5 shows the calculated efficiency of 5 PFLs (lens 'C' in Table 3.3) at 5 different locations on a 75mm wafer. The range in efficiencies is <5% across the wafer. With further process optimization, PFL uniformity across the wafer should improve slightly. These high efficiencies should be sufficient to enable coded PFL arrays.

Returning to the ARDE issue, I am now able to quantify the effective decrease in efficiency resulting from the reduced etch rate seen in smaller ridges. The efficiency for the structures shown in Fig. 4.14 was calculated to be only 57%, while the center of the

same lens was calculated to have an efficiency of over 85%. Thus, the ARDE problem must be improved before PFLs of increased radius with high efficiency can be realistically fabricated. A possible solution will be outlined in Chapter 5 of this thesis as part of the discussion on future work

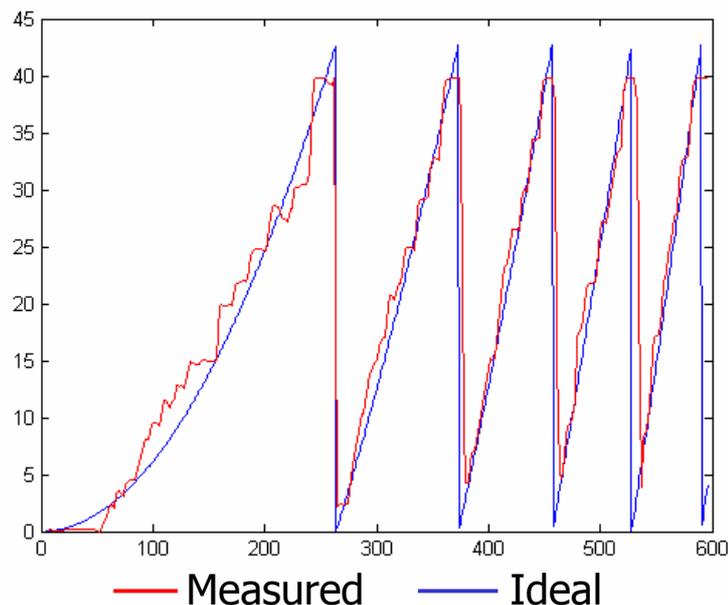
#### **4.4.3. Lens design comparison**

Finally, the method of calculating profile efficiency is used here to compare and contrast 2 main parameters used in the optical mask design for the lenses described in Table 3.3. First, the accuracy of the Gaussian approximation and the choice of a pixel set will be discussed as they effect the calculated efficiency, using lens designs ‘A’ and ‘B’. Second, lens designs ‘B’ and ‘C’ will be used to investigate the difference in calculated efficiency resulting from using a different number of gray levels to fabricate each ridge. The difference in calculated efficiency at the outer edges of each lens will not be directly compared due to the non-linear ARDE effects that have already been discussed at length and are not a product of the optical mask design.

Referring back to Table 3.3, lens ‘A’ and ‘B’ have identical focal lengths, etch depths, and number of levels. The main differences between these two lenses are the lens diameter and the Gaussian approximation. In the Gaussian approximation, the fit parameters,  $A_0$  and  $\gamma$ , are slightly different for the two lenses, and although the number of levels used were the same, the pixel sets were different. By measuring only the inner 600 $\mu\text{m}$  profile, we are able to isolate the effect on the calculated efficiency from improving the Gaussian approximation and pixel set. Fig. 4.22 shows an optical profilometer measurement of lens ‘A’ which had a calculated efficiency of 66.5%. Fig.

4.23 shows an optical profilometer measurement of lens ‘B’ which had a calculated efficiency of 81.7%. Thus, by improving the accuracy of the Gaussian approximation and the pixel set, an increase in calculated efficiency of roughly 15% was achieved.

The improvement in calculated efficiency is attributed to two primary factors: First, as discussed in section 3.4.2, the pixel set used to fabricate lens ‘B’ did not include some of the larger pixels whose  $Tr$  values had large separations. Looking at Fig. 4.22, the top 3-4 levels are spaced quite far apart (vertically) while the bottom levels are bunched together. By replacing the low  $Tr$  pixels with pixels having intermediate  $Tr$  values, the levels may be evenly distributed, enabling higher lens efficiencies. Second, now that the pixel set does not include the low  $Tr$  pixels, the Gaussian fit parameters were adjusted to improve the curve fit for only the remaining pixels. Thus, the “bubble” in the profile seen in the center of the first ridge in Fig. 4.22 is reduced in Fig. 4.23. Further improving the accuracy of the Gaussian fit (as discussed in section 3.4.1) could

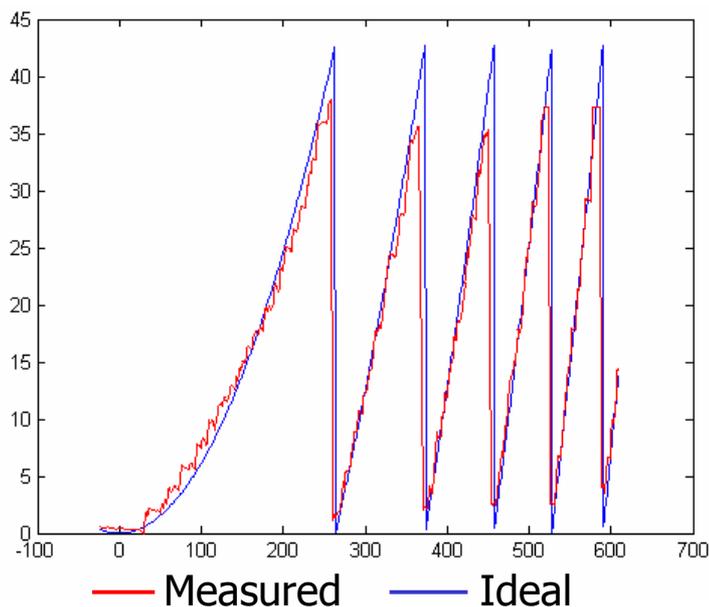


*Fig. 4.22: Optical profilometer measurement of a silicon lens ‘A’ design versus the ideal case. Calculated efficiency of this lens was 66.5%.*

eliminate any small remaining “bubble” in the profile, but the Gaussian/exponential form of the approximation must be kept for easy integration into the mask layout design.

Comparing lens ‘B’ and lens ‘C’, they both use the more accurate Gaussian fit parameters, and are designed for identical focal lengths and etch depths. The primary differences are again in the lens diameter, which will be ignored by only measuring the center of the lens, and in the number of gray levels used to approximate the profile. Lens ‘B’ utilizes 32 total gray levels, or 16 levels per phase depth, while lens ‘C’ utilizes half as many gray levels, 16 total and 8 per phase depth. Again, lens ‘B’ with an efficiency of 81.7% is shown in Fig. 4.23 and lens ‘C’ with an efficiency of 87.9% was shown previously in Fig. 4.21.

In comparing the two figures, it can be seen that the height of lens ‘B’ was just shy of the height of lens ‘C,’ which would contribute to some of the difference in calculated efficiency. However, these structures were both located on the same wafer and underwent the same processing steps, and are typical examples of the structures seen



*Fig. 4.23: Optical profilometer measurement of a silicon lens ‘B’ design versus the ideal case. Calculated efficiency of this lens was 81.7%.*

across the wafer. The primary conclusion I would like to draw from these two designs is that there is no significant gain in the calculated efficiency of profiles designed using 16 levels per phase depth versus 8 levels per phase depth. (The theoretical efficiency difference between ideal 16 level and 8 level designs using equation (1.1) would be 98.7% vs. 95.0%). In fact, the lens ‘C’ structures (with only 8 levels per phase depth) on a single wafer had calculated efficiencies that were consistently ~5% more efficient than the lens ‘B’ structures on the same wafer (16 levels per phase depth). One possible reason for lens ‘C’ to have equal to or higher efficiencies is that larger gray levels are more sharply defined and thus retain their expected heights with greater integrity. Additionally, as the width of the ridges is decreased, it will no longer be possible to fit every pixel in every ridge. This limitation means that as the radius increases, even though there are 16 pixels per phase depth available, not all levels will be used in thin ridges, and we are essentially forced into an 8 level per phase depth design.

In summation, both lens ‘B’ and lens ‘C’ have significantly higher calculated efficiencies than lens ‘A’, and the calculated efficiency of the lens ‘C’ design is consistently equal to or higher than the lens ‘B’ design, even though lens ‘B’ utilizes a larger number of gray levels. Since ridge widths decrease as the PFL radius increases, not every level in the larger gray level set will be used in each ridge, essentially reducing a large gray level set into a smaller gray level set. Therefore, I conclude that gray-scale PFL designs using more than 8 gray levels per phase depth will exhibit a negligible gain in efficiency, if any at all.

## **4.5. Summary**

This chapter has presented all fabrication aspects of developing a silicon PFL using gray-scale technology. Lithography results were obtained showing the successful fabrication of photoresist PFL structures. Limitations were discussed regarding the intensity profiles created as the ridge dimensions are reduced and fewer pixels are used to define each ridge. Optical profilometer measurements of photoresist profiles confirmed that the Gaussian approximation method introduced in Chapter 3 of this thesis is indeed a valid method for accurately defining 3D photoresist profiles. Pattern transfer of the 3D photoresist structures with extremely low etch selectivity was achieved through the introduction of a novel Oxygen-only etch step to the standard Bosch process. By modulating the length of the Oxygen-only step, the etch selectivity may be tuned over a wide range without large effects on other etch characteristics. Finally, a method was developed to estimate the efficiency of different silicon PFL profiles measured with an optical profilometer. These calculated efficiencies were used to evaluate and compare the different designs and fabrication steps developed in this thesis. Chapter 5 will summarize the work presented in the first 4 chapters, as well as discuss areas of design and fabrication that will be addressed as part of the future work involved in this research.

## 5. Summary and Future Work

### 5.1. Summary

This thesis has outlined the steps taken to develop a phase Fresnel lens in silicon with modest focal lengths ( $\sim 100\text{m}$ ) for soft X-ray energies using gray-scale technology. PFLs fabricated with this technology have the potential to exhibit high angular resolution and high efficiency, as well as serve as the basis of an investigation into larger space-bound PFL telescopes for gamma-ray energies. Further development of the techniques and issues discussed in this thesis should produce PFLs appropriate for testing in an existing X-ray beam line at NASA XRCF in Huntsville, Al, or a beam line currently under construction at NASA Goddard Space Flight Center (GSFC) in Greenbelt, MD. If ground tests of a silicon PFL are successful, and high angular resolution and high efficiency are achieved, then we will have taken the first step towards a high efficiency PFL telescope with the ability to probe nearby active galactic nuclei, event horizons around black holes, and other high energy phenomena with unprecedented accuracy.

The fabrication of the PFL's presented in this thesis was achieved through the use of gray-scale lithography and deep reactive ion etching (DRIE), two MEMS fabrication techniques. Gray-scale lithography enables complex 3D structures to be fabricated in one lithography step using standard processing equipment. By manipulating the placement of various sized opaque pixels on an optical mask, an intensity pattern is generated on the photoresist surface that will form the 3D structure in the photoresist after development. This 3D photoresist structure is then transferred into a silicon substrate using DRIE to create the final silicon structure.

Analysis was presented regarding the design of individual pixels given mask fabrication limitations, and the construction of a pixel set as it effects the design and fabrication of a PFL. Millions of pixels are required to produce any particular structure, and each must be selected and placed with high accuracy using an automated system. A solution to this problem was the development of the Gaussian approximation method, introduced in Chapter 3. Based on a calibration mask, the Gaussian approximation enables a simple relation between the pixel size and the corresponding height in photoresist on the wafer. This relationship is easily reversed to calculate the pixel size required to produce a desired photoresist height. A C-program queries each pixel location, and using the desired PFL profile as an input, selects the pixel with a size closest to ideal. Since the intensity pattern is generated by pixels of finite width, a digital nature of the approximation was apparent in the photoresist, causing non-uniform photoresist profiles of thinner ridges. To minimize this effect, the PFL design was modified to incorporate a  $4\pi$  phase depth. The primary advantage of a deeper phase depth design is that the width of each ridge is essentially doubled in the photoresist, making it easier to achieve a uniform gradient intensity profile. An additional advantage of selecting an increased phase depth design is that the required etch depth is larger, increasing the necessary etch selectivity (helpful considering the low etch selectivity already required to transfer an entire 3D photoresist pattern in shallow depths).

Achieving the proper etch selectivity is paramount to the success of gray-scale technology. As such, the DRIE system at ARL was characterized by varying a number of DRIE etch recipe parameters in the standard Bosch process in an attempt to change and control the etch selectivity. Although the etch selectivity was modulated, it was often at

the expense of other etch properties, such as sidewall profile, and the etch selectivity was still not low enough. The desired etch depth of a  $4\pi$  phase depth design ( $43\mu\text{m}$ ) required an etch selectivity of 20 or below (assuming  $2\text{-}3\mu\text{m}$  resist), so the etch selectivity had to be decreased dramatically by a new method. In this thesis, I therefore introduced the concept of a modified Bosch process with an Oxygen-only step. This modified process adds an additional step to the Bosch cycle, which uses an Oxygen plasma to increase the etch rate of the photoresist and not the silicon, thus decreasing the etch selectivity. Using this new process, recipes with low etch selectivity were achieved. Furthermore, by simply modulating the length of the oxygen step, the etch selectivity could be tuned over a wide range without significant effect on sidewall profile.

Once PFL profiles had been fabricated, metrology methods for measuring both photoresist and silicon profiles were discussed. Contact profilometry was deemed appropriate for measuring photoresist profiles before pattern transfer into the silicon since it does not disturb the photoresist profile. Optical profilometry was then used to measure the high aspect ratio ridges of each silicon PFL after DRIE that could not be measured with contact profilometry. Ray-tracing was employed to develop a method of estimating PFL efficiency from a silicon profile. This method was shown to agree well with theoretical predictions of ideal stepped profiles. Optical profilometer measurements were subsequently imported into a mathematical programming tool (MATLAB) to calculate the expected efficiency of each silicon profile.

The maximum calculated efficiency of a measured 8 level per phase depth gray-scale profile was 87.9% (an ideal 8 level per phase depth profile has a maximum calculated efficiency of 94.6%). Thus, the PFLs developed in this thesis should exhibit a

large increase in efficiency over those fabricated in previous work. I believe a combination of multiple effects caused the discrepancy between the measured and ideal profile efficiencies. First, the etch selectivity and lithography processing were not perfect, and likely never will be. Thus, small additional phase shifts from a high etch selectivity will add up quickly and deteriorate lens performance. Second, the available gray levels are not spaced evenly in the vertical domain, as is assumed in the ideal max efficiency calculation, meaning the step approximation of the curve is uneven causing a small decrease in efficiency. Third, the Gaussian approximation could be improved using offsets to more accurately predict photoresist heights. Fourth, lingering aspect ratio dependent effects cause the final etch depth of thinner ridges to be reduced. The reduction in depth of thinner ridges causes a poor approximation to the desired function and deteriorates PFL efficiency.

## **5.2. Future Work**

As these silicon PFLs move towards a testing phase, there are two primary issues that must be addressed as part of the future work in this research. First, the ARDE problem currently causes a large decrease in efficiency for thinner ridges. Thus, smaller diameter lenses with wide ridges are preferred to produce high efficiency lenses. However, the desire for high efficiency must be balanced with the desire for large diameter lenses which will collect more photons during a beam line test. Second, the attenuation of photons in a bulk silicon material may not be neglected. If the bulk silicon substrate is not thinned beneath the PFL, all photons will be absorbed in the bulk and no imaging will take place. Proposed solutions to both problems are outlined below.

### 5.2.1. ARDE Compensation

The tendency for thinner ridges to exhibit a certain degree of vertical lag was shown in Chapter 4 to severely deteriorate PFL efficiency at larger radii, effectively limiting the final diameter of high efficiency PFLs. However, these PFLs are being fabricated with gray-scale technology, which is already being used to define the vertical dimensions of each lens. The current problem is somewhat analogous to Fig. 5.1(a) which shows uniform photoresist heights for each ridge that exhibits lag once transferred into the silicon. Using the flexibility afforded by gray-scale technology, this lag may be compensated for in the optical mask design. By designing wider ridges to have higher bottom gray levels than thinner ridges, as shown in Fig. 5.1(b), the thinner ridges will begin etching before the wider ridges. Thus, by anticipating the ARDE effect the optical mask design can be modified to give thinner ridges a head start during the etching process. When the etch is completed, the thinner ridges will have effectively etched for a longer amount of time, and ideally to the same depth as the wider ridges.

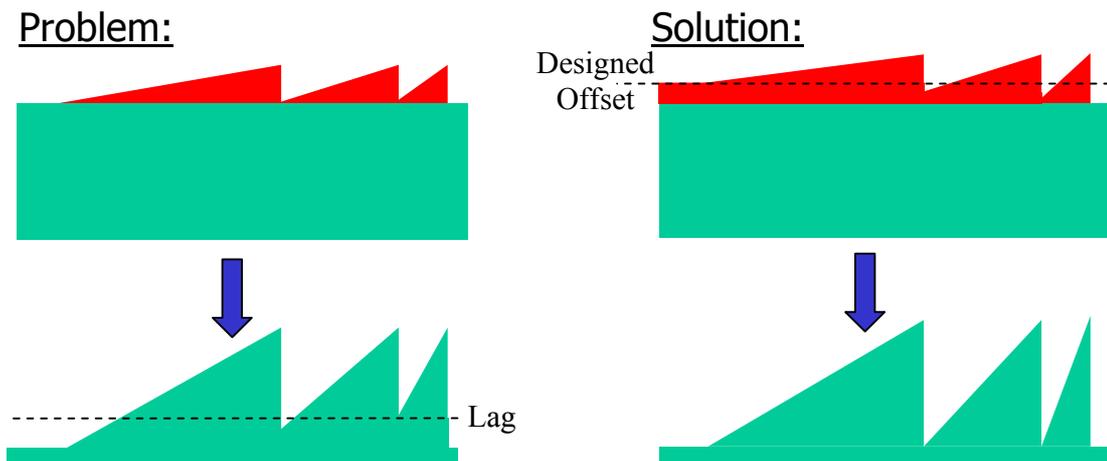


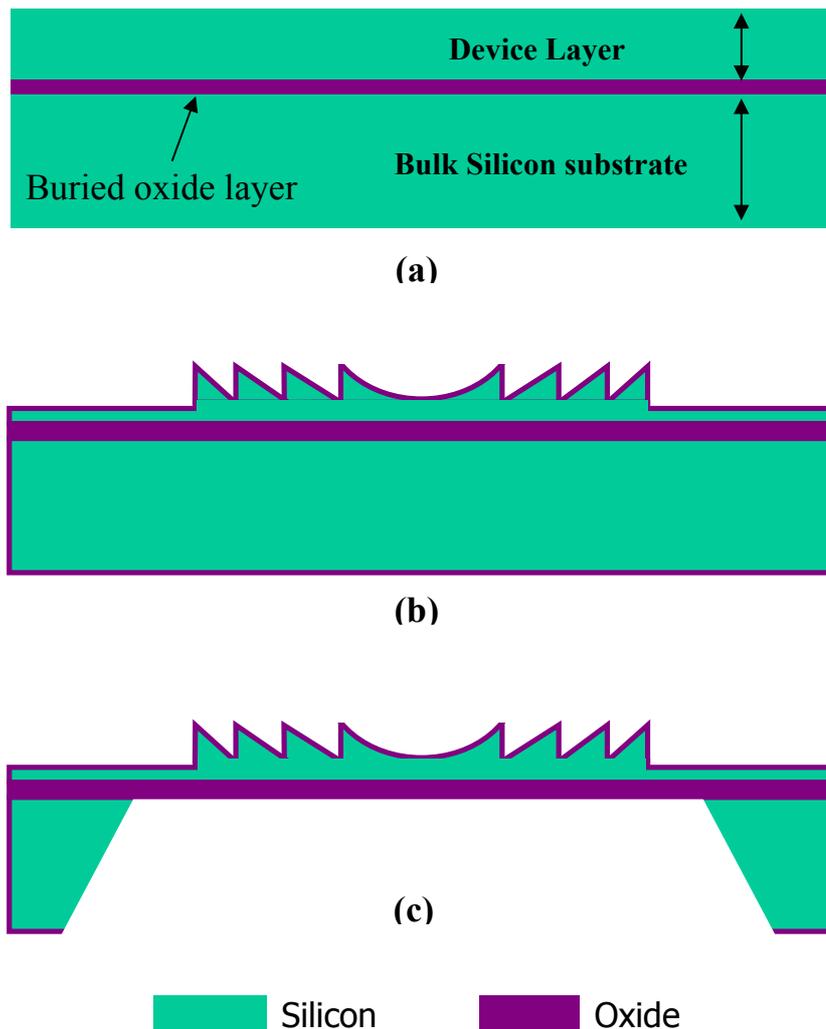
Fig. 5.1: Possible method of reducing aspect ratio dependent effects. (a) Thin ridges in an ideal photoresist pattern are not etched as deeply into the silicon as the larger ridges are etched. (b) An offset in the photoresist gives the thinner ridges a head start and all ridges etch to the appropriate depth.

A starting point to compensate for the aspect ratio dependence can be found in the ARDE measurements shown in Fig. 4.15. Although these measurements are for rectangular trenches, and will not account for geometrical effects, Fig. 4.15 does exhibit the fact that the severity of the ARDE effect changes with trench width. Thus, any calibration solution must be a continuous function of radius because a PFL profile exhibits a continuous change in ridge width as the radius increases.

### **5.2.2. Bulk Silicon Removal**

Due to the photon attenuation length of X-rays in silicon, the removal of the bulk silicon substrate is essential for the eventual testing in an X-ray beam line of PFLs fabricated using gray-scale technology. For the energies being considered in this thesis ( $\sim 10\text{keV}$ ) the photon attenuation length is approximately  $100\mu\text{m}$ . Therefore, a  $400\mu\text{m}$  thick wafer will block all incident photons before they reach the  $40\mu\text{m}$  thick PFL on the opposite side. A method of removing the bulk silicon selectively below the PFL is proposed in Fig. 5.2. The starting material is a silicon-on-insulator (SOI) wafer, which has two silicon layers separated by a thin silicon-dioxide layer, said to be ‘buried’ beneath the ‘device layer’, see Fig. 5.2(a). The thickness of the device layer can be specified when ordering from a particular vendor, and should be selected to be slightly larger than the thickness of the final PFL to be fabricated. Since the oxide layer is buried, the gray-scale lithography process and the DRIE transfer should remain unchanged.

After PFL fabrication, a thermal oxidation step could be used to grow a thin ( $\sim 2000$  angstrom)  $\text{SiO}_2$  layer over the entire wafer, as shown in Fig. 5.2(b). Using front-to-back alignment, the  $\text{SiO}_2$  layer could be patterned on the backside to open windows



*Fig. 5.2: Bulk silicon removal process. (a) Begin with Silicon-on-Insulator (SOI) wafer with device layer slightly thicker than desired lens thickness. (b) Fabricate PFL using gray-scale technology as described earlier, and grow a small thermal oxide layer. (c) Pattern oxide on backside of wafer and use TMAH wet etch to remove silicon bulk beneath the PFL. Buried oxide layer serves as an etch stop.*

beneath each PFL, creating a mask for a deep etch of the silicon substrate, with the buried oxide layer acting as an etch stop. This deep etch step could be done utilizing either DRIE or a wet anisotropic etch with tetra-methyl-ammonium-hydroxide (TMAH), which has extremely high selectivity of silicon to oxide. The final cross section should look like Fig. 5.2(c). The result of this process will be a gray-scale silicon PFL suspended on a local SiO<sub>2</sub> membrane with a small amount of silicon remaining on the device layer

(which can be adjusted by buying wafers with the appropriate device layer thickness). Thus, the absorption of the bulk silicon is minimized without losing the structural benefit of a 400 $\mu$ m wafer for mounting and handling the lenses.

### **5.3. Conclusion**

This thesis has successfully demonstrated gray scale technology as a valid method for creating accurate 3D PFL profiles in silicon. A new Gaussian approximation method has been introduced that is capable of defining complex profiles in a single lithography step. Methods of tuning etch selectivity during DRIE were investigated, and a Bosch process has been modified to include a unique Oxygen-only step to achieve uncommonly low etch selectivities. Moreover, the techniques developed herein can be extended to any application requiring precise definition of arbitrary 3D profiles in silicon. PFL profile measurements were then combined with an analytical model to calculate the expected efficiency of fabricated PFL profiles. The PFLs developed in this research have not yet been tested in an X-ray beam line, however early measurements indicate that they should perform with high efficiency in ground based testing. Such tests will set the stage for the next generation of space-bound gamma-ray telescopes to be developed by NASA that promise increases of many orders of magnitude in sensitivity and angular resolution.

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